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Allanite at high pressure: effect of REE on the elastic behaviour of epidote-group minerals

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Abstract

The compressional behaviour of a natural allanite from Lago della Vecchia (upper Cervo valley, Italy) metagranitoids

$[A^I(Ca_{0.69}Fe^{2+}_{0.31})_{\Sigma 1.00}A^{II}(Ca_{0.46}Ce_{0.24}La_{0.12}Sm_{0.02}Pr_{0.05}Nd_{0.09}Th_{0.02})_{\Sigma 1.00}$

$M^I(Al_{0.65}Fe^{3+}_{0.34}Ti_{0.02})_{\Sigma 1.01}M^2(Al_{0.99})M^3(Fe^{3+}_{0.54}Fe^{2+}_{0.36}Mg_{0.06}Ti^{4+}_{0.02}Al_{0.01})_{\Sigma 0.99}$

$Si^1, Si^2, Si^3(Si_{2.80}Al_{0.20})_{\Sigma 3.00}O_{11}(OH, O)]$ has been investigated up to 16 GPa (at 298 K) by

means of *in-situ* synchrotron single-crystal X-ray diffraction. Experiments have been

conducted under hydrostatic conditions, using a diamond anvil cell and the mix

methanol:ethanol:water=16:3:1 (up to 10 GPa) and neon (up to 16 GPa) as pressure-

transmitting media. No phase transition has been observed within the pressure-range

investigated; data collected in decompression prove that, at least up to 16 GPa (at 298

K), the deformation mechanisms are fully reversible. A third-order Birch-Murnaghan

Equation of State (BM-EoS) was fitted to the *P-V* data (up to 10 GPa), giving: $V_0 =$

$470.2(2) \text{ \AA}^3$, $K_{P_0, T_0} = 131(4) \text{ GPa}$ and $K' = 1.9(8)$. The evolution of the lattice parameters

with pressure shows a slight anisotropic compression pattern, with

$K_{P_0, T_0}(a):K_{P_0, T_0}(b):K_{P_0, T_0}(c) = 1.24:1.52:1$. The monoclinic β -angle decreases

monotonically with pressure, with: $\beta_P(^{\circ}) = \beta_{P_0} - 0.0902(4)P$ ($R^2 = 0.997$, with P in

GPa).

The main deformation mechanisms at the atomic scale are described based on a series

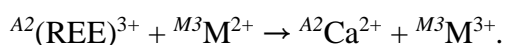
of structure refinements at different pressures. A comparison between the

compressional behavior of allanite, epidote and clinozoisite is carried out.

Keywords: allanite, epidote, synchrotron single-crystal X-ray diffraction, high-pressure, compressibility.

70 Introduction

71 Allanite is a sorosilicate and a member of the epidote group, with general
72 crystal formula $A(1)A(2)M(1)M(2)M(3)(\text{SiO}_4)(\text{Si}_2\text{O}_7)\text{O}(\text{OH})$, where the $A(1)$ and $A(2)$
73 are sites with coordination number $\text{CN}>6$ and mainly occupied by Ca, and $M(1)$, $M(2)$
74 and $M(3)$ are octahedral sites ($\text{CN}=6$) mainly occupied by Al and Fe^{3+} (Dollase 1971;
75 Franz and Liebscher 2004; Armbruster et al. 2006). All the members of the epidote
76 group are monoclinic in symmetry, with structure topology consistent with the space
77 group $P2_1/m$, although possible symmetry reduction in some epidotes (to Pm , $P2_1$ or
78 $P\bar{1}$) has been suggested as effect of cation ordering (Franz and Liebscher 2004). The
79 structure of the epidote group minerals has single silicate tetrahedra (SiO_4), double
80 silicate tetrahedra (Si_2O_7), and continuous chains of MO_6 and $\text{MO}_4(\text{OH})_2$ octahedra
81 (parallel to the b -axis) as main building-block units. The octahedra are bridged by
82 single SiO_4 and double Si_2O_7 tetrahedral groups, in a configuration as that shown in
83 (Fig. 1). Clinozoisite (ideally $^{A1,A2}\text{Ca}_2^{M1,M2,M3}\text{Al}_3(\text{SiO}_4)(\text{Si}_2\text{O}_7)\text{O}(\text{OH})$) can be
84 considered as the reference structure of the epidote group minerals, in which the three
85 independent octahedral M sites ($M1$, $M2$ and $M3$) are fully occupied by Al and the
86 two independent A sites ($A1$ and $A2$) are occupied by Ca. The complex crystal-
87 chemistry of the epidote group led the Commission of the International Mineralogical
88 Association to divide it into three subgroups (Armbruster et al. 2006). The allanite
89 subgroup contains rare-earth elements (REE) rich minerals, typified by the
90 eponymous mineral “allanite”. The crystal chemistry of the allanite subgroup
91 members may be derived from that of clinozoisite, by homovalent substitutions and
92 one coupled heterovalent substitution, as follows:



94 In this light, the (cationic) site population is represented by: $A1 = M^{2+}$, $A2 = M^{3+}$, $M1$
95 $= M^{3+}$, $M2 = M^{3+}$, and $M3 = M^{2+}$, and the general formula of allanite is:

96 $^{A1}(\text{Ca})^{A2}(\text{REE}^{3+})^{M1,M2}(\text{Al})_2^{M3}(\text{Fe}^{2+})(\text{SiO}_4)(\text{Si}_2\text{O}_7)\text{O}(\text{OH})$ (Dollase 1971; Gieré and
97 Sorensen 2004; Armbruster et al. 2006).

98 Epidotes mainly occur in low-grade metamorphic rocks (250–400°C, 1-2
99 kbars). However, Poli and Schmidt (1998) showed that epidotes are stable over a wide
100 range of pressure and temperature in continental and oceanic crust. Magmatic
101 epidotes were also reported and described by Schmidt and Poli (2004). A series of
102 studies showed how the stability of epidote group minerals is not only influenced by

103 pressure and temperature, but also by the Al/Fe³⁺ ratio, oxygen fugacity, fluid
104 composition and solution pH (*e.g.*, Holdaway 1972; Liou 1973; Bird and Helgeson
105 1980; Bird et al. 1988; Klemd 2004). In addition, a series of *in-situ* high-pressure
106 (HP) and high-temperature (HT) experiments have been devoted to epidote group
107 minerals, and in particular to clinozoisite and epidote *sensu stricto*, in order to derive
108 volume and axial compressibility or thermal expansion (*e.g.*, Catti et al. 1988;
109 Holland et al. 1996; Pawley et al. 1996; Comodi and Zanazzi 1997; Franz and
110 Liebscher 2004; Liebscher 2004; Gatta et al. 2010, 2011a, 2011b; Qin et al. 2016),
111 along with *P*- or *T*-induced deformation mechanisms at the atomic scale (*e.g.*, Comodi
112 and Zanazzi 1997; Gatta et al. 2010, 2011a). However, to the best of our knowledge,
113 no experiments have so far been devoted to the behaviour of allanite at non-ambient
114 conditions by *in-situ* experiments. Consequently, bulk and axial compressibilities or
115 thermal expansion coefficients of this mineral are completely unknown, and the role
116 played by the REE replacing Ca at the *A* sites, or the occurrence of Fe²⁺ at the
117 octahedral *M* sites, is still obscure. In this light, the aim of this study is the
118 investigation of the HP-behavior of a natural allanite, from Lago della Vecchia (upper
119 Cervo valley, Italy), by *in-situ* synchrotron single-crystal diffraction with a diamond
120 anvil cell, in order to: provide reliable thermodynamic parameters for petrologic
121 modeling, report any potential *P*-induced phase transition and describe the main
122 deformation mechanisms at the atomic scale *via* single-crystal structure refinements.
123 A comparison between the compressional parameters of allanite and those of epidote
124 *sensu stricto* and clinozoisite is carried out.

125

126

127

128 **Materials and experimental methods**

129 Natural single-crystals of allanite from Lago della Vecchia (upper Cervo valley, Italy)
130 metagranitoids were used in this study. The host rock is characterized by
131 heterogenous deformation due to stain partitioning (Corti et al. 2017) during
132 development of *HP-LT* blueschist-facies dominant fabric, which represent a re-
133 equilibration following the metamorphic peak in the eclogite facies, under thermally
134 depressed conditions (Corti et al. 2018). Chemical microanalyses in wavelength-
135 dispersive mode (EPMA-WDS) were performed on a series of optically homogeneous
136 sub-millimetric crystals, using a JEOL JXA-8200 microprobe at the Earth Sciences
137 Department, University of Milano. The system was operated using an accelerating
138 voltage of 15 kV, a beam current of 5 nA, a beam diameter of 5 μm , and a counting
139 time of 30 s on the peaks and 10 s on the backgrounds. A series of natural and
140 synthetic standards were used. The raw data were corrected for matrix effects using
141 the protocol implemented in the JEOL suite of programs. The crystals of allanite,
142 selected for this study, were found to be compositionally homogeneous. The average
143 unit-formula, based on more than 40 point-analyses and calculated following the
144 protocol recommended by IMA Commission (Armbruster et al. 2006), on the basis of
145 13 oxygen atoms, is:
146 $A1(\text{Ca}_{0.69}\text{Fe}^{2+}_{0.31})_{\Sigma 1.00}A2(\text{Ca}_{0.46}\text{Ce}_{0.24}\text{La}_{0.12}\text{Sm}_{0.02}\text{Pr}_{0.05}\text{Nd}_{0.09}\text{Th}_{0.02})_{\Sigma 1.00}$
147 $M1(\text{Al}_{0.65}\text{Fe}^{3+}_{0.34}\text{Ti}_{0.02})_{\Sigma 1.01}M2(\text{Al}_{0.99})M3(\text{Fe}^{3+}_{0.54}\text{Fe}^{2+}_{0.36}\text{Mg}_{0.06}\text{Ti}^{4+}_{0.02}\text{Al}_{0.01})_{\Sigma 0.99}$
148 $Si1, Si2, Si3(\text{Si}_{2.80}\text{Al}_{0.20})_{\Sigma 3.00}\text{O}_{11}(\text{OH}, \text{O})$. Further details pertaining to experimental
149 protocols and EPMA-WDS data statistics are given in Corti et al. (2019).

150 High-pressure synchrotron X-ray single-crystal diffraction experiments were
151 performed at the Extreme Conditions Beamline P02.2 at DESY/PETRA-III. X-rays
152 with an energy of 42.7 keV (0.2904 \AA wavelength) were used, with a focusing spot of
153 ~ 8.5 (H) x 1.8 (V) μm^2 originating from a compound refractive lenses (CRL) system
154 consisting of 110 Be lenses with a radius of 50 μm (400 μm beam acceptance) and a
155 focal length of 1,221 mm. Two prismatic single-crystals of allanite ($\sim 50 \times 50 \times 15$
156 μm^3) were selected for the *HP* experiments, loaded, respectively, in two symmetric
157 diamond anvil cells (DAC), equipped with Bohler–Almax design diamonds/seats
158 with a 70° opening and 300- μm culets size. For the first DAC, a 250- μm -thick
159 rhenium gasket was pre-indented to 50 μm and then drilled with 150 μm hole, in
160 which the crystal of allanite along with some calibrated ruby spheres (for pressure

161 determination, according to Mao et al. 1986) were placed. Neon was used as
162 hydrostatic pressure transmitting medium (Klotz et al. 2009). For the second DAC, a
163 250- μm -thick steel gasket was pre-indented to 60 μm and then drilled with 150 μm
164 hole, in which the crystal of allanite and ruby micro-spheres were located. In this case,
165 the methanol:ethanol:water = 16:3:1 mix was used as hydrostatic P -transmitting fluid
166 up to 10 GPa (Angel et al. 2007). For both the experiments, pressure was increased
167 with an automated pressure-driven system and measured with the online
168 ruby/alignment system. Diffraction images were acquired on a PerkinElmer XRD
169 1621 flat panel detector, using an in-house script for collecting step-scan diffraction
170 images. Sample to detector distance (402.34 mm) was calibrated using a CeO_2
171 standard (NIST 674a). The diffraction images were then converted to conform to the
172 “Esperanto” format of the program CrysAlis (Rigaku – Oxford Diffraction 2018;
173 Rothkirch et al. 2013). The diffraction data were first collected with the crystals in the
174 DAC and without any P -medium (*i.e.* ambient pressure). A pure ω -scan ($-33^\circ \leq \omega \leq$
175 $+33^\circ$), with a step size of 0.5° and an exposure time of 0.5 s/frame, was used during
176 data collection. X-ray diffraction peaks were then indexed and their intensities were
177 integrated and corrected for Lorentz-polarization (Lp) effects, using the CrysAlis
178 package. Scaling and correction for absorption (due to the DAC components) was
179 applied by the semi-empirical *ABSPACK* routine implemented in CrysAlis. The
180 reflection conditions were consistent with those of the space group $P2_1/m$. HP data for
181 the experiments with methanol:ethanol:water mix as P -fluid were collected up to ~ 8.5
182 GPa; those pertaining to the experiment with neon as P -transmitting medium up to
183 ~ 16.3 GPa (the hydrostatic limit of neon, according to Klotz et al. 2009). No evidence
184 of phase transitions was observed within the P -range investigated. Data collected in
185 decompression proved that allanite behaves elastically at least up to 16 GPa (at 298
186 K) (Table 1) under hydrostatic conditions. The unit-cell parameters of allanite with P ,
187 based on the two different ramps (*i.e.*, in methanol:ethanol:water mix and in neon) are
188 listed in Table 1.

189 The isotropic structure refinements, based on the intensity data of the two HP
190 ramps, were conducted using the software SHELXL97 (Sheldrick 1997, 2008),
191 starting from the structure model of Dollase (1971) and Bonazzi et al. (2009), in the
192 space group $P2_1/m$. To reduce the number of variables to refine, and considering the
193 average chemical composition of the allanite of this study, the atomic sites were

194 modelled as follows: the *A1* and *A2* sites were modelled with a mixed (Ca + Ce) X-
 195 ray scattering curve, and the fraction of Ca vs. Ce were refined; the *M1* and *M2*
 196 octahedral sites as populated by Al only and the *M3* site as populated by (Fe + Al),
 197 and the fraction Fe vs. Al were refined; the three independent tetrahedral sites (*i.e.*,
 198 *Si1*, *Si2* and *Si3*) were modelled as fully occupied by Si. For all the refinements,
 199 convergence was rapidly achieved and, at the end of the last cycles of refinement, no
 200 significant correlation was observed in the variance-covariance matrix of the refined
 201 parameters. The principal statistical parameters of the structure refinements are listed
 202 in Table 2. Atomic coordinates and site occupancies of selected structure refinements
 203 are given in Table S1. Bond distances and other relevant structural parameters are
 204 reported in Tables S2 and 3.

205

206 **Results: Elastic behaviour of allanite at high-pressure**

207 The evolution of the lattice parameters of allanite with pressure is shown in
 208 Fig. 2 and Fig. S1, which shows that no phase transition or change of the deformation
 209 mechanisms occur within the *P*-range investigated (*i.e.*, at least up to 16 GPa at 298
 210 K). A second- and a third-order Birch-Murnaghan Equation of State (BM-EoS) (Birch
 211 1947; Angel 2000) were fitted to the *P-V* data pertaining to the experiment with
 212 methanol:ethanol:water mix as *P*-transmitting fluid (*i.e.*, the most populated *P*-ramp),
 213 using the EOS-FIT program (by RJ Angel, www.rossangel.com). This isothermal EoS
 214 is based on the assumption that the high-pressure strain energy in a solid can be
 215 expressed as a Taylor series in the Eulerian finite strain, defined as $fe = [(V_0/V)^{2/3} -$
 216 $1]/2$, and allows to obtain the bulk modulus ($K_{P_0,T_0} = V(\partial P/\partial V)_{T_0} = \beta^{-1}_{P_0,T_0}$, where
 217 β_{P_0,T_0} is the volume compressibility coefficient at room conditions) and its *P*-
 218 derivatives. Expansion in the Eulerian strain polynomial has the following form:

$$219 \quad P(fe) = 3K_{P_0,T_0} fe (1 + 2fe)^{5/2} \{ 1 + 3/2(K' - 4)fe + 3/2[K_{P_0,T_0}K'' + (K' - 4)(K' - 3) + 35/9]$$

$$220 \quad fe^2 + \dots \}, \text{ where } K' = \partial K_{P_0,T_0} / \partial P \text{ and } K'' = \partial^2 K_{P_0,T_0} / \partial P^2.$$

221 The BM-EoS parameters, simultaneously refined using the data weighted by their
 222 uncertainties in *P* and *V*, are listed in Table 4. Using a second-order BM-EoS fit,
 223 convergence is achieved with: $V_0 = 470.6(2)\text{\AA}^3$ and $K_{P_0,T_0} = 122(1)$ GPa. A better fit is
 224 obtained using a third-order BM-EoS with: $V_0 = 470.2(2)\text{\AA}^3$, $K_{P_0,T_0} = 131(4)$ GPa and
 225 $K' = 1.9(8)$. The use of a third-order BM-EoS in energy, to model the compressional
 226 behaviour of allanite, is also corroborated by the evolution of the Eulerian finite strain

227 vs. “normalized stress” plot (fe - Fe plot, with $Fe = P/[3fe(1+2fe)^{5/2}]$; Angel 2000),
228 shown in Fig. 3: the weighted linear regression through the data points yields $Fe(0)=$
229 131(2) GPa as intercept values and the (negative) slope of the regression line gives
230 rise to a K' value of 1.9(6), in good agreement with the third-order BM-EoS fit.

231 The confidence ellipses at 68.3% level ($\Delta\chi^2 = 2.30, \pm 1\sigma$), 95.4% level ($\Delta\chi^2$
232 $= 6.17, \pm 2\sigma$) and 99.7% level ($\Delta\chi^2 = 11.8, \pm 3\sigma$) were calculated starting from the
233 variance-covariance matrix of K_{P_0, T_0} and K' obtained from the least-square procedure
234 (third-order BM-EoS fit previously described; Angel 2000). The ellipses are strongly
235 elongated with negative slope (Fig. 4), showing a negative correlation of the
236 parameters K_{P_0, T_0} and K' .

237 The evolution of the lattice parameters with pressure shows a slight anisotropic
238 compressional pattern. The “axial bulk moduli”, calculated with a second-order
239 “linearized” BM-EoS (Angel 2000 for details), are: $K_{P_0, T_0}(a) = 114(2)$ GPa for the a -
240 axis, $K_{P_0, T_0}(b) = 140(4)$ GPa for the b -axis, and $K_{P_0, T_0}(c) = 92(1)$ GPa for the c -axis,
241 with a general anisotropic compressional scheme:
242 $K_{P_0, T_0}(a):K_{P_0, T_0}(b):K_{P_0, T_0}(c) = 1.24:1.52:1$ (Table 4). The second-order BM-EoS fits
243 provide the best figure of merit. The monoclinic β -angle decreases linearly with
244 pressure, with: $\beta_P(^{\circ}) = \beta_{P_0} (^{\circ}) - 0.0902(4)P$ ($R^2 = 0.997$, with P in GPa) (Fig. 2 and
245 S1).

246 Magnitude and orientation of the principal unit-strain coefficients between
247 room pressure and the maximum P achieved (*i.e.*, $\Delta P = 8.51$ GPa, ramp in
248 methanol:ethanol:water mix), derived on the basis of the finite Eulerian strain tensor,
249 were calculated with the *Win_Strain* software (by RJ Angel, www.rossangel.com).
250 The following Cartesian axial system was chosen: $x//a^*$ and $y//b$. The strain ellipsoid
251 is oriented with the mid axis (ϵ_2) parallel to the b -axis, and the major (ϵ_1) and minor
252 (ϵ_3) axes lying in the (010)-plane: ϵ_1 describes an angle of $136.6(2)^{\circ}$ from a (and thus
253 $22.3(2)^{\circ}$ from c), as shown in Fig. 1. The elastic behaviour of allanite based on the
254 unit-strain coefficients between 0.0001 and 8.51 GPa is more anisotropic if compared
255 to that deduced only along the principal crystallographic directions, being $\epsilon_1 = -$
256 $0.02849(5)$, $\epsilon_2 = -0.0192(5)$, and $\epsilon_3 = -0.01306(8)$ GPa $^{-1}$, with the resulting anisotropic
257 scheme: $\epsilon_1:\epsilon_2:\epsilon_3 = 2.18:1.47:1$.

258 As only a few data-points were collected in Ne (Tables 1 and 2), such an
259 experiment was mainly aimed to demonstrate that, at least up to 16 GPa, allanite is
260 still crystalline. Thus, Ne-data were not used for the compressional analysis.

261

262 **Results: Deformation mechanisms at the atomic scale**

263 The mechanisms at the atomic scale that govern the anisotropic compression of
264 the allanite structure can be described in terms of intra- and inter-polyhedral re-
265 arrangement in response to the applied pressure. Intra-polyhedral deformations are
266 usually described in terms of compression of the bond distances or by distortion (*i.e.*,
267 with or without bond-distances compression), inter-polyhedral rearrangements in
268 terms of polyhedral tilting.

269 If we consider the *P*-induced atomic displacements in allanite structure, the
270 oxygen sites *O8* and *O9* show the most pronounced displacements from their
271 positions refined at 0.0001 GPa. *O9* is the bridging oxygen between the *T1* and *T2*
272 tetrahedra, describing the angle *T1-O9-T2* of $\sim 145.5^\circ$ at 0.0001 GPa, which decreases
273 to $\sim 140.4^\circ$ at 8.5 GPa (Table S2); the difference is remarkable, if we consider the *P*-
274 range. This tilting mechanism, governed by the displacement of the *O9* site, affects
275 the shape of the 5-membered rings of polyhedra (*M2-T3-M2-T2-T1*, Fig. 1), confining
276 the cavities in which the *A1* site lies: the contraction of the $O3 \leftrightarrow O1$, $O6 \leftrightarrow O1$ and
277 $O9 \leftrightarrow O5$ “diameters” is significantly different, being respectively ~ 0.09 , ~ 0.06 , and
278 ~ 0.03 Å within the *P*-range 0.0001-8.5 GPa. *O9* is also bonded to the *A1* site, and the
279 displacement of the *O9* leads to a change of the *A1-O9* bond length: ~ 3.095 Å at
280 0.0001 GPa and ~ 3.074 Å at 8.5 GPa (Table S2).

281 *O8* is the bridging oxygen between the *T2* and *M3* polyhedra. The *T2-O8-M3*
282 angle is $\sim 130.3^\circ$ at 0.0001 GPa and drastically decreases to $\sim 123.0^\circ$ at 8.5 GPa (Table
283 S2). The displacement of the *O8* site, and the aforementioned and co-related
284 polyhedral tilting, affect the shape evolution with *P* of the 8-membered rings of
285 polyhedra (*M2-T3-M3-T2-M2-T3-M3-T2*), in which the *A2* site lies (Fig. 1). More
286 specifically, the $O8 \leftrightarrow O8$ “diameter” is pronouncedly shortened by ~ 0.30 Å at 8.51
287 GPa and, in the same *P*-range, the $O3 \leftrightarrow O3$, $O6 \leftrightarrow O10$ and $O6 \leftrightarrow O6$ diameters are
288 shortened by ~ 0.20 Å, ~ 0.15 Å, and ~ 0.16 Å, respectively (Fig. 1, Table 3). This leads
289 to a more rectangular-edged ring at high pressure, affecting mainly the length of the
290 unit-cell edge parallel to the *c*-axis.

291 The aforementioned polyhedral tilting mechanisms are coupled with the intra-
292 polyhedral distortion and compression in response to the applied pressure, which are
293 energetically more costly and, therefore, less pronounced at low- or mid- P regimes.
294 The evolution of the intra-polyhedral bond distances and angles shows that, within the
295 P -range investigated, tetrahedra behaves as quasi-rigid units at a first approximation
296 (with minor bond-distances shortening or polyhedral distortion), octahedra are more
297 affected by compression and distortion (though not dramatic), whereas the large $A1$ -
298 and $A2$ -polyhedra are the most affected by distortion and bond-distances shortening in
299 response to the applied pressure (Table S2). However, the quality of the structural
300 data at high pressure allow us to observe a different behaviour among the three
301 independent tetrahedra: $T1$ and $T2$ are essentially rigid (*i.e.*, difference in their bond
302 lengths of the same order of the *e.s.ds*), but $T3$ tends to deform significantly, in
303 particular in response to the shortening of the $T2-O2$ distance of about 0.03 Å (with
304 $\Delta P = 8.5$ GPa; Table S2). Even among the three independent octahedra, we can
305 observe different magnitude of (compression+distortion): within the P -range
306 investigated, the maximum shortening of the intra-polyhedral bond lengths is ~ 0.04 Å
307 for the $M2$ octahedron, ~ 0.06 Å for the $M1$ octahedron, and ~ 0.10 Å for the $M3$
308 octahedron. Therefore, the Fe-rich $M3$ octahedron is the most distorted one under
309 hydrostatic compression, even in response to the pronounced $T2-O8-M3$ compression
310 described above.

311 One open question concerns the role played by the H-bonding scheme on the P -
312 induced structure evolution of allanite. In allanite structure (as in all the epidote-group
313 members), there is only one unique H site. $O10$ acts as *donor* and $O4$ acts as *acceptor*
314 of the H-bond (*i.e.*, $O10-H\dots O4$); $O4$ is the oxygen co-shared by two $M1$ and one $M3$
315 octahedra and $O10$ is the bridging oxygen between the edge-sharing $M2$ octahedra.
316 Proton, donor and acceptor are confined in the 5-membered ring. It is impossible to
317 refine the H site coordinates in the HP structure refinements. However, we can
318 intuitively consider that the H-bonding geometry can explain why in the 5-membered
319 ring the major deformation mechanism acts on one side through the $T1-O9-T2$ tilting,
320 as the $T1$ and $T2$ tetrahedra are unaffected by the H-bonding. As a consequence, the
321 $O3\leftrightarrow O1$ shortening is more pronounced than the $O6\leftrightarrow O1$ and $O9\leftrightarrow O5$ ones (Fig. 1,
322 Tables S2 and 3).

323

324 Discussion and conclusions

325 To the best of our knowledge, this is the first study on the compressional
326 behaviour of allanite, here described on the basis of *in-situ* synchrotron single-crystal
327 diffraction data. The experimental findings of this study confirm that allanite
328 preserves its crystallinity and behaves elastically at least up to 16 GPa (at 298 K),
329 under hydrostatic compression (Table 2).

330 The Eulerian unit-strain ellipsoid, calculated between 0.0001 and 8.51 GPa,
331 confirms that the lowest and the highest compression directions lie on the (010)-plane,
332 as shown in Fig. 1: the softest direction (ϵ_1) describes an angle of $136.6(2)^\circ$ from
333 [100], and as a consequence the stiffest direction (ϵ_3) describes an angle of $46.6(2)^\circ$
334 from [100]. A recalculation of magnitude and orientation of the compressional unit-
335 strain ellipsoids of epidote with 0.74 Fe a.p.f.u. (based on the data of Gatta et al.
336 2011b, for $\Delta P = 8.30$ GPa, $P_{\min} = 0.0001$ GPa), of epidote with 0.79 Fe a.p.f.u. (based
337 on the data of Qin et al. 2016, for $\Delta P = 9.70$ GPa, $P_{\min} = 0.0001$ GPa), and clinozoisite
338 with 0.40 Fe a.p.f.u. (based on the data of Qin et al. 2016, for $\Delta P = 8.80$ GPa, $P_{\min} =$
339 0.0001 GPa), was done, using the same Cartesian axial system and strain definition
340 (*i.e.*, Eulerian) adopted for allanite:

- 341 1) in epidote with 0.79 Fe a.p.f.u., the stiffest direction ($\epsilon_3 = -0.0232(2)$ GPa⁻¹)
342 is parallel to [010], whereas the mid ($\epsilon_2 = -0.0236(3)$ GPa⁻¹) and the softest
343 ($\epsilon_1 = -0.0258(8)$ GPa⁻¹) directions lie on (010), with the softest one
344 describing an angle of $63(5)^\circ$ with [100]; anisotropic compressional scheme:
345 $\epsilon_1:\epsilon_2:\epsilon_3 = 1.11:1.01:1$, *i.e.* there is an almost circular section of the ellipsoid
346 in which ϵ_2 and ϵ_3 are dispersed; the monoclinic β angle decreases
347 monotonically with P (Qin et al. 2016);
- 348 2) in epidote with 0.74 Fe a.p.f.u., the stiffest direction ($\epsilon_3 = -0.01646(8)$ GPa⁻¹)
349 is parallel to [010], whereas the mid ($\epsilon_2 = -0.01978(8)$ GPa⁻¹) and the
350 softest ($\epsilon_1 = -0.02352(6)$ GPa⁻¹) directions lie on (010), with the softest one
351 describing an angle of $126.7(8)^\circ$ with [100]; anisotropic compressional
352 scheme: $\epsilon_1:\epsilon_2:\epsilon_3 = 1.43:1.20:1$; the monoclinic β angle decreases
353 monotonically with P (Gatta et al. 2011b);
- 354 3) in clinozoisite (0.40 Fe a.p.f.u.), the stiffest direction ($\epsilon_3 = -0.0138(2)$ GPa⁻¹)
355 is parallel to [010], whereas the mid ($\epsilon_2 = -0.0145(2)$ GPa⁻¹) and the
356 softest ($\epsilon_1 = -0.0234(1)$ GPa⁻¹) directions lie on (010), with the softest one

357 describing an angle of $174(1)^\circ$ with [100]; anisotropic compressional
358 scheme: $\epsilon_1:\epsilon_2:\epsilon_3 = 1.70:1.05:1$; the monoclinic β angle decreases
359 monotonically with P (Qin et al. 2016).

360 On this basis, some general conclusions can be drawn:

- 361 1) The softest directions in allanite, epidote(s) and clinozoisite lie all on (010);
- 362 2) The stiffest directions in epidote and clinozoisite are consistently oriented
363 parallel to [010], whereas in allanite it lies on (010) (*i.e.*, $\epsilon_3 \angle [100] =$
364 $46.6(2)^\circ$);
- 365 3) The compressional schemes of allanite, epidote(s) and clinozoisite are
366 significantly different: more anisotropic in allanite ($\epsilon_1:\epsilon_2:\epsilon_3 = 2.18:1.47:1$),
367 less in epidote ($\epsilon_1:\epsilon_2:\epsilon_3 = 1.43:1.20:1$ with 0.74 Fe a.p.f.u, and $\epsilon_1:\epsilon_2:\epsilon_3 =$
368 $1.11:1.01:1$ with 0.79 Fe a.p.f.u) and clinozoisite ($\epsilon_1:\epsilon_2:\epsilon_3 = 1.70:1.05:1$).
- 369 4) Concerning the joint epidote-clinozoisite, it is observable a drastic rotation of
370 the unit-strain ellipsoid in response to the different Fe content: the softest
371 direction describes an angle from [100] of $174(1)^\circ$ in clinozoisite with 0.40 Fe
372 a.p.f.u., of $126.7(8)^\circ$ in epidote with 0.74 Fe a.p.f.u. and of $63(5)^\circ$ in epidote
373 with 0.79 Fe a.p.f.u., *i.e.* the lower the Fe content, the higher the angle
374 $\epsilon_1 \angle [100]$.

375 The stiffest direction can easily assume a different orientation in response to small
376 compositional changes, due to the modest anisotropy observed ($\epsilon_2:\epsilon_3 = 1.01:1$ and
377 $1.20:1$ in epidote, $1.05:1$ in clinozoisite, and $1.47:1$ in allanite).

378 Our data on allanite show that the inter- and intra-polyhedral mechanisms,
379 described on the basis of the structure refinements at high pressure, are not sufficient
380 to explain, unambiguously, magnitude and orientation of the unit-strain ellipsoid. In
381 this light, only some general considerations can be done:

- 382 1) The Eulerian unit-strain ellipsoid shows that the stiffest direction (ϵ_3)
383 lies on (010), it is perpendicular to the softest one and is likely governed
384 by the presence of (almost incompressible) edge-sharing $M3-M1-M3$
385 clusters of octahedra (almost parallel to ϵ_3), connected to the $A2$ sites
386 (Fig. 1, Table S2).
- 387 2) The intermediate one (ϵ_2) is parallel to [010]. The compression of the
388 allanite structure along [010] is likely hindered by the presence of the
389 (infinite) edge-sharing $M1O_6$ octahedral chains, running along [010]

390 (Fig. 1, Table S2). Edge-sharing octahedral chains act as “pillars”, and
391 the shortening along the chain direction can only occur through intra-
392 polyhedral deformation, by homogeneous or non-homogeneous (*i.e.*,
393 polyhedral distortion) bond-distances compression.

394 3) The most compressible direction (ϵ_1 , as defined above) in allanite
395 structure lies also on (010). Tilting and (compression+distortion) of the
396 polyhedra, described above, generate, in turns, the deformations of the
397 5- and 8-membered rings of polyhedra. The most pronounced
398 compression directions of the rings (represented *e.g.* by the diameters
399 $O3 \leftrightarrow O1$ of the 5-membered ring and by $O8 \leftrightarrow O8$ of the 8-membered
400 ring) can play an important role on the orientation of ϵ_1 (Fig. 1, Table
401 S2).

402 The compressional elastic anisotropy described on the basis of the “linearised
403 bulk moduli” along the three crystallographic axes, all obtained on the basis of a
404 second-order BM-EoS fit (as the truncation to second-order provide the best figures of
405 merit), is: $K_{P0,T0}(a):K_{P0,T0}(b):K_{P0,T0}(c) = 1.24:1.52:1$ in allanite, $K_{T0}(a):K_{T0}(b):K_{T0}(c) =$
406 $1.13:1.48:1$ in epidote with 0.74 Fe a.p.f.u. (recalculated from Gatta et al. 2011b) and
407 $K_{T0}(a):K_{T0}(b):K_{T0}(c) = 1:1.19:1.04$ in epidote with 0.79 Fe a.p.f.u. (Qin et al. 2016),
408 and $K_{P0,T0}(a):K_{P0,T0}(b):K_{P0,T0}(c) = 1:1.81:1.52$ in clinozoisite with 0.40 Fe a.p.f.u.
409 (Qin et al. 2016).

410 The bulk compression of allanite, epidote and clinozoisite is significantly
411 different (Fig. 2). Using a third-order BM-EoS, the following parameters are obtained:
412 $K_{P0,T0}=131(4)$ GPa and $K'=1.9(8)$ for allanite, $K_{P0,T0}=111(3)$ GPa and $K'=7.6(7)$ for
413 epidote with 0.74 Fe a.p.f.u. (Gatta et al. 2011b) and $K_{P0,T0}=115(2)$ GPa and
414 $K'=3.7(2)$ for epidote with 0.79 Fe a.p.f.u. (Qin et al. 2016), and $K_{P0,T0}=142(3)$ GPa
415 and $K'=5.2(4)$ for clinozoisite with 0.40 Fe a.p.f.u. (Qin et al. 2016). Epidote is the
416 softest one, clinozoisite is the stiffest and allanite lies in between.

417 On the basis of a comparative analysis of the compressional behavior of
418 epidotes and clinozoisites reported in the literature, along with those obtained
419 experimentally in their study, Qin et al. (2016) concluded that increasing Fe content
420 reduces the bulk modulus and increases the first P -derivative (using the BM-EoS
421 formulation). This behavior was attributed to the differences in compression behavior
422 due to the addition of Fe at the $M3$ site (replacing Al) in the epidote structure.

423 Additional *in-situ* HP Raman data collected by Qin et al. (2016) further suggest that
424 the difference in compressibility between epidote and clinozoisite is likely to be
425 different compressibility of FeO₆- and AlO₆-octahedra, coordinated by the M3 sites.
426 In allanite, the scenario is more complex: Fe and Al mainly populate the octahedral
427 sites (Fe occurs principally at the M3 site, like in epidote), but a fraction of Al likely
428 replaces Si at the tetrahedral sites and, more important, Fe along with a series of other
429 LREE elements (mainly Ce and La) replace Ca at the A1 and A2 sites. In this light, the
430 conclusion of Qin et al (2016), about the role played by iron content on the
431 compressional behavior, cannot be simplistically extended to allanite, as more
432 variables can influence the different behavior of allanite if compared to the other two
433 aforementioned members of the epidote group.

434

435 **Acknowledgements**

436 PETRA-III synchrotron facility (Hamburg, Germany) is acknowledged for provision
437 of beamtime at P02.2 beamline. G.D.G., S.M., L.C., D.C., P.L., M.M. acknowledge
438 the support of the Italian Ministry of Education (MIUR) through the project
439 “Dipartimenti di Eccellenza 2018–2022”. W. Crichton and an anonymous reviewer
440 are thanked for the revision of the manuscript.

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Table 1. Lattice parameters of allanite at different pressures, based on the two datasets collected using the methanol:ethanol:water mix (HP-mew) and neon (HP-Ne) as P -transmitting fluids (P -uncertainty: ± 0.1 GPa)

Experiment	P (GPa)	a (Å)	b (Å)	c (Å)	β (Å)	V (Å ³)
HP-mew ^a	0.0001	8.9116(3)	5.738(2)	10.1447(4)	114.979(4)	470.2(2)
HP-mew ^b	0.0001	8.9130(2)	5.735(2)	10.1484(3)	114.993(3)	470.2(2)
HP-mew	1.17	8.8882(4)	5.719(2)	10.1071(4)	114.872(6)	466.1(2)
HP-mew	1.60	8.8781(4)	5.712(2)	10.0911(3)	114.832(5)	464.42(9)
HP-mew	1.97	8.8718(4)	5.707(2)	10.0821(4)	114.805(5)	463.4(1)
HP-mew	2.81	8.8497(2)	5.697(8)	10.0509(2)	114.707(3)	460.33(7)
HP-mew	3.30	8.8416(4)	5.6890(2)	10.0383(4)	114.684(5)	458.8(1)
HP-mew	3.61	8.8325(3)	5.6880(9)	10.0293(3)	114.647(3)	457.96(7)
HP-mew	3.99	8.8223(2)	5.6834(8)	10.0152(2)	114.593(3)	456.62(7)
HP-mew	4.86	8.7996(3)	5.674(1)	9.9865(3)	114.519(4)	453.63(8)
HP-mew	5.08	8.7963(2)	5.6686(9)	9.9813(2)	114.512(3)	452.84(7)
HP-mew	5.55	8.7851(2)	5.6657(9)	9.9667(2)	114.462(3)	451.55(7)
HP-mew	6.45	8.7655(2)	5.6550(9)	9.9412(2)	114.388(3)	448.80(7)
HP-mew	7.07	8.7466(3)	5.644(2)	9.9177(3)	114.321(4)	446.2(1)
HP-mew	7.61	8.7384(3)	5.638(2)	9.9073(3)	114.287(4)	444.94(9)
HP-mew	8.51	8.7231(3)	5.629(2)	9.8817(3)	114.225(4)	442.50(9)
HP-Ne ^c	1.39	8.8881(6)	5.7303(3)	10.111(2)	114.88(2)	467.2(2)
HP-Ne ^c	9.90	8.668(3)	5.6171(2)	9.839(6)	113.87(6)	438.1(3)
HP-Ne ^c	14.64	8.563(3)	5.5643(2)	9.722(7)	113.56(7)	424.6(4)

^aSample in the DAC without P -medium

^bAfter decompression.

^cFor the experiments in Ne, the P -value at any data point represents the average value obtained by the ruby-fluorescence method and the BM-EoS fit based on the *mew* experiments.

Table 2. Details pertaining to the data collections and structure refinements of allanite at different pressures (P -uncertainty: ± 0.1 GPa).

P (GPa)	0.0001 ^a	1.17	1.60	1.97	3.61	3.99	4.86	5.08	5.55
a (Å)	8.9116(3)	8.8882(4)	8.8781(4)	8.8718(4)	8.8325(3)	8.8223(2)	8.7996(3)	8.8223(2)	8.7996(3)
b (Å)	5.738(2)	5.719(2)	5.712(2)	5.707(2)	5.6880(9)	5.6834(8)	5.674(1)	5.6834(8)	5.674(1)
c (Å)	10.1447(4)	10.1071(4)	10.0911(3)	10.0821(4)	10.0293(3)	10.0152(2)	9.9865(3)	10.0152(2)	9.9865(3)
β (Å)	114.979(4)	114.872(6)	114.832(5)	114.805(5)	114.647(3)	114.593(3)	114.519(4)	114.593(3)	114.519(4)
V (Å ³)	470.2(2)	466.1(2)	464.42(9)	463.4(2)	457.96(7)	456.62(7)	453.63(8)	456.62(7)	453.65(8)
Space group	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$
λ (Å)	0.28940	0.28940	0.28940	0.28940	0.28940	0.28940	0.28940	0.28940	0.28940
θ_{\max} (°)	18.08	17.99	18.01	18.02	18.10	18.11	18.15	18.11	18.15
No. measured reflections	2272	2147	2186	2127	2137	2170	2020	2170	2020
No. unique reflections	1106	1189	1185	1168	1148	1135	1059	1135	1059
No. unique reflections with $F_o > 4\sigma(F_o)$	1106	1189	1185	1168	1148	1135	1059	1135	1059
No. refined parameters	66	63	63	63	63	63	63	63	63
R_{int}	0.0159	0.0189	0.0203	0.0221	0.0274	0.0207	0.0172	0.0207	0.0172
R_1 (F) with $F_o > 4\sigma(F_o)$	0.0356	0.0351	0.0396	0.0466	0.0452	0.0403	0.0404	0.0403	0.0404
wR_2 (F^2)	0.0952	0.0931	0.1046	0.1220	0.1229	0.1021	0.1057	0.1021	0.1057
GooF	1.319	1.198	1.326	1.509	1.447	1.325	1.387	1.325	1.387
Residuals (e ⁻ /Å ³)	-0.86/+1.01	-0.77/+1.12	-1.06/1.16	-1.12/1.30	-1.27/1.50	-1.34/1.03	-1.45/1.13	-1.34/1.03	-1.45/1.13
P (GPa)	6.45	7.07	7.61	8.51	1.39 ^b	9.90 ^b	14.64 ^b		
a (Å)	8.7655(2)	8.7466(3)	8.7384(3)	8.7231(3)	8.8881(6)	8.668(3)	8.563(3)		
b (Å)	5.6550(9)	5.644(2)	5.638(2)	5.6475(2)	5.7303(3)	5.6171(2)	5.5643(2)		
c (Å)	9.9412(2)	9.9177(3)	9.9073(3)	9.8817(3)	10.111(2)	9.839(6)	9.722(7)		
β (Å)	114.388(3)	114.321(4)	114.287(4)	114.225(4)	114.88(2)	113.87(6)	113.56(7)		
V (Å ³)	448.80(7)	446.2(1)	444.94(9)	445.2(3)	467.2(2)	438.1(3)	424.6(4)		
Space group	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$	$P2_1/m$		
λ (Å)	0.28940	0.28940	0.28940	0.28940	0.28940	0.28940	0.28940		
θ_{\max} (°)	17.93	17.97	17.98	18.02	18.11	17.43	17.63		
No. measured reflections	1980	1834	2067	2158	1657	1432	1494		
No. unique reflections	1064	985	1053	1078	946	741	724		
No. unique reflections with $F_o > 4\sigma(F_o)$	1064	985	1053	1076	945	736	723		
No. refined parameters	63	63	63	63	63	63	64		
R_{int}	0.0232	0.0225	0.0190	0.0209	0.0176	0.0244	0.0215		
R_1 (F) with $F_o > 4\sigma(F_o)$	0.0421	0.0498	0.0385	0.0377	0.0442	0.0609	0.0479		
wR_2 (F^2)	0.1072	0.1206	0.1000	0.0985	0.1107	0.1523	0.1195		
GooF	1.331	1.481	1.350	1.302	1.480	1.970	1.601		
Residuals (e ⁻ /Å ³)	-1.35/1.02	-1.51/0.96	-1.10/1.06	-1.21/1.21	-1.38/1.03	-1.47/1.70	-1.03/1.23		

^aSample in the DAC without P -medium. The Ca vs. Ce fractions at the $A1$ and $A2$ sites and the Al vs. Fe fraction at the $M3$ site were refined at 0.0001 GPa and then kept fixed at higher pressure.

^bNeon used as pressure transmitting medium.

Table 3. Principal “diameters” (Å) of the 8-membered ring (O6↔O10, O3↔O3, O6↔O6 and O8↔O8) and of the 5-membered one (O3↔O1, O6↔O1, O9↔O5) at different pressures (P -uncertainty: ± 0.1 GPa).

Experiment	P (GPa)	O6↔O10	O3↔O3	O6↔O6	O8↔O8	O3↔O1	O6↔O1	O9↔O5
HP-mew ^a	0.0001	7.032(9)	7.297(10)	7.702(9)	4.778(8)	3.615(5)	4.364(11)	4.674(5)
HP-mew	1.17	7.011(10)	7.251(14)	7.677(7)	4.707(8)	3.606(5)	4.352(15)	4.672(5)
HP-mew	1.60	7.005(9)	7.251(12)	7.668(7)	4.691(9)	3.601(5)	4.349(13)	4.669(5)
HP-mew	1.97	7.001(12)	7.245(10)	7.660(9)	4.682(11)	3.596(6)	4.347(13)	4.672(6)
HP-mew	3.61	6.971(8)	7.198(8)	7.637(9)	4.624(10)	3.581(5)	4.338(9)	4.671(6)
HP-mew	3.99	6.968(7)	7.184(8)	7.631(7)	4.619(9)	3.579(5)	4.337(9)	4.655(5)
HP-mew	4.86	6.951(9)	7.162(10)	7.612(9)	4.587(9)	3.571(5)	4.332(11)	4.652(5)
HP-mew	5.08	6.948(8)	7.159(8)	7.612(9)	4.581(10)	3.568(5)	4.334(9)	4.657(6)
HP-mew	5.55	6.942(8)	7.146(8)	7.610(9)	4.560(10)	3.563(5)	4.326(9)	4.651(6)
HP-mew	6.45	6.925(8)	7.131(8)	7.592(9)	4.538(9)	3.554(5)	4.321(9)	4.649(6)
HP-mew	7.07	6.912(10)	7.119(11)	7.579(10)	4.515(11)	3.543(6)	4.314(12)	4.644(7)
HP-mew	7.61	6.901(8)	7.114(10)	7.566(9)	4.502(9)	3.539(4)	4.314(11)	4.644(5)
HP-mew	8.51	6.883(8)	7.101(10)	7.547(7)	4.478(9)	3.531(4)	4.309(11)	4.643(5)

^aSample in the DAC without P -medium

Table 4. Compressional parameters of allanite obtained by second- and third-order Birch-Murnaghan Equation of State fits (see text for details).

	V_0, l_0 ($\text{\AA}^3, \text{\AA}$)	K_{P_0, T_0} (GPa)	K'	EoS order
V_0	470.6(2)	122(1)	4 (fixed)	2 nd order
a_0	8.918(3)	114(2)	4 (fixed)	2 nd order
b_0	5.735(2)	140(4)	4 (fixed)	2 nd order
c_0	10.149(2)	92(1)	4 (fixed)	2 nd order
V_0	470.2(2)	131(4)	1.9(8)	3 rd order

Figure 1. Crystal structure of allanite viewed down [010] and [001], and overlying unit-strain ellipsoid based on Eulerian finite strain calculated between ambient pressure and 8.51 GPa [$\epsilon_2 // b$, ϵ_1 and ϵ_3 lying in the (010)-plane, $\epsilon_1 \angle a = 136.6(2)^\circ$; $\epsilon_1 = -0.02849(5)$, $\epsilon_2 = -0.0192(5)$, and $\epsilon_3 = -0.01306(8)$ GPa⁻¹, $\epsilon_1 : \epsilon_2 : \epsilon_3 = 2.18 : 1.47 : 1$]. Si-tetrahedra (coordinated by *Si1*, *Si2* and *Si3* sites) and Al/Fe-octahedra (coordinated by the *M1*, *M2* and *M3* sites) are shown as closed-faces polyhedra; large spheres represent the *A1* and *A2*-sites. The 8-membered ring of polyhedra (with the diameters $O3 \leftrightarrow O3$, $O10 \leftrightarrow O6$, $O6 \leftrightarrow O6$ and $O8 \leftrightarrow O8$) and the 5-membered ring (with the diameters $O3 \leftrightarrow O1$, $O1 \leftrightarrow O6$ and $O9 \leftrightarrow O5$) are also shown (see text for further details).

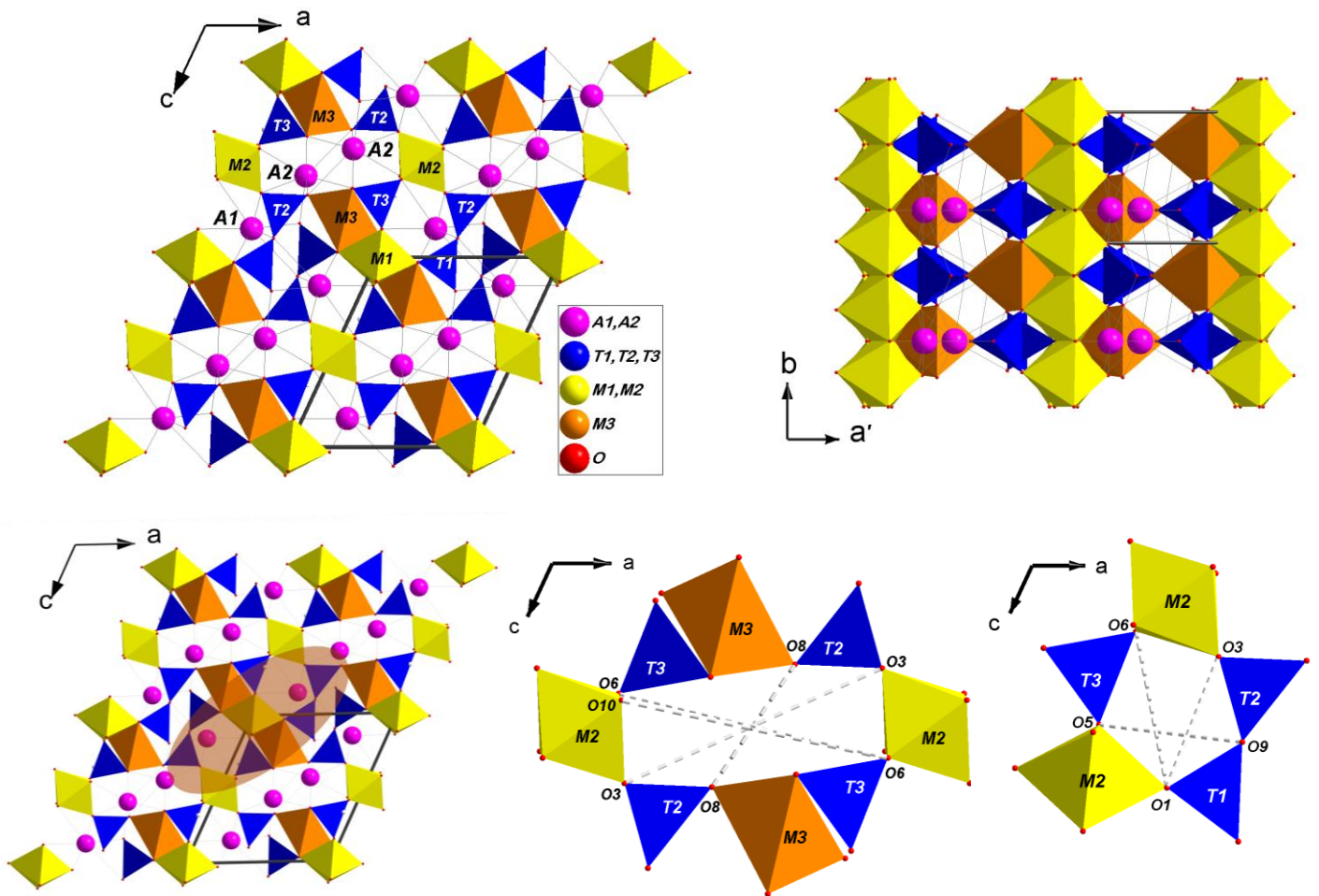
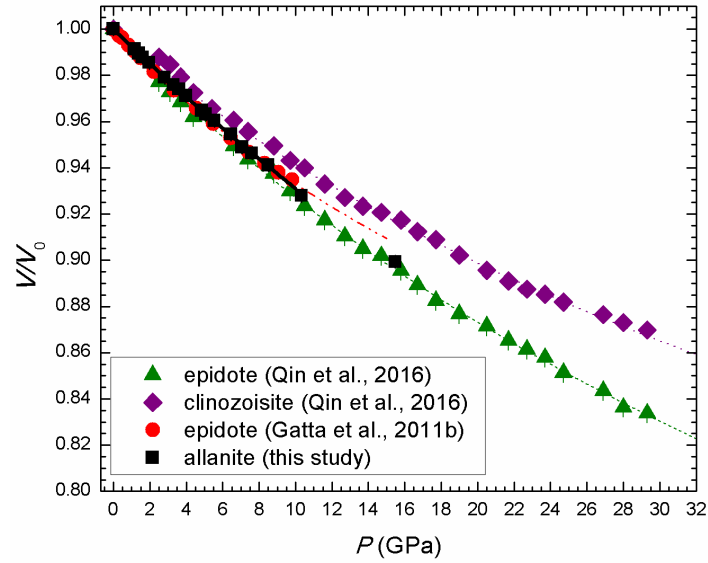
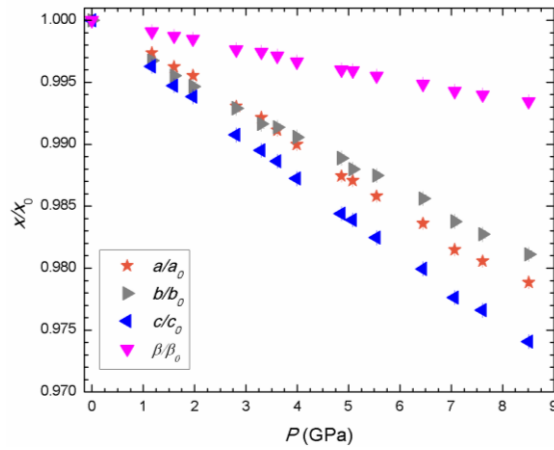


Figure 2. Evolution of the lattice parameters of allanite with P (GPa). (a) For the unit-cell volume, the solid lines represent the third-order BM-EoS fits. The axial compressional behaviours of (b) allanite (this study), (c) epidotes with 0.74 Fe a.p.f.u. (Gatta et al. 2011b), (d) epidotes with 0.79 Fe a.p.f.u. (Qin et al. 2016), and (e) clinozoisite with 0.40 Fe a.p.f.u. (Qin et al. 2016) are also shown.

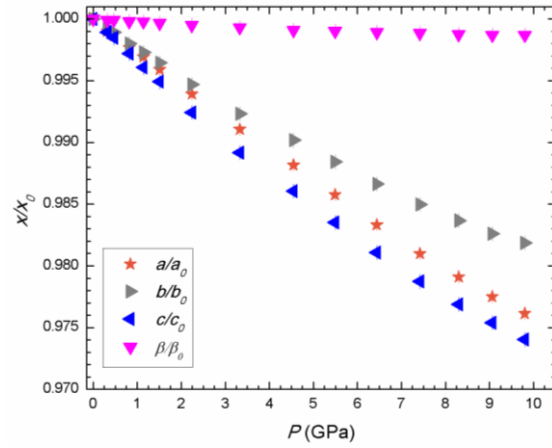
a)



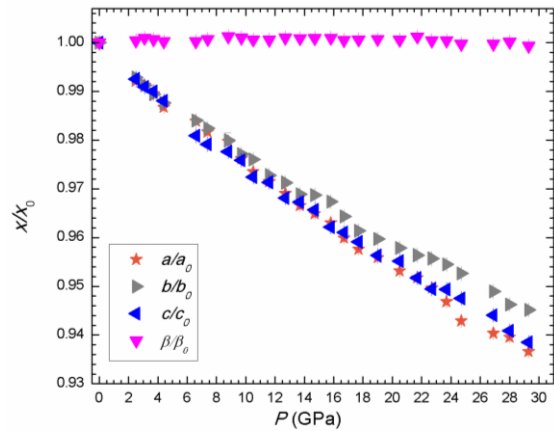
b)



c)



d)



e)

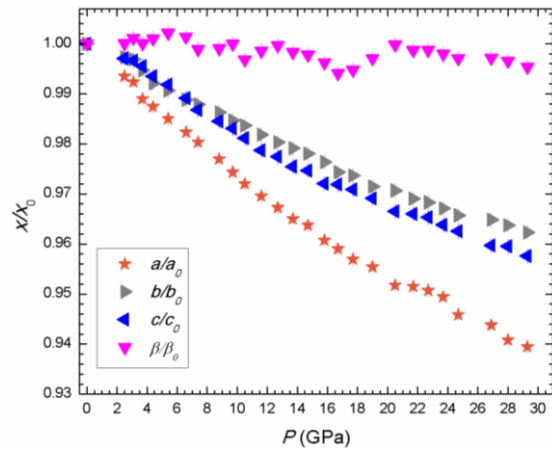


Figure 3. Normalized stress ($F_e = P/[3fe(1+2fe)^{5/2}]$) vs. Eulerian finite strain ($fe = [(V_0/V)^{2/3} - 1]/2$) plot. The *esds* have been calculated according to Heinz and Jeanloz (1984). The solid line is a weighted linear fit through the data.

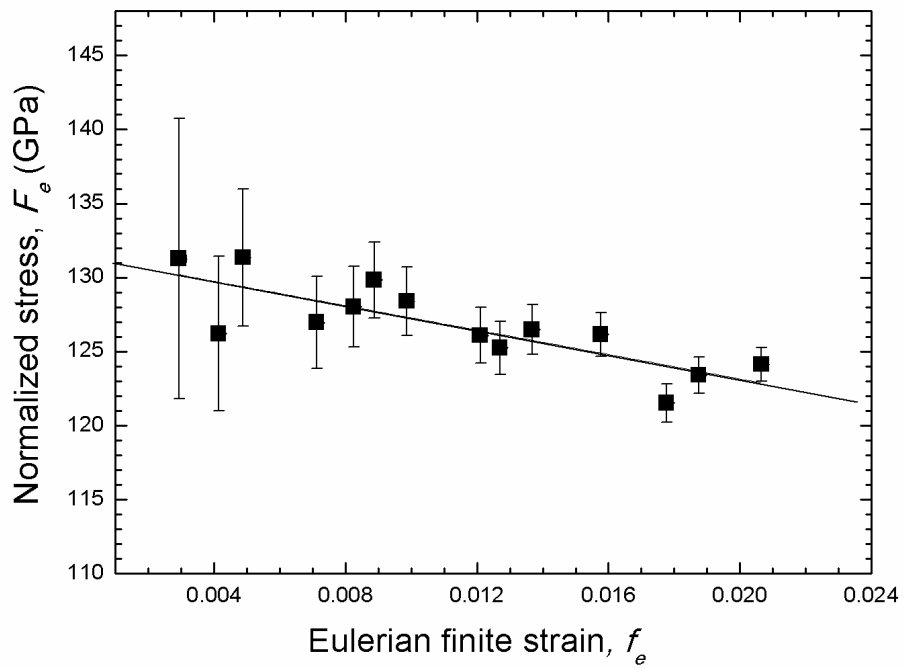


Figure 4. Confidence ellipses at 68.3% level ($\Delta\chi^2=2.30, \pm 1\sigma$, solid black line), 95.4% level ($\Delta\chi^2=6.17, \pm 2\sigma$, dashed black line) and 99.7% level ($\Delta\chi^2=11.8, \pm 3\sigma$, dotted black line) calculated starting from the variance-covariance matrix of K_{P_0,T_0} and K' obtained from the least-square procedure. The K_{P_0,T_0} and K' of Gatta et al. (2011b) for epidote (0.74 Fe a.p.f.u.; red circle), of Qin et al. (2016) for epidote (0.79 Fe a.p.f.u.; dark green triangle), and of Qin et al. (2016) for clinozoisite (0.40 Fe a.p.f.u.; purple diamond) are added for comparison (see text for further details). Error bars: ± 1 esds.

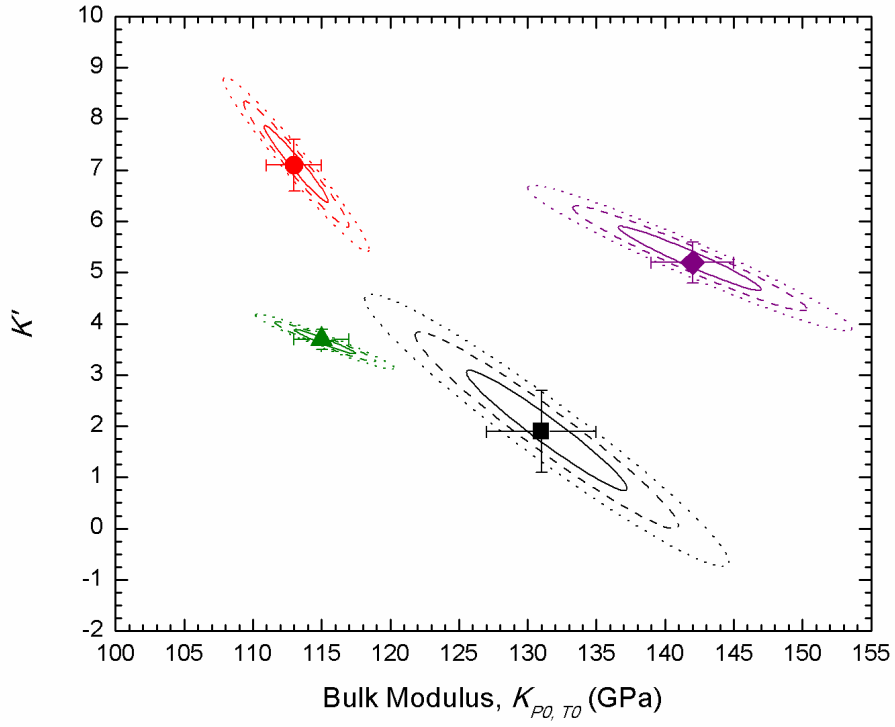


Figure 5. Evolution with P of the (normalised) $O3\leftrightarrow O3$, $O6\leftrightarrow O6$, $O8\leftrightarrow O8$ and $O10\leftrightarrow O6$ “diameters” of the 8-membered ring of polyhedra, and of the $O3\leftrightarrow O1$, $O1\leftrightarrow O6$ and $O9\leftrightarrow O5$ “diameters” of the 5-membered ring (see Fig. 1 and text for further details).

