

Dinitrogen as a universal electron acceptor in solid-state chemistry: an example of uncommon metallic compounds $\text{Na}_3(\text{N}_2)_4$ and NaN_2

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ABSTRACT: With the exception of Li, alkali metals do not react with elemental nitrogen neither at ambient conditions nor at elevated temperatures, requiring the search for alternative synthetic routes to their nitrogen-containing compounds. Here using a controlled decomposition of sodium azide NaN_3 at high pressure conditions we synthesize two novel compounds $\text{Na}_3(\text{N}_2)_4$ and NaN_2 both containing dinitrogen anions. NaN_2 synthesized at 4 GPa might be the common intermediate in high-pressure solid-state metathesis reactions where NaN_3 is used as a source of nitrogen, while $\text{Na}_3(\text{N}_2)_4$ opens a new class of compounds, where $[\text{N}_2]$ units accommodate a non-integer formal charge of -0.75. This finding can dramatically extend the expected compositions in other group 1-2 metal-nitrogen systems. Electronic structure calculations show the metallic character for both compounds.

KEYWORDS: High-pressure chemistry, diazenides, nitrogen, nitrides, sodium azide, azides, pernitride, nitrogen fixation

Introduction

Homonuclear dinitrogen anions are common intermediates in biological and organo-metallic synthetic chemistry, and play an important role in the processes of nitrogen reduction to ammonia.^{1–3} In extended solid state compounds, nitrogen typically present in a form of a nitride anion N^{3-} and does not form catenated polyanions (with the exception of azides). The first alkaline-earth diazenides SrN , SrN_2 and BaN_2 containing $[\text{N}_2]^{2-}$ anions were synthesized from elements at high-pressure conditions only in the early 2000s.^{4,5} Later Schnick *et al.* used controlled decomposition of alkali and alkaline-earth metal azides at 3–12 GPa to obtain diazenides Li_2N_2 , CaN_2 , SrN_2 and BaN_2 .^{6,7} Diamond anvil cell techniques allowed the synthesis of a series of transition-metal compounds MN_2 ($\text{M} = \text{Ti}, \text{Cr}, \text{Fe}, \text{Co}, \text{Ni}, \text{Cu}, \text{Ru}, \text{Rh}, \text{Pd}, \text{Re}, \text{Os}, \text{Ir}, \text{Pt}$) at pressures 30–70 GPa.^{8–17} Further pressure increase leads to extended nitrogen catenation and formation of various polynitrogen species such as polytetrazenes $[-\text{N}-\text{N}=\text{N}-]_n$ in FeN_4 and Hf_2N_{11} ,^{18–20} polyacetylene-like chains in MgN_4 ,²¹ $\text{Hf}_4\text{N}_{20}\cdot\text{N}_2$, $\text{WN}_8\cdot\text{N}_2$, $\text{Os}_5\text{N}_{28}\cdot 3\text{N}_2$,²⁰ $\text{ReN}_8\cdot\text{N}_2$,²² pentazolate *cyclo*- N_5^- in CsN_5 ²³ and LiN_5 .²⁴

In MN_2 compounds metals usually possess their common oxidation states, while dinitrogen anion formally accommodates from 1 to 4 electrons. The degree of the charge transfer from metal to nitrogen significantly affects the properties of materials. For example, pernitrides of Pt, Ir, Os, Ti with $[\text{N}_2]^{4-}$ units and metals in the oxidation state IV are much more incompressible than $\text{M}^{\text{III}}\text{N}_2$ ($\text{M} = \text{Cr}, \text{Fe}, \text{Co}, \text{Ni}, \text{Ru}, \text{Rh}$) with $[\text{N}_2]^{3-}$, diazenides $\text{M}^{\text{II}}\text{N}_2$ ($\text{M} = \text{Ca}, \text{Ba}, \text{Sr}$) with $[\text{N}_2]^{2-}$ and $\text{M}^{\text{I}}\text{N}_2$ ($\text{M} = \text{Li}, \text{Cu}$) with $[\text{N}_2]^-$.

Despite a very wide range of metals with extremely different properties the stoichiometry of their compounds containing exclusively $[\text{N}_2]^{x-}$ anions is always MN_2 with the exception of recently-synthesized lithium diazenide Li_2N_2 .⁷ Here we synthesize two novel compounds in the Na-N system: NaN_2 at ~4 GPa and a compound with an unusual composition

$\text{Na}_3(\text{N}_2)_4$ at ~28 GPa containing $[\text{N}_2]$ units with non-integer formal charge of -0.75. $\text{Na}_3(\text{N}_2)_4$ opens a new class of $[\text{N}_2]$ – containing compounds with variable stoichiometry.

The most common high-pressure route to nitrogen-rich compounds is a direct reaction between metal and nitrogen in a laser-heated diamond anvil cell. However, this method has disadvantages such as inherent inhomogeneity of the reaction mixture with varying metal: nitrogen ratio across the sample. The use of azides as precursors offers a solution to this problem. Furthermore, azides provide a nitrogen-rich environment with $\text{N}/\text{M} \geq 3$, and this nitrogen is already activated, *i.e.* the activation barrier for the reaction is usually lower than in the reaction of metal with the triply bound N_2 molecule.

Sodium azide NaN_3 was an object of many high-pressure studies primarily due to its application as a high energy density material. Structural studies of NaN_3 at high pressures are challenging since the compound loses its crystallinity on compression. There are at least three structurally characterized phases of sodium azide: α - NaN_3 ($C2/m$), β - NaN_3 ($R-3m$) and γ - NaN_3 ($I4/mcm$).²⁵ Two high-pressure phases above 18 and 29 GPa were observed by Zhu *et al.* using XRD,²⁶ but the structures were not determined. High-pressure Raman spectroscopy studies by Eremets *et al.*²⁷ suggested the transformation of azide anions to larger nitrogen clusters in compressed NaN_3 , however structural characterization of these transitions is missing. Based on *ab initio* calculations Zhang *et al.* suggested that azide anions in compressed NaN_3 would condense to hexazine N_6 rings above 58 GPa.²⁸ Peiris *et al.*²⁹ and Holtgrewe *et al.*³⁰ demonstrated that photolysis of NaN_3 leads to the formation of non-crystalline products. Here we established an alternative route for synthesis of Na-N compounds from sodium azide by infrared laser heating at high pressure in diamond anvil cells.

Experimental section

Samples preparation

The high-pressure high-temperature behavior of NaN_3 (99.5%, Sigma-Aldrich) was studied on three samples. For single-crystal XRD studies (samples #1 and #2), a powder of sodium azide NaN_3 was placed in a sample chamber of a BX90 diamond anvil cell equipped with Boehler-Almax type diamonds.^{31,32} Re foil preindented to a thickness of 30 μm served as a gasket. A ruby chip was placed inside the sample chamber for pressure measurement. Sample #3 in a symmetric diamond anvil cell and with standard-cut diamonds was used for additional Raman measurements. The samples were compressed up to 26 (Sample #1), 4 GPa (Sample #2) and 25 GPa (Sample #3) and laser-heated ($\lambda = 1064$ nm) using double-sided laser-heating systems of the beamlines P02.2 (Petra III, DESY, Hamburg, Germany)³³, HPCAT (APS, Argonne, USA) and GSECARS (APS, Argonne, USA) respectively. No pressure-transmitting medium was used in all experiments. For samples #1 and #2, it was not possible to get the precise temperature of laser-heating, because the thermal radiation emitted by the sample was very weak. From the brightness of the heating spot, the temperature can be estimated to be below 1000 K. The sample #3 was heated up to $\sim 1900(200)$ K as determined by the black body radiation fit.

X-ray diffraction

XRD measurements of the sample #1 were performed at the beamline P02.2 of Petra III (DESY, Hamburg, Germany) with the X-ray beam ($\lambda = 0.2891$ Å) focused down to $1.8 \times 2 \mu\text{m}^2$ by a Kirkpatrick-Baez mirror system and diffraction patterns were collected on a PerkinElmer XRD 1621 flat-panel detector. XRD measurements of the sample #2 were performed at the beamline 16ID-B (APS, Argonne, USA) with the X-ray beam ($\lambda = 0.4066$ Å) focused down to $5 \times 5 \mu\text{m}^2$ by a Kirkpatrick-Baez mirror system and diffraction patterns were collected on a Pilatus 1M detector. XRD measurements of the sample #3 were performed at the beamline 13IDD (GSECARS, APS, Argonne, USA) with the X-ray beam ($\lambda = 0.2952$ Å) focused down to $3 \times 3 \mu\text{m}^2$ by a Kirkpatrick-Baez mirror system and diffraction patterns were collected on a Pilatus 1M detector (CdTe). For the single-crystal XRD measurements samples were rotated around a vertical ω -axis in a range $\pm 35^\circ$. The diffraction images were collected with an angular step $\Delta\omega = 0.5^\circ$ and an exposure time of 1s or 2s/frame. For the analysis of the single-crystal diffraction data (indexing, data integration, frame scaling and absorption correction) we used the *CrysAlisPro* software package. Specific details of the multi-grain dataset analysis are given in the Supporting Information and in the Figure S1. To calibrate an instrumental model in the *CrysAlisPro* software, i.e., the sample-to-detector distance, detector's origin, offsets of goniometer angles, and rotation of both X-ray beam and the detector around the instrument axis, we used a single crystal of orthoenstatite ($(\text{Mg}_{1.93}\text{Fe}_{0.06})(\text{Si}_{1.93}\text{Al}_{0.06}\text{O}_6)$, *Pbca* space group, $a = 8.8117(2)$, $b = 5.18320(10)$, and $c = 18.2391(3)$ Å). Using the Olex2 crystallography software package, the structures were solved with the ShelXT structure solution program³⁴ using Intrinsic Phasing and refined with the ShelXL³⁵ refinement package using Least Squares minimization. The powder diffraction images were integrated to powder patterns with Dioptas software.³⁶ Le-Bail fits of the diffraction patterns were performed with the Jana2006 software.³⁷ CSD-1999694 and CSD-1999711 contain the supplementary crystallographic data for this paper. These data can be obtained free of charge from FIZ Karlsruhe via www.ccdc.cam.ac.uk/structures

Raman spectroscopy

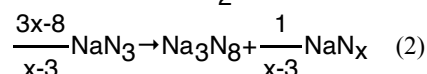
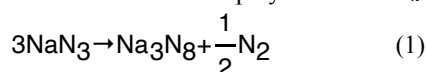
Raman spectra were measured using the GSECARS Raman system with the excitation wavelength of 532 nm. The full description of the Raman system has been published elsewhere.³⁸

Theoretical calculations

The minimum enthalpy structures were recalculated with the norm conserving part of the PAW.³⁹ Energy cut-off of 750-830 eV using the density functional theory (DFT) code CASTEP⁴⁰ and exchange-correlation functional in Perdew-Burke-Ernzerhof (PBE) approximation.⁴¹ Thirty irreducible k-points were used for electronic Brillouin zone integration and Tkachenko-Scheffler dispersion correction was enabled.⁴² The Brillouin zone integrals were performed using Monkhorst-Pack grids⁴³ with spacings between grid points of less than 0.01 Å^{-1} . The structures were considered converged when the force on each atom was less than 0.002 eV/Å and total energy tolerance was better than 10^{-7} eV . Deviation of the stress tensor from that defined by the target pressure was less than 0.001 GPa. The phonon dispersion and phonon frequencies calculations were performed using a finite displacements method implemented in CASTEP code.⁴⁴ Finite displacement phonons are calculated only at the commensurate q-points at the cost of creating a supercell. The population analysis was carried out as implemented in CASTEP.⁴⁵

Results and discussion

The laser heating of NaN_3 at 26 GPa leads to an irreversible chemical reaction that is evidenced by the changes in the optical properties of the sample (heated area becomes opaque, see inset in Figure 1a). After the heating, the pressure increased up to ~ 28 GPa. The reaction product has a rich spotty diffraction pattern that allowed using single-crystal XRD analysis (Figure 1b, Table S1). The methodology of such multigrain analysis is described in a number of our recent publications (e.g., Ref. ¹⁵). The diffraction pattern could be indexed with the tetragonal body-centered unit cell with $a = 4.9597(16)$ and $c = 16.29(7)$ Å. Successful structure solution was achieved in a space group *I4₁/amd* (No. 141) and revealed the composition of the new compound as Na_3N_8 . Full crystallographic information for this compound is provided in the Table S1 and in the supporting *cif* file. The expected decomposition reaction of NaN_3 should result in either the release of molecular nitrogen, or in the formation of polynitrides NaN_x with $x > 3$:



Experimental information on compounds NaN_x with $x > 3$ is limited to a Na-pentazolate framework MPF-1, which is not expected to form at high pressure due to its open zeolitic architecture.⁴⁶ However, recently Steele and Oleynik predicted that sodium pentazolate NaN_5 would be among thermodynamically stable phases in the Na-N system at elevated pressures.⁴⁷ In order to get an insight into the mechanism of NaN_3 decomposition, we have performed Raman spectroscopy measurements (Figures S2-S4). Although we could identify peaks corresponding to the nitrogen stretching vibration ($\nu_1 = 2376 \text{ cm}^{-1}$ at ~ 12 GPa, Figure S2) there is an indication that some other nitrogen-containing phases may be present in the sample chamber (Figure S4). Therefore, both reactions 1 and 2 may take place. It is likely that nitrogen and products NaN_x might not be well crystallized, to produce strong diffraction peaks, which would allow their unambiguous XRD identification in a mixture with Na_3N_8 . The reaction (1) is also supported by our theoretical calculations, suggesting that it is exothermic with $\Delta H = -0.261 \text{ eV}$ per NaN_3 unit.

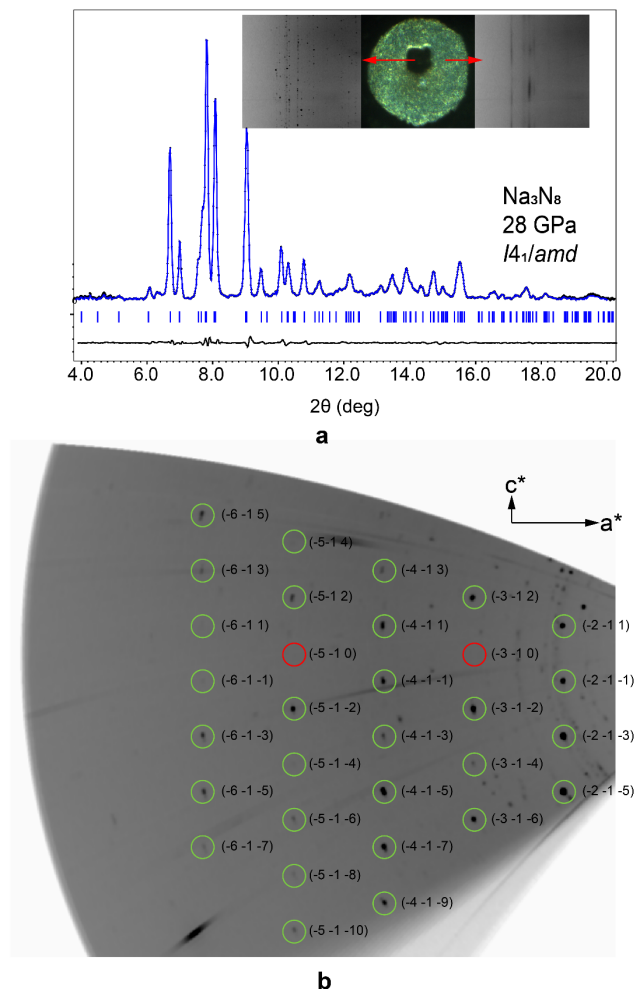


Figure 1. (a) Powder diffraction pattern of Na_3N_8 at ~ 28 GPa, $\lambda = 0.2891$ Å. Inset shows and optical image of the heated sample area at ~ 28 GPa and cake slices of the diffraction images of non-heated and heated sample areas (b) Reconstructed ($h-1l$) precession image from the single-crystal XRD dataset at 28 GPa. Reflections hkl with $h+k+l = 2n+1$ are absent due to the I -centering of the lattice. Reflections $hk0$ with $h = 2n+1$ or $k = 2n+1$ (marked by red circles) are absent due to the glide plane symmetry operations of the space group $I4_1/amd$.

Na_3N_8 has an unprecedented structure type. However, two symmetry-independent sodium atoms Na1 and Na2 occupying Wyckoff sites $4a$ and $8e$ respectively form a substructure isostructural to $\alpha\text{-ThSi}_2$ (Figure 2a). Each Na2 atom has three close Na2 neighbors with $d(\text{Na-Na}) = 2.81$ and 2.78 Å at 28 GPa. These distances are close to those in $bcc\text{-Na}$ at similar pressures (2.79 Å).⁴⁸ With Na2-Na2-Na2 angles close to 120° , Na2 atoms form one of the basic 3-connected three-dimensional nets (*ths*) described by a vertex symbol $10_210_410_4$.⁴⁹ Two nitrogen atoms N1 and N2 occupy Wyckoff sites $16h$ and $16f$ respectively and form N1-N1 and N2-N2 dinitrogen dumbbells with $d(\text{N1-N1}) = 1.147(3)$ Å and $d(\text{N2-N2}) = 1.149(3)$ Å at 28 GPa. N1-N1 units are surrounded by seven sodium atoms forming a distorted pentagonal bipyramid as shown in the Figures 2b-2d, while N2-N2 units are surrounded by six sodium atoms that form a distorted octahedron. Na1 atoms are coordinated by eight N_2 units in a side-on manner, while Na2 – by eight end-on N_2 and one side-on N_2 units. Side-on coordination usually leads to a greater charge transfer⁵⁰, which is in a good agreement with our calculations of Mulliken atomic charges (Table S2).

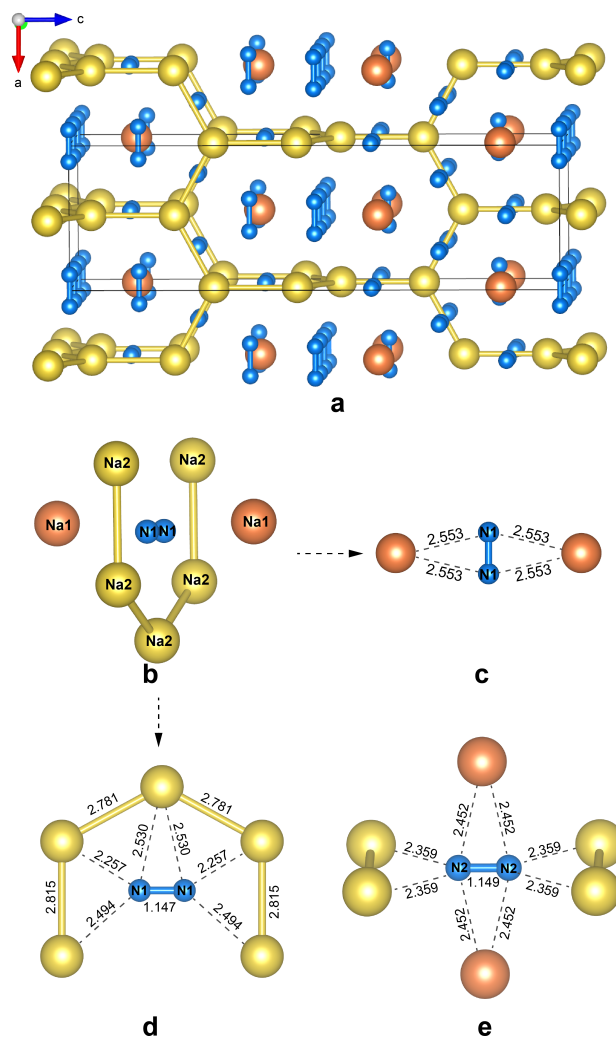


Figure 2. (a) Crystal structure of Na_3N_8 at 28 GPa. Shortest Na-Na contacts are indicated by yellow sticks (b)-(e) Coordination environments of nitrogen molecules.

To study the stability region of Na_3N_8 the sample was decompressed in a few steps to ambient pressure. Na_3N_8 remains in the sample chamber down to 7.7 GPa. Lattice parameters increase with decreasing pressure with c/a ratio approaching, but not reaching, the ideal value for the *ths* net of $2\sqrt{3}$. One can notice a slight kink in the pressure-dependence of the unit cell parameters a and c at ~ 15 GPa (Figure S5). We should note here that while the experiment was performed under very non-hydrostatic conditions the evolution of deviatoric stresses may influence the behavior of the lattice parameters on compression or decompression. Furthermore, the certain systematic error in pressure determination may occur due to the pressure gradient between the pressure marker (ruby) and the heated spot. However, we should also mention that the refined N1-N1 and N2-N2 distances that are similar above 15 GPa, begin to deviate at lower pressures (Figure S5). The N2-N2 distance unexpectedly decreases with decreasing pressure and tends to the N-N distance in triply bound dinitrogen molecule. This may indicate the onset of charge redistribution between two dinitrogen units and an onset of further decomposition.

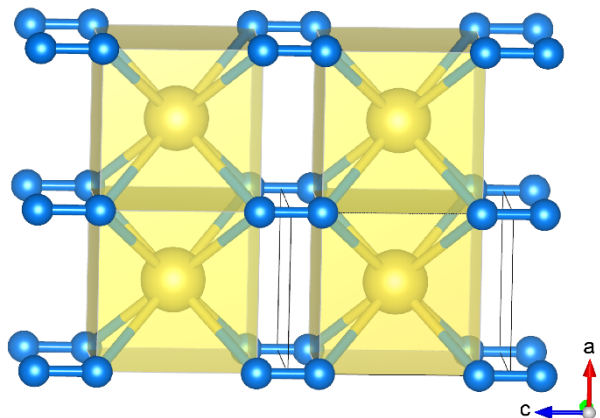


Figure 3. Crystal structure of NaN_2 at 4 GPa. Nitrogen and sodium atoms shown as blue and yellow spheres, respectively. NaN_8 coordination polyhedra are highlighted. $d(\text{N-N}) = 1.161(9) \text{ \AA}$, $d(\text{Na-N}) = 2.582(3) \text{ \AA}$.

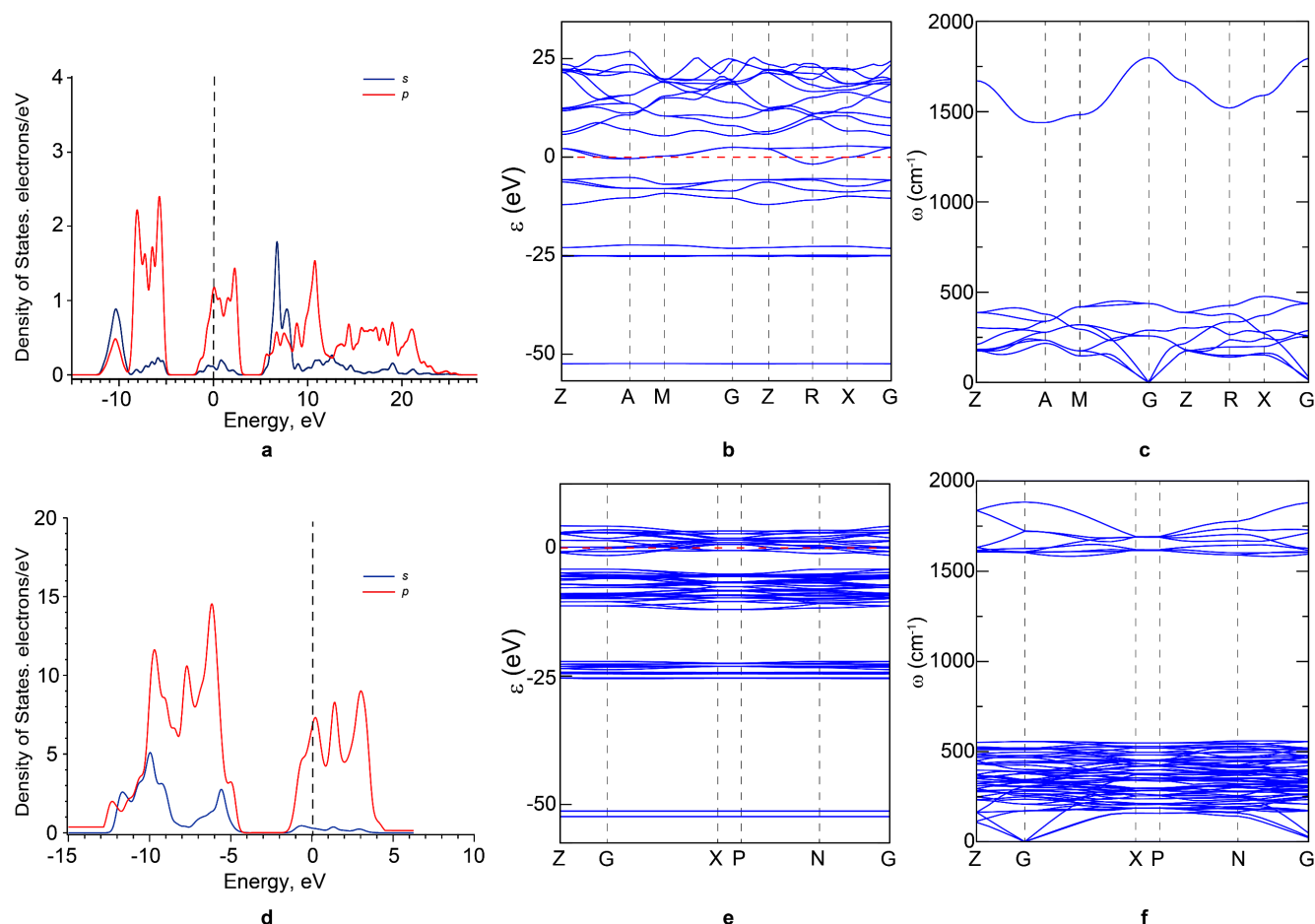
Further decompression to 4 GPa resulted in the disappearance of all single-crystalline spots in the diffraction pattern and in the formation of a powder-like sample. Primitive tetragonal unit cell with $a = 3.0009(13)$ and $c = 4.101(3) \text{ \AA}$ describes the positions of the diffraction peaks with a good agreement (Figure S6). To recrystallize this compound, the sample was slightly heated by the infrared laser ($\lambda = 1064 \text{ nm}$), which allowed the analysis of the crystal structure by means of single-crystal X-ray diffraction. Structure solution and refinement revealed the composition of this compound as NaN_2 (Table S1). NaN_2 is isostructural to $\alpha\text{-FeSi}_2$ and has a space group $P4/mmm$ (No. 123). It can be described as a layer packing structure, in which the layers are stacked along $[001]$. The layers consisting of face-sharing NaN_8 slightly distorted cubes are interconnected through N-N bonds with $d(\text{NN}) = 1.161(9) \text{ \AA}$ at 4 GPa (Figure 3). Powder-like NaN_2 was also obtained in a separate experiment by heating sodium azide compressed at 4.5 GPa (Figure S6). On decompression NaN_2 is stable down to at least 2.8 GPa but decomposes with the formation of $bcc\text{-Na}$ at ambient pressure (Figure S6).

Both Na_3N_8 and NaN_2 have very unusual electron counts. The chemical formula of Na_3N_8 can be rewritten as $\text{Na}_3(\text{N}_2)_4$ and if we assume that Na is in its standard oxidation state +I, the charge on each dinitrogen unit would be -0.75 . Obviously, 0.75 electrons cannot be localized on the nitrogen antibonding π^* orbitals and these electrons must be either delocalized, or the compound may be an electride $\text{Na}^+(\text{N}_2)_4 \cdot 3e^-$, with electrons serving as anions like in some subnitrides (e.g. $[\text{Ca}_2\text{N}]$

$+e^-$).⁵¹ A more unlikely scenario is that Na_3N_8 is a van der Waals compound without a charge transfer from sodium to nitrogen atoms. Similar considerations are valid for NaN_2 too. The first clue on the bonding situation comes from the analysis of interatomic nitrogen-nitrogen distances. Assuming the complete charge transfer from sodium to nitrogen atoms, the N-N bond orders in Na_3N_8 and NaN_2 should be 2.625 and 2.5, respectively. Therefore, in this case N-N distances should be only slightly longer than the triple $\text{N}\equiv\text{N}$ bond.

Indeed, the interatomic nitrogen-nitrogen distances in Na_3N_8 and in NaN_2 (1.147 and 1.161 \AA) are in a very good agreement with the formal N-N bond orders and clearly follow the trend $d_{\text{NN}}(\text{N}\equiv\text{N}) < d_{\text{NN}}(\text{Na}_3\text{N}_8) < d_{\text{NN}}(\text{NaN}_2) < d_{\text{NN}}(\text{diazonides}) < d_{\text{NN}}(\text{pernitrides})$. We note here that N-N dimers are very incompressible. For instance the N-N single bond length variation in ReN_2 between 0 and 42 GPa is about 0.05 \AA ¹⁵, while higher-order N-N bonds are expected to be even more incompressible. Therefore, for qualitative purposes, it is justified to compare N-N distances in compounds at slightly different pressures. The elongation of N-N distances in Na_3N_8 and in NaN_2 compared to $\text{N}\equiv\text{N}$ suggests that charge transfer occurs in both compounds and the electrons are delocalized between nitrogen p -states, implying the metallic character of both compounds, which agrees with the observed black metallic color of the samples. In order to get a deeper insight into the bonding nature, we have performed electronic structure calculations that confirmed that both compounds are metallic (Figure 4) and the main contributions of bands at the Fermi level comes from the N $2p$ states.

Hitherto, several uncommon diazenides such as $(\text{Sr}^{2+})_8(\text{N}^{3-})_4([\text{N}_2]^{2-}) \cdot 2e^-$ or $(\text{Li}^+)_2(\text{Ca}^{2+})_3([\text{N}_2]^{2-})_3 \cdot 3e^-$ were synthesized at high-pressure conditions.^{52,53} These compounds contain diazenide anions $[\text{N}_2]^{2-}$ and the delocalized electrons responsible for their metallic properties. By analogy with subnitrides such as $(\text{Ba}^{2+})_3(\text{N}^{3-}) \cdot 3e^-$, these compounds were called subdiazonides. On the contrary, both NaN_2 and Na_3N_8 may be considered as undercharged diazenides. Dinitrides with undercharged diazenide anions $[\text{N}_2]^-$ were recently reported in Li-N and Cu-N systems. Laniel *et al.* synthesized LiN_2 in a reaction between Li_3N and nitrogen at 10.5 GPa, while CuN_2 was synthesized from elements at $\sim 50 \text{ GPa}$.¹⁰ Both compounds have hexagonal $\text{NiAs } P6_3/mmc$ structure type with N-N distances of $\sim 1.2 \text{ \AA}$. Undercharged $[\text{N}_2]^-$ diazenide groups are also present in Hf_2N_{11} compound synthesized at 1 Mbar.²⁰



An unprecedented example of Na_3N_8 demonstrates that $[\text{N}_2]$ units with non-integer formal charge exist. The $[\text{N}_2]$ unit may therefore serve as a versatile electron acceptor in solid-state chemistry and may exist not only in compounds with simple AB_2 or A_2B_2 stoichiometries.

Na-N system.⁴⁷ Our calculation of the convex hull diagram at 28 GPa shows that NaN_2 with $Cmmm$ symmetry emerges as the thermodynamically stable phase (in agreement with Ref. ⁴⁷), while Na_3N_8 is slightly metastable (but is on the hull within the accuracy of the calculations). The calculations of Steele and Oleynik considered Na_xN_y structures having up to 16 atoms in the unit cell and this could explain why Na_3N_8 , containing 44 atoms in the unit cell (22 atoms in the primitive cell) was not predicted. The existence of Na_3N_8 may have a consequence on the calculated stability regions of predicted NaN_5 , and Na_2N_5 compounds.⁴⁷

formal charges including non-integer ones. This opens a new class of solid-state compounds containing homonuclear dinitrogen anions and having various compositions. It is likely that compounds with M_3N_8 stoichiometry may be found for group 2 metals with similar ionic radii as Na^+ (especially Ca and Sr). Replacing Na by the alkaline-earth metal in Na_3N_8 would lead to compounds $M^{II}_3N_8$ with $[N_2]^{1.5-}$.

Azides and NaN_3 in particular are often used in a synthesis of nitrides,^{56–60} where they are commonly considered as solid sources of nitrogen, taking in account their ambient-pressure decomposition route ($2NaN_3 \rightarrow 2Na + 3N_2$). Our study shows that sodium azide can be already decomposed at 4 GPa, which results in the formation of activated $[N_2]^-$ species together with molecular N_2 and this mechanism may play a crucial role for the high-pressure solid-state metathesis synthetic approach that recently became widespread for the synthesis of nitrides.^{61–63} The formation of stable Na_3N_8 compound with a very unexpected stoichiometry is an important indication that theoretical calculations, which became widespread in the high-pressure community^{64,65} should always include variable-stoichiometry search and should not limit the unit cell contents to a few atoms for the sake of the computation time. Therefore, these experiments revealing unexpected compositions are an important contribution for the adjustments of computational strategies.

ASSOCIATED CONTENT

Supporting Information

Crystallographic information files for Na_3N_8 and NaN_2 . Details of structure refinement, powder diffraction patterns, Raman spectra. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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Author Contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

Funding Sources

Research was sponsored by the Army Research Office and was accomplished under the Cooperative Agreement Number W911NF-19-2-0172.

ACKNOWLEDGMENT

Parts of this research were carried out at the Extreme Conditions Beamline (P02.2) at DESY, a member of Helmholtz Association (HGF). Portions of this work were performed at HPCAT (Sector 16), Advanced Photon Source (APS), Argonne National Laboratory. HPCAT operations are supported by DOE-NNSA's Office of Experimental Sciences. Raman spectroscopy experiments were performed at GeoSoilEnviroCARS (The University of Chicago, Sector 13), Advanced Photon Source (APS), Argonne National Laboratory. GeoSoilEnviroCARS is supported by the National Science Foundation - Earth Sciences (EAR - 1634415) and Department of Energy- GeoSciences (DE-FG02-94ER14466). The Advanced Photon Source is a U.S. Department of Energy (DOE) Office of Science User Facility operated for the DOE Office of Science by Argonne National Laboratory under Contract No. DE-AC02-06CH11357.

ABBREVIATIONS

XRD, X-ray diffraction; DAC, diamond anvil cell

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SYNOPSIS

Here using a controlled decomposition of sodium azide NaN_3 at high pressure conditions two novel compounds $\text{Na}_3(\text{N}_2)_4$ and NaN_2 were synthesized. Both contain dinitrogen anions. NaN_2 might be the common intermediate in high-pressure solid-state metathesis reactions where NaN_3 is used as a source of nitrogen. $\text{Na}_3(\text{N}_2)_4$ opens a new class of compounds, where $[\text{N}_2]$ units accommodate a non-integer formal charge.

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