



ParisTech



# Nd<sup>3+</sup> environment and solubility in aluminoborosilicate glasses

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# Context : research for nuclear glasses with high waste loading

- Rare earths (RE) are a significant part of the fission products (FP) and good simulants of trivalent actinides
- RE oxide concentration is  $\sim 4$  wt% in the current nuclear glass (R7T7). It will amount to  $\sim 10$  wt% in future nuclear glasses with higher waste loading

Chemical family	weight (kg/U)
Alkalis (Cs, Rb)	3
Alkaline-earths (Sr, Ba)	2.4
<b>Rare-earths</b>	<b>10.2</b>
Transition metals (Mo, Zr, Tc)	7.7
Chalcogens (Se, Te)	0.5
Halogens (I, Br)	0.2
Noble metals (Ru, Rh, Pd)	3.9
Others (Ag, Cd, Sn, Sb...)	0.1

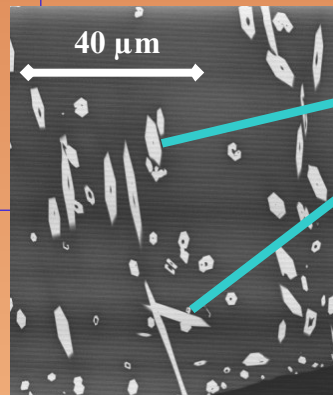


*Main families of FP in  $\text{UO}_2$  spent fuel initially enriched with 3.5%  $^{235}\text{U}$*

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New glass compositions must avoid crystallization of big crystals during melt cooling



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# Aims of the study and method

- Characterizing the  $\text{Nd}^{3+}$  environment in a model glass (A)
- Studying the impact of  $\text{Nd}_2\text{O}_3$  content on the aluminoborosilicate network structure and on the crystallization properties
- Draw structure – crystallization relationships and test compositional variations to improve the solubility of  $\text{Nd}^{3+}$  in the glass

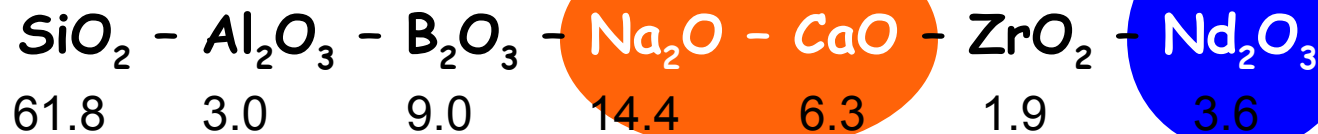
Model glass A (mol%) :

$\text{SiO}_2$	-	$\text{Al}_2\text{O}_3$	-	$\text{B}_2\text{O}_3$	-	$\text{Na}_2\text{O}$	-	$\text{CaO}$	-	$\text{ZrO}_2$	-	$\text{Nd}_2\text{O}_3$
61.8		3.0		9.0		14.4		6.3		1.9		3.6

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Model glass A (mol%) :



Use of glass series and spectroscopic tools (NMR, Raman, Vis absorption, EXAFS)

- Simple reference glasses
- $\text{CaO}/\text{Na}_2\text{O}$  ratio and alkali type
- $\text{Nd}_2\text{O}_3$  content from 0 to 30 wt% (7.5 mol%)

Focus on the  $\text{Nd}^{3+}$  environment and crystallization

# Nd<sup>3+</sup> environment in the model glass

## Reference glasses

Glass	Composition (mol%)
Na-silicate	74.38 SiO <sub>2</sub> - 21.29 Na <sub>2</sub> O - 4.33 Nd <sub>2</sub> O <sub>3</sub>
Na-rich borate	64 B <sub>2</sub> O <sub>3</sub> - 35 Na <sub>2</sub> O - 1 Nd <sub>2</sub> O <sub>3</sub>
Na-poor borate	79 B <sub>2</sub> O <sub>3</sub> - 20 Na <sub>2</sub> O - 1 Nd <sub>2</sub> O <sub>3</sub>
Nd aluminosilicate	75 SiO <sub>2</sub> - 15 Al <sub>2</sub> O <sub>3</sub> - 10 Nd <sub>2</sub> O <sub>3</sub>
Nd metaborate	75 B <sub>2</sub> O <sub>3</sub> - 25 Nd <sub>2</sub> O <sub>3</sub>

NBO's

BO's and  
NBO's

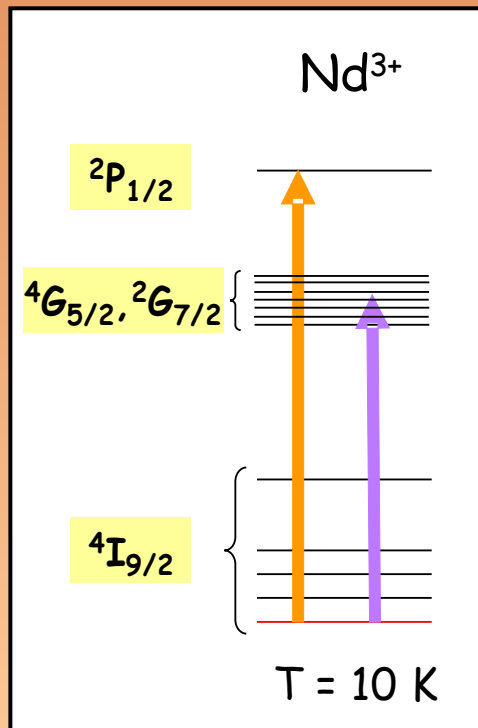
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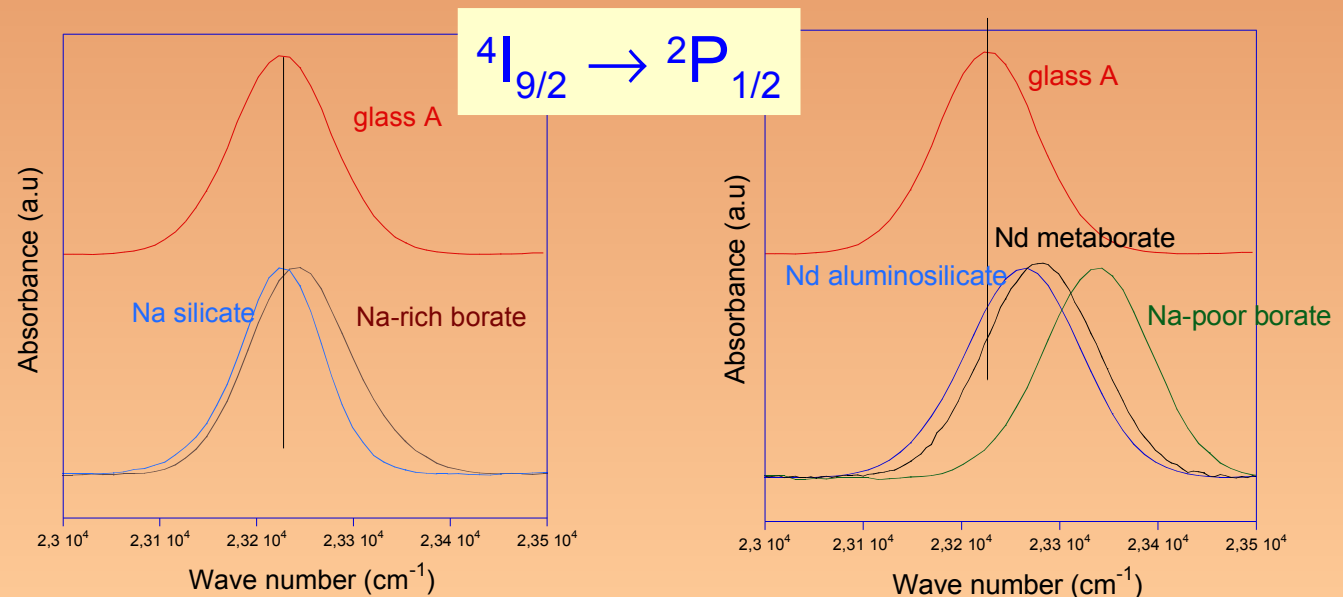
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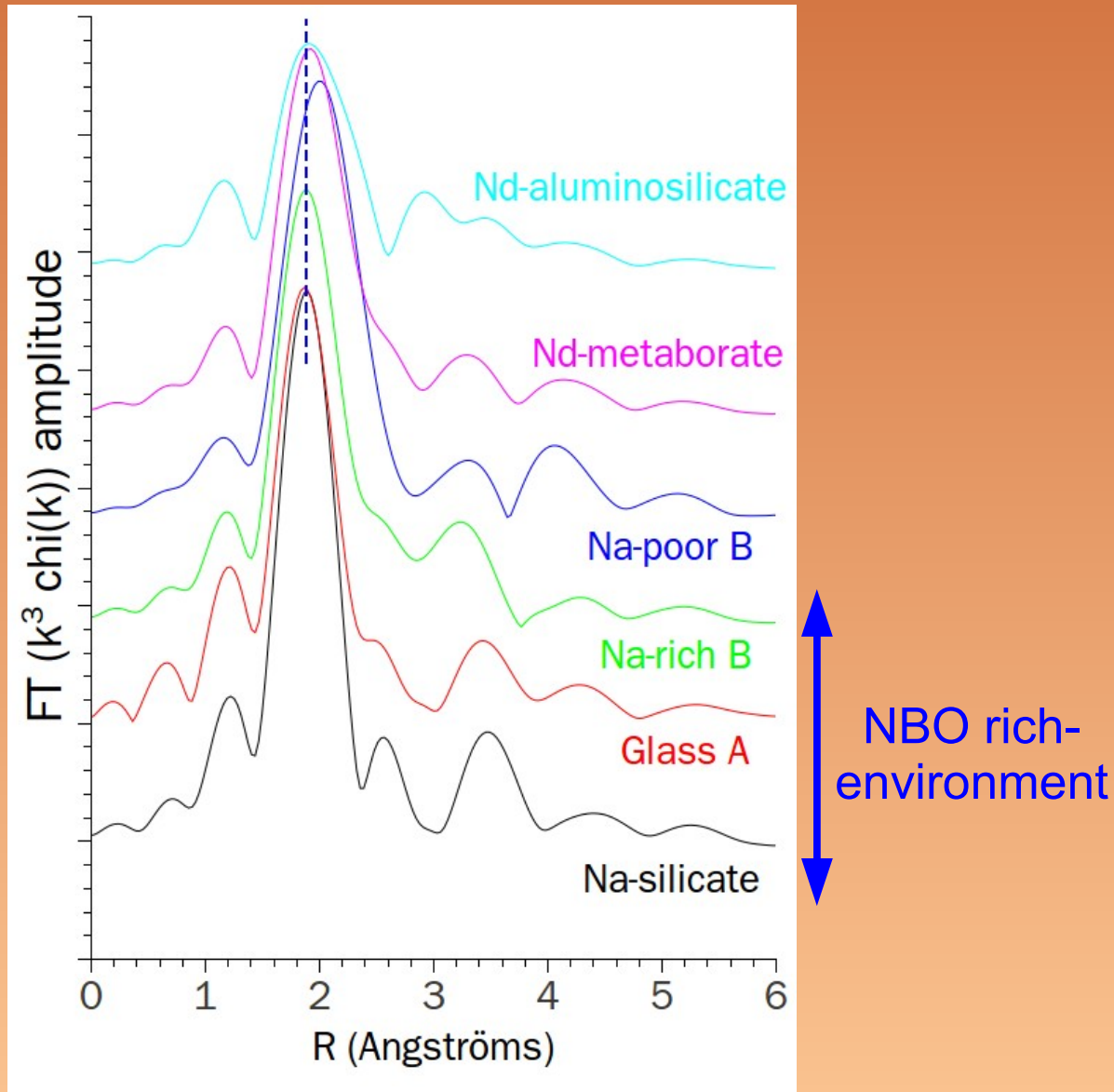


## Study by optical absorption spectroscopy at low temperature

(Gatterer et al. JNCS 231 (1998) 189; Dymnikov et al. JNCS 215 (1997) 83)



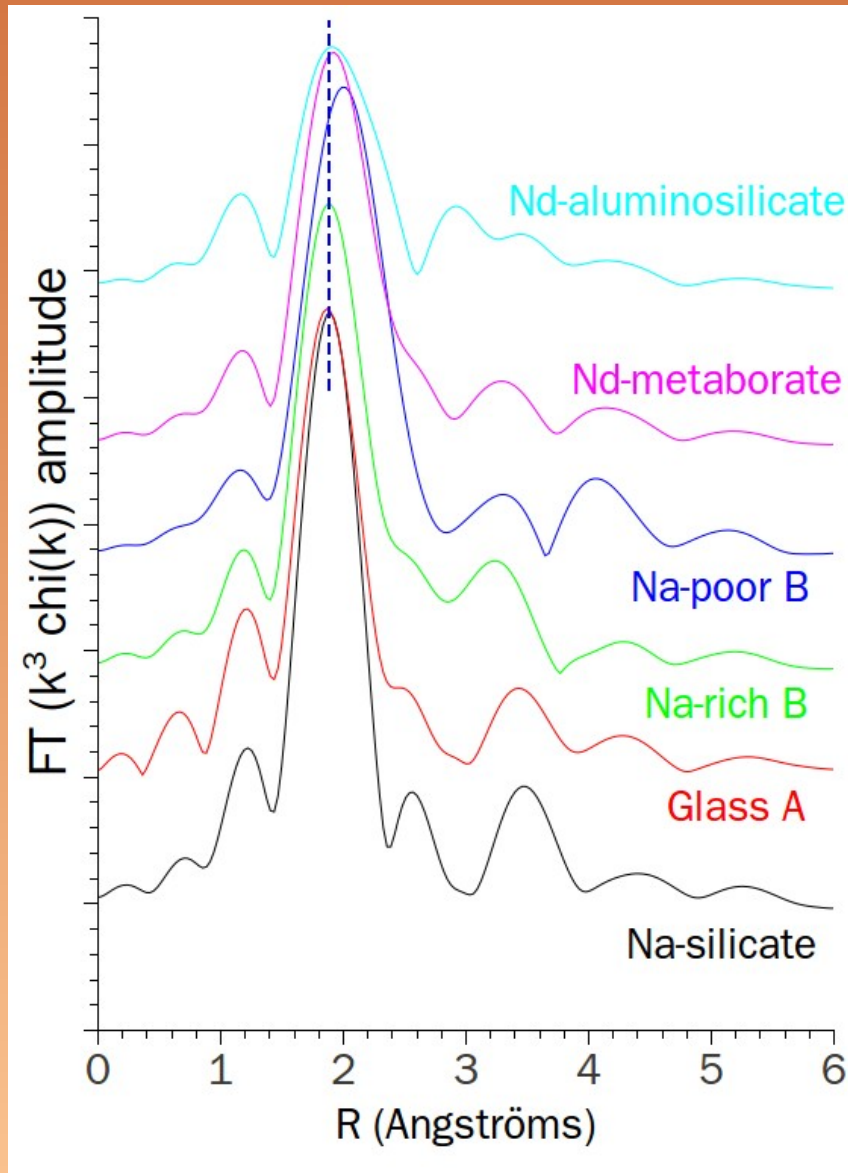
# Nd<sup>3+</sup> environment in the model glass



Nd  $L_3$ -edge EXAFS at 77 K



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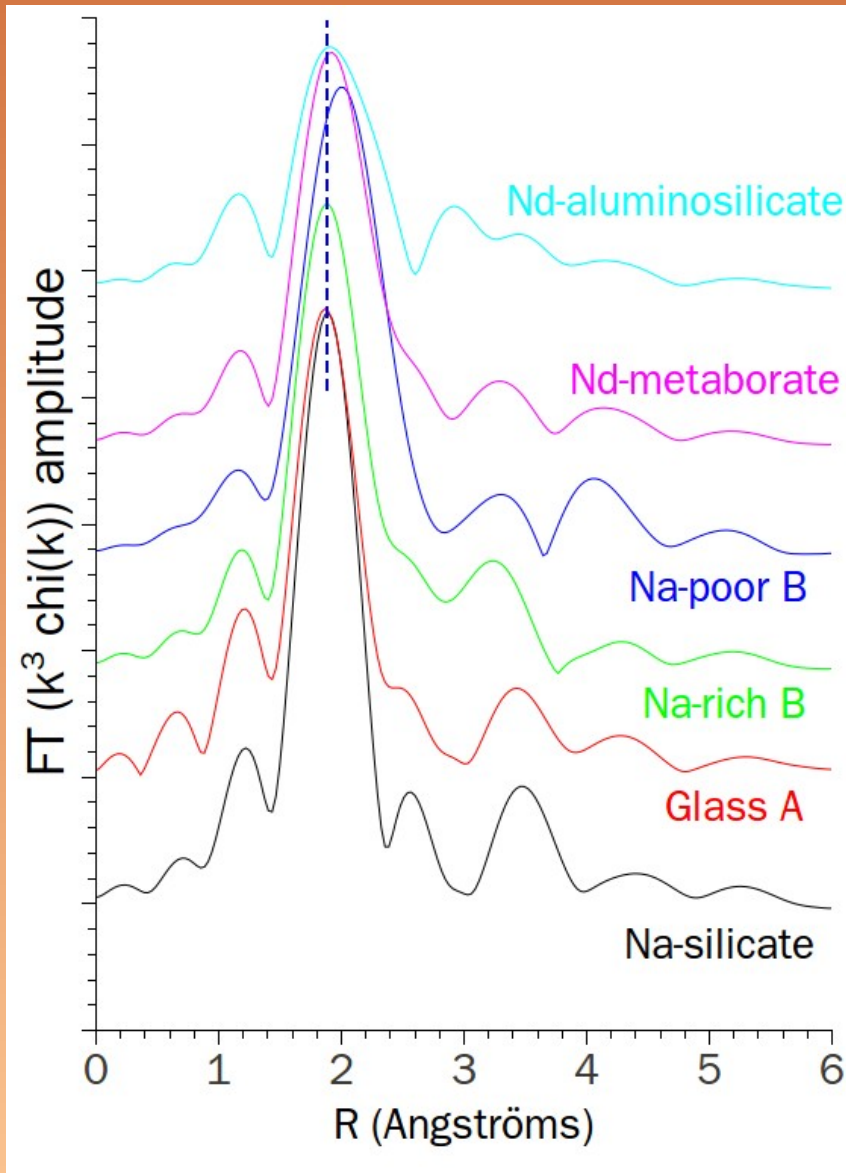


Nd L<sub>3</sub>-edge EXAFS at 77 K

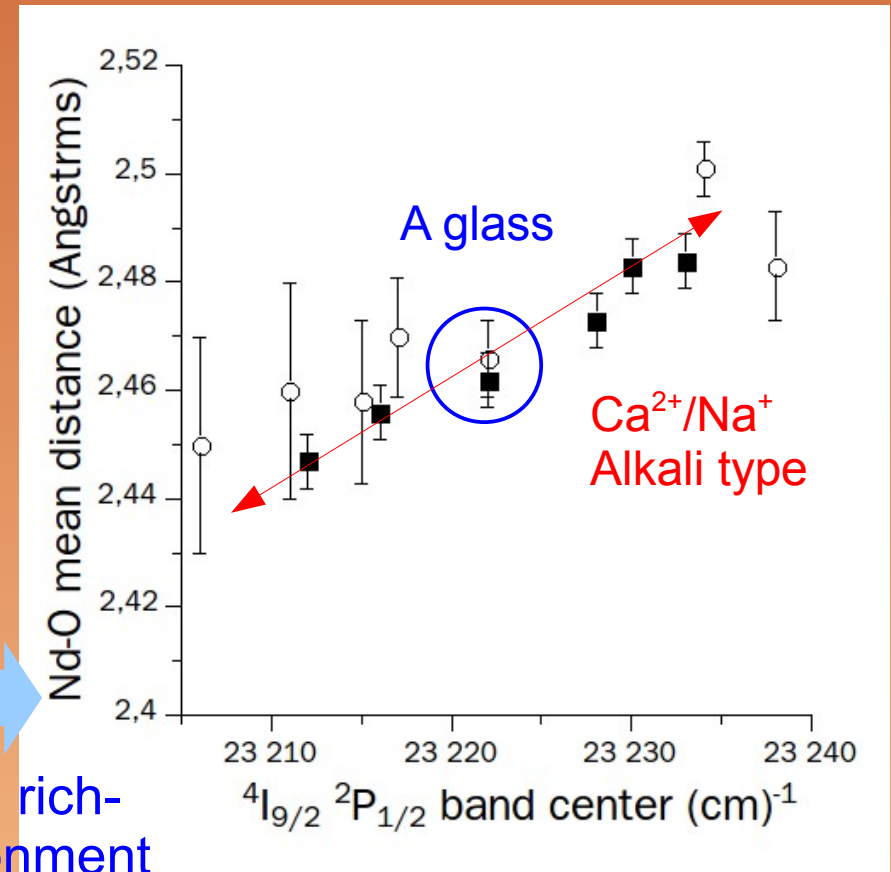
NBO rich-environment

- Well-defined coordination sphere
- 6 to 8 NBO's
- Mean dNd-O between 2.44 Å and 2.50 Å depending on Ca<sup>2+</sup>/Na<sup>+</sup> ratio or alkali type
- Consistency between dNd-O bond distance and nephelauxetic effect

# Nd<sup>3+</sup> environment in the model glass

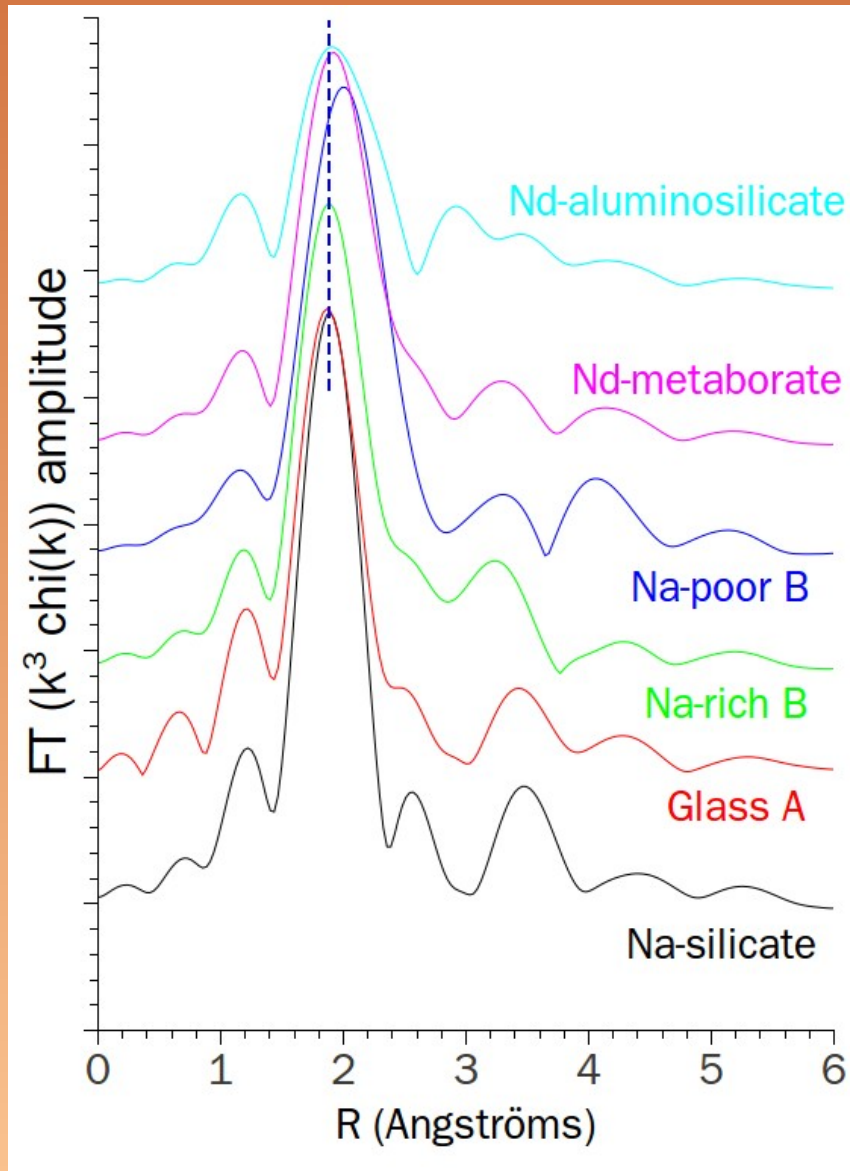


Nd L<sub>3</sub>-edge EXAFS at 77 K



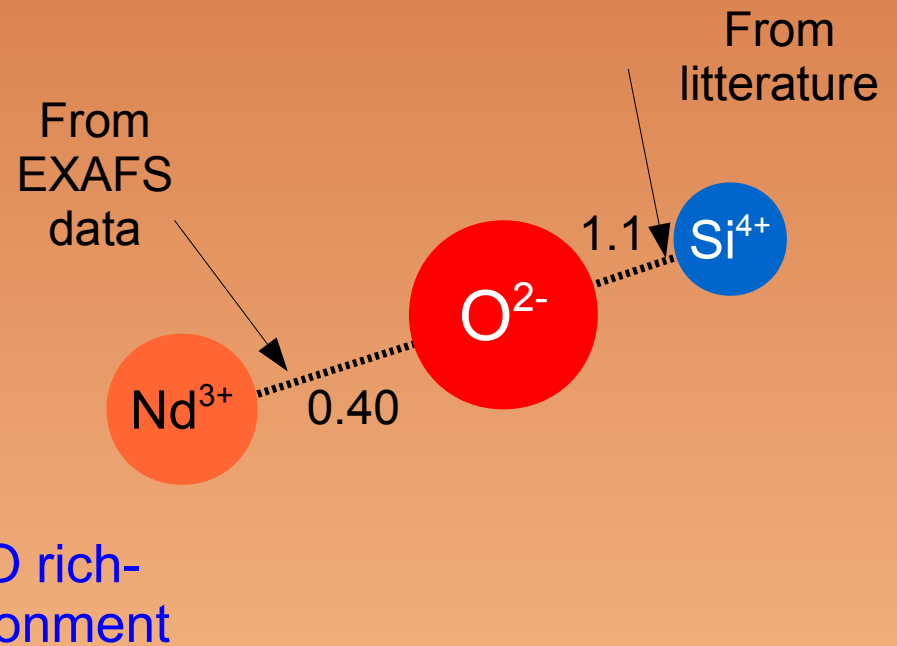
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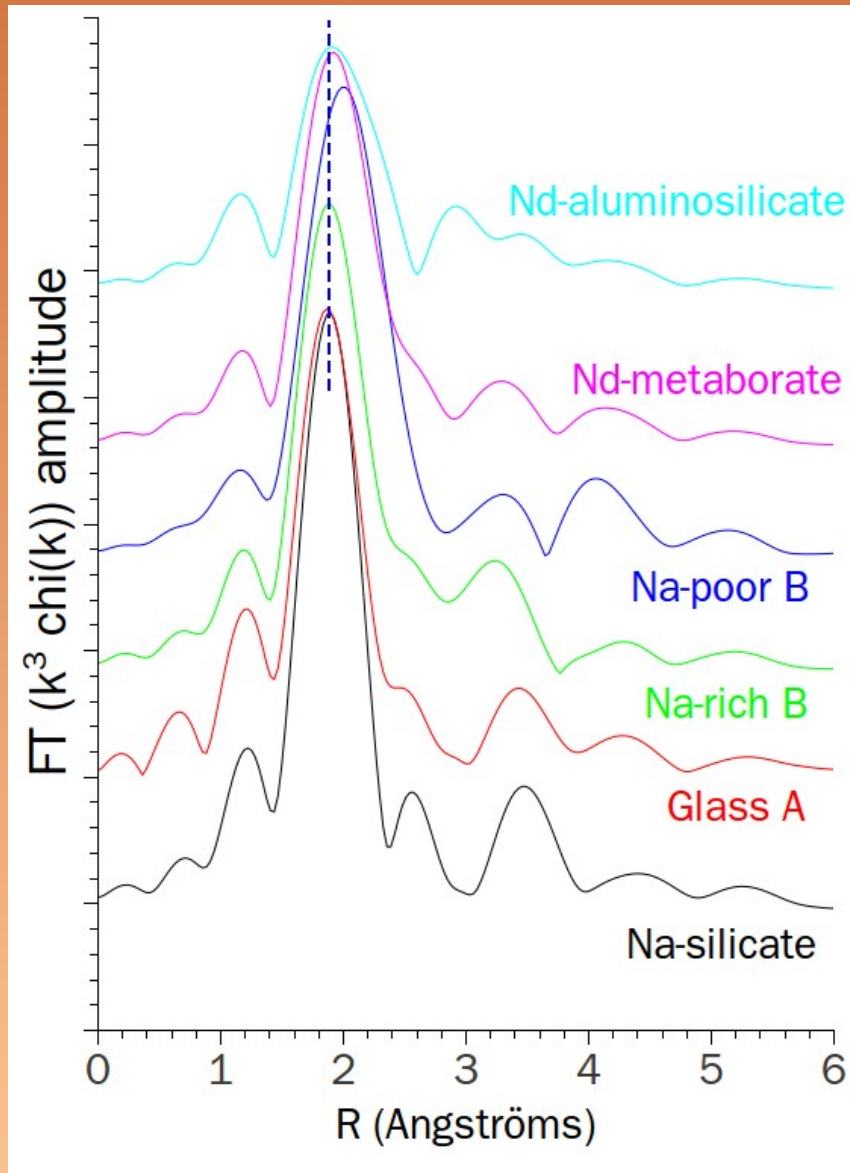


Nd L<sub>3</sub>-edge EXAFS at 77 K

Pauling rules and Bond Valence -  
Bond Length model of oxides  
(Brese & O'Keefe, Acta Cryst. B, 1991)

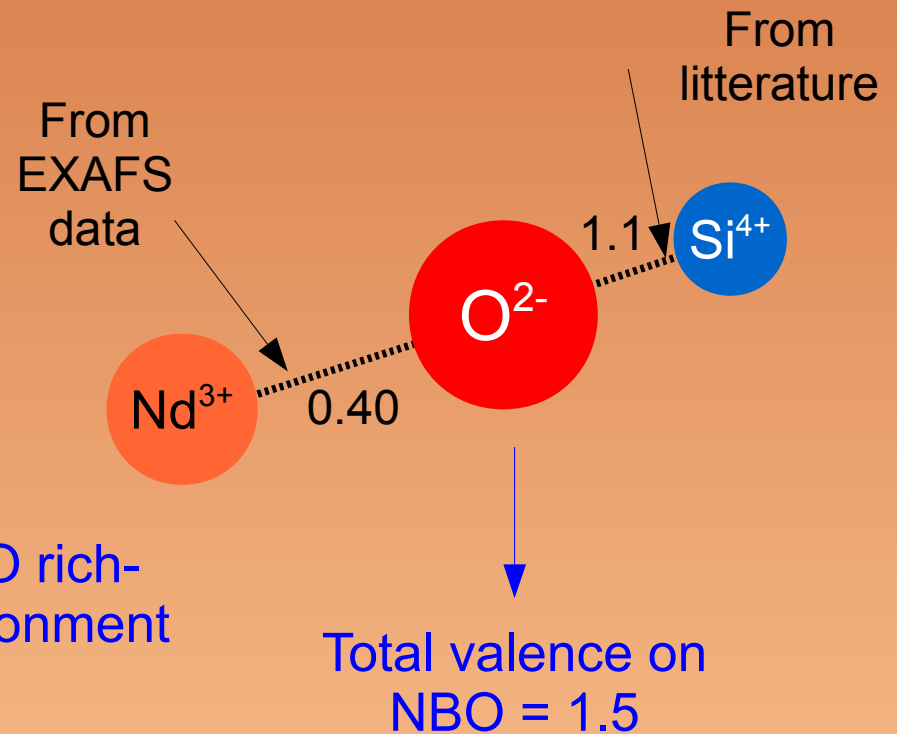


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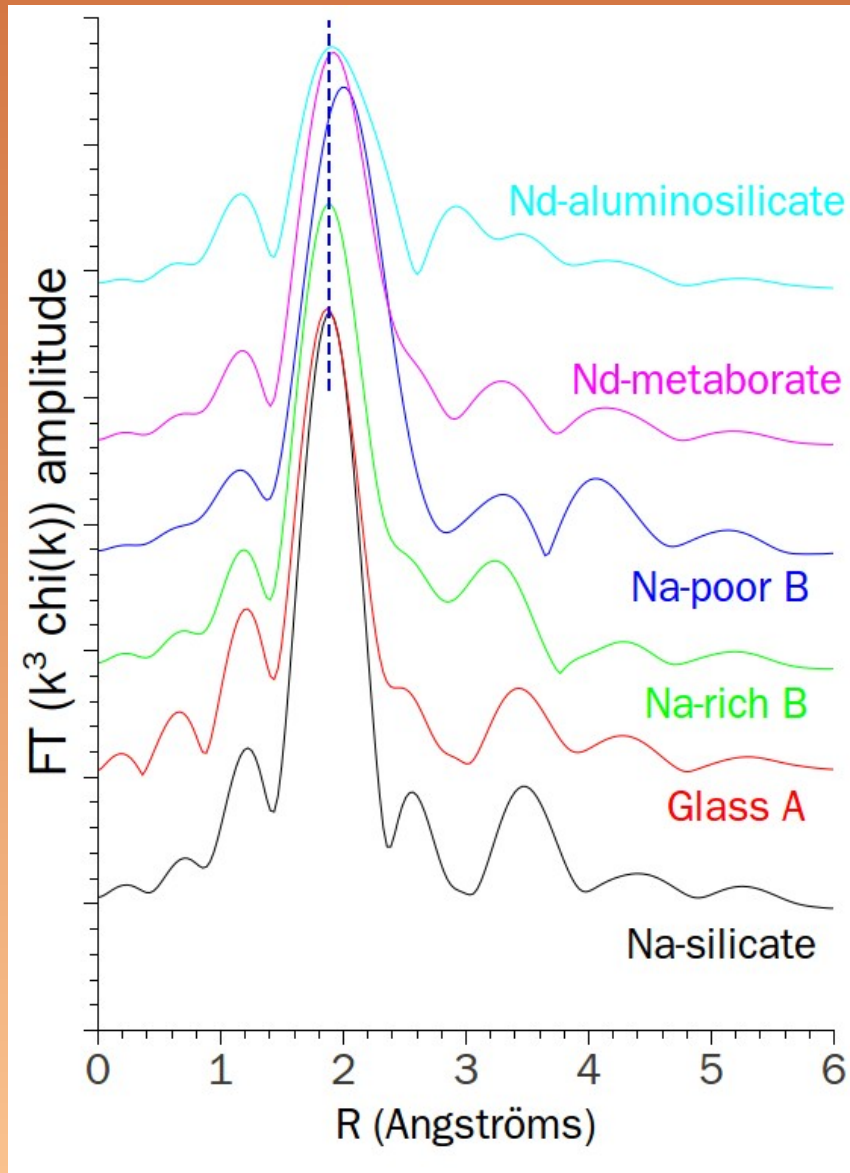


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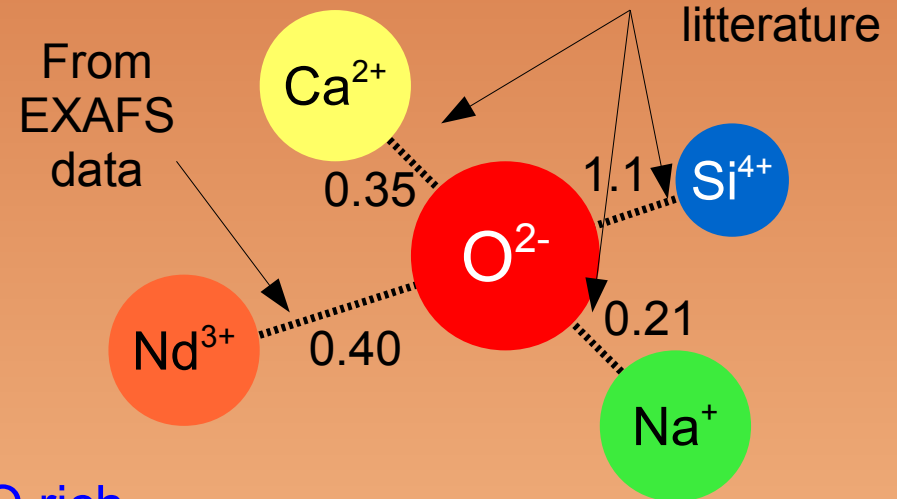
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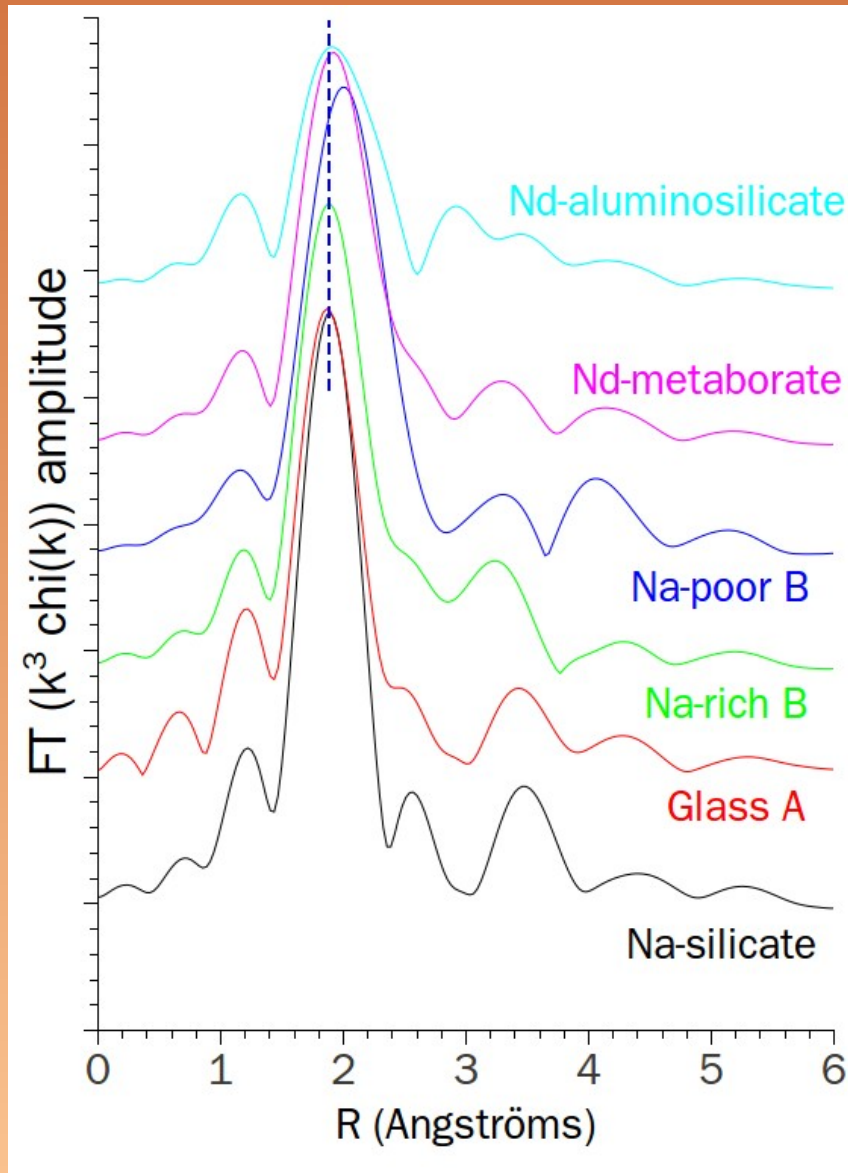
litterature



NBO rich-  
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# Nd<sup>3+</sup> environment in the model glass



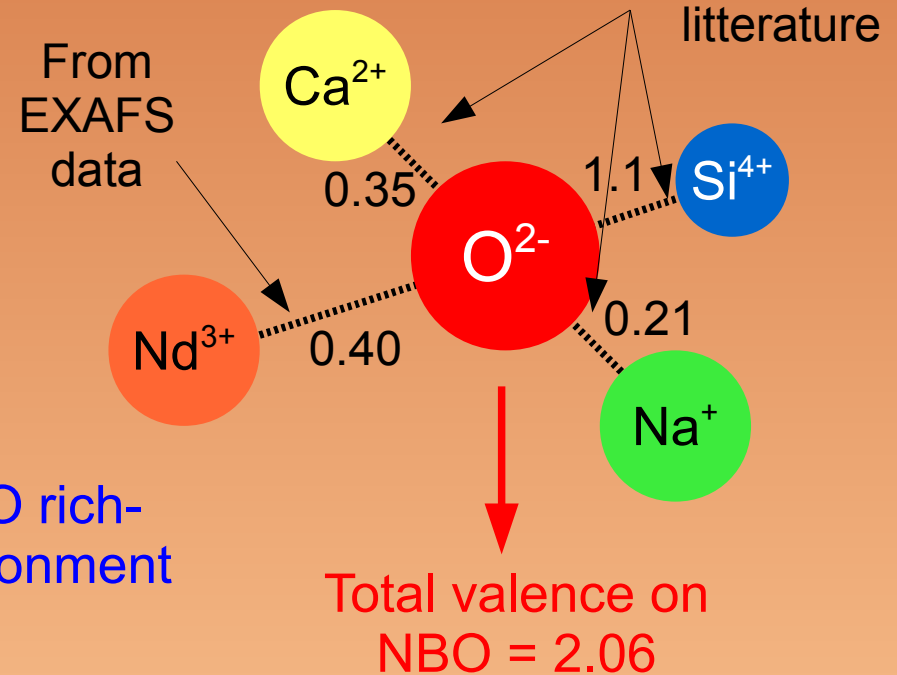
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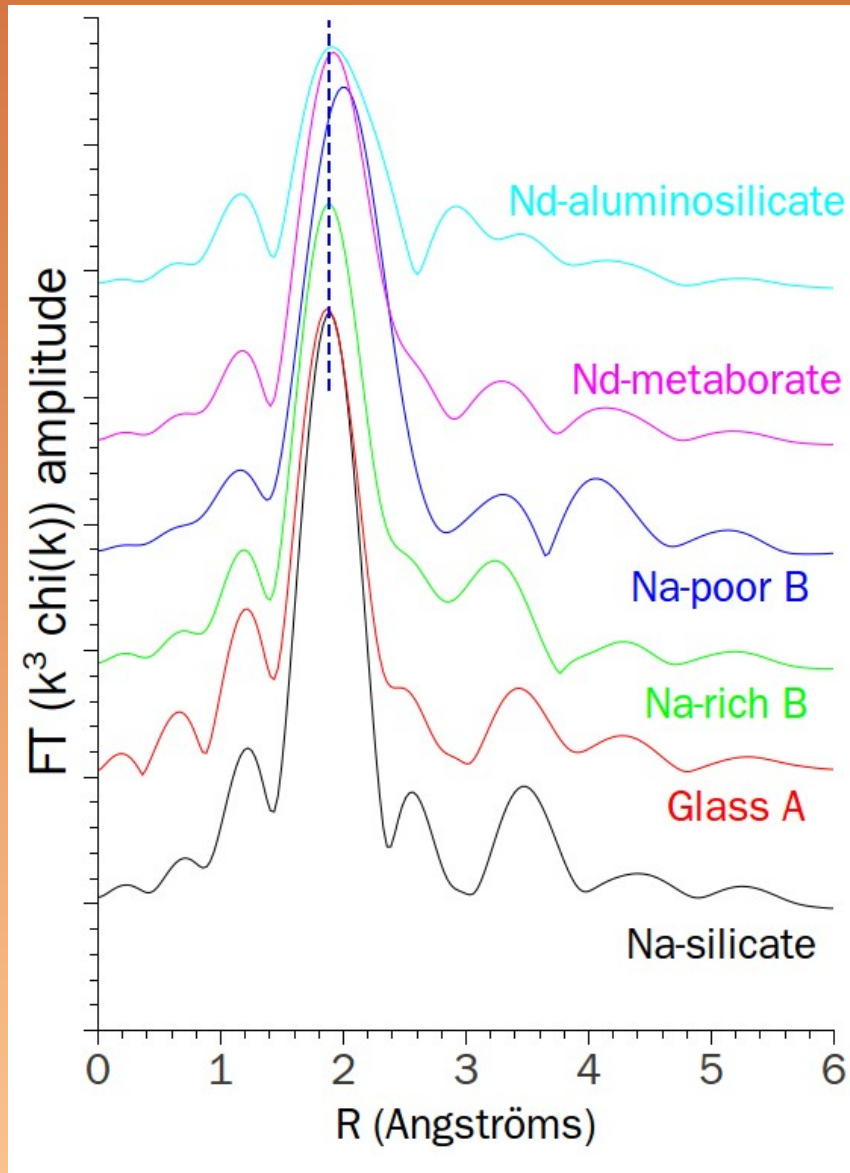
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From

litterature



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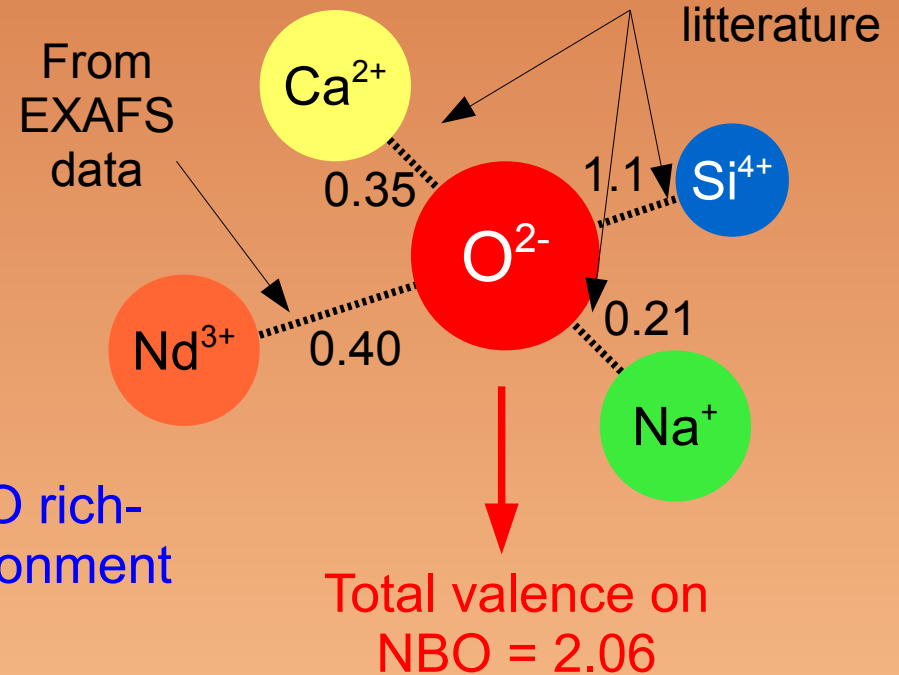
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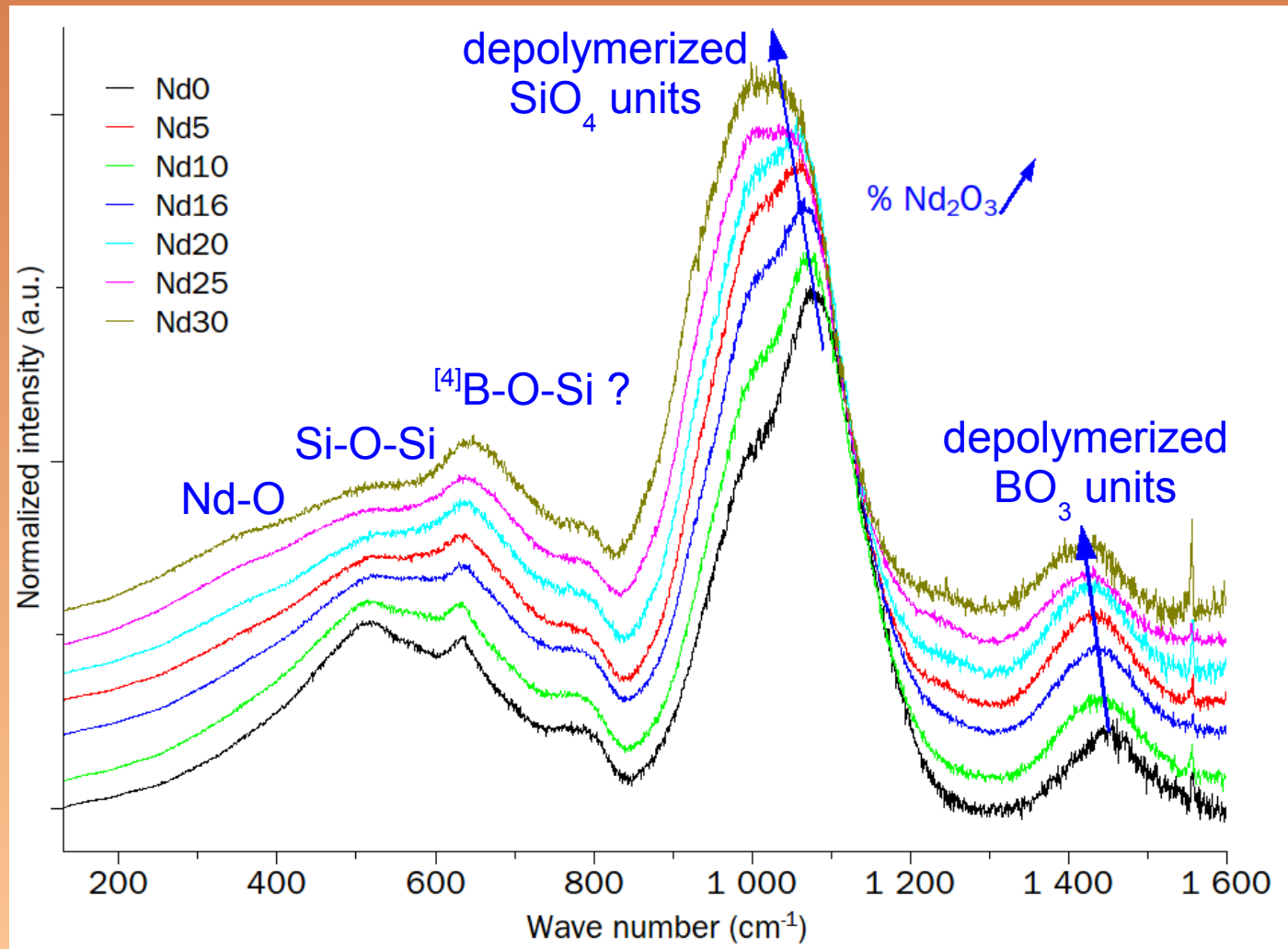
From

litterature



No evidence for Ca<sup>2+</sup> or Na<sup>+</sup>  
preferential charge compensation

# Effect of $\text{Nd}_2\text{O}_3$ content on the aluminoborosilicate network



With increasing %  
 $\text{Nd}_2\text{O}_3$

Low-frequency shift of

- Depolymerized  $\text{BO}_3$

band

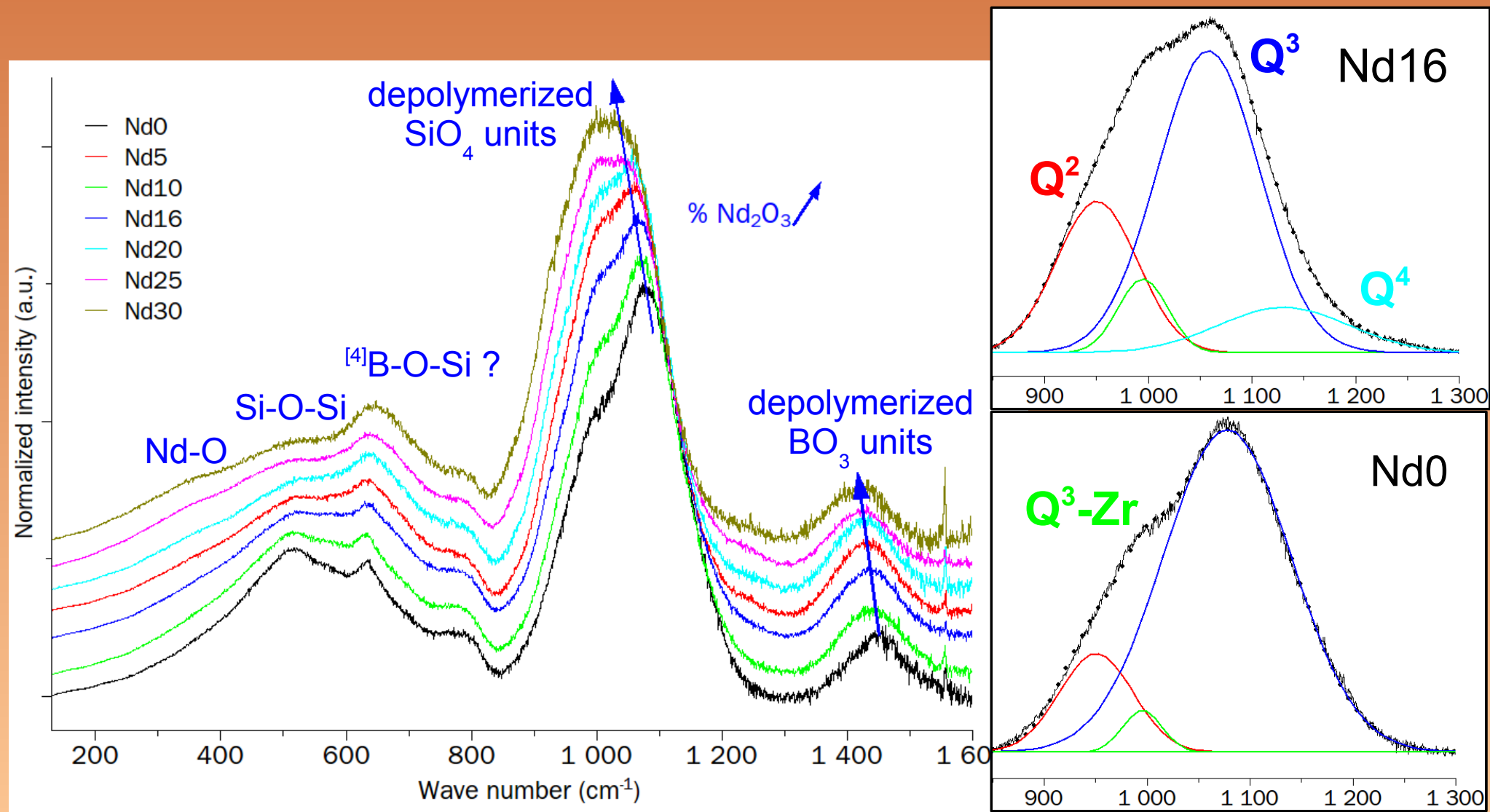
- $\text{Q}^3$  band

B-O-Si band is affected

Raman spectra ( $\lambda_{\text{exc}} = 488 \text{ nm}$ ) on glass A series with 0 to 30 wt%  $\text{Nd}_2\text{O}_3$

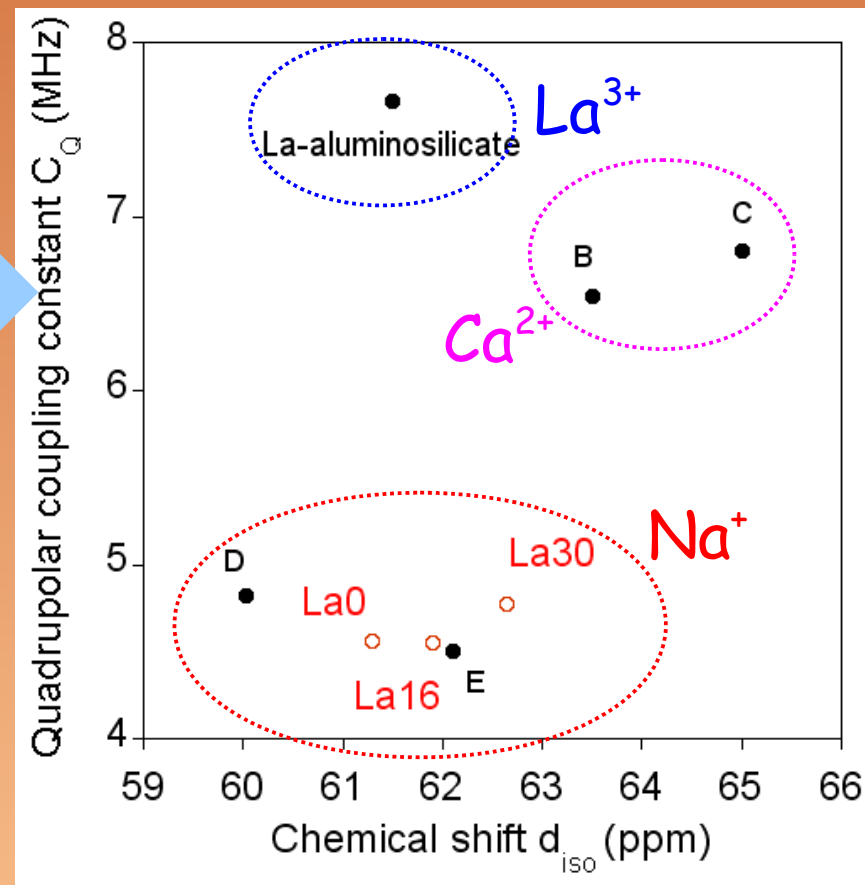
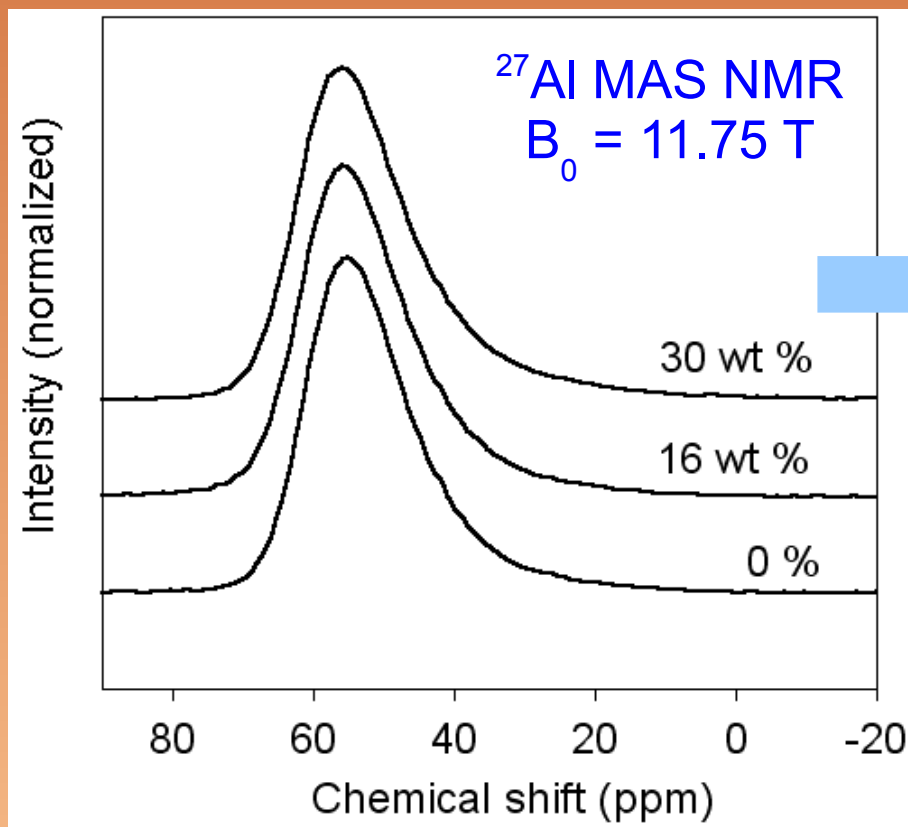


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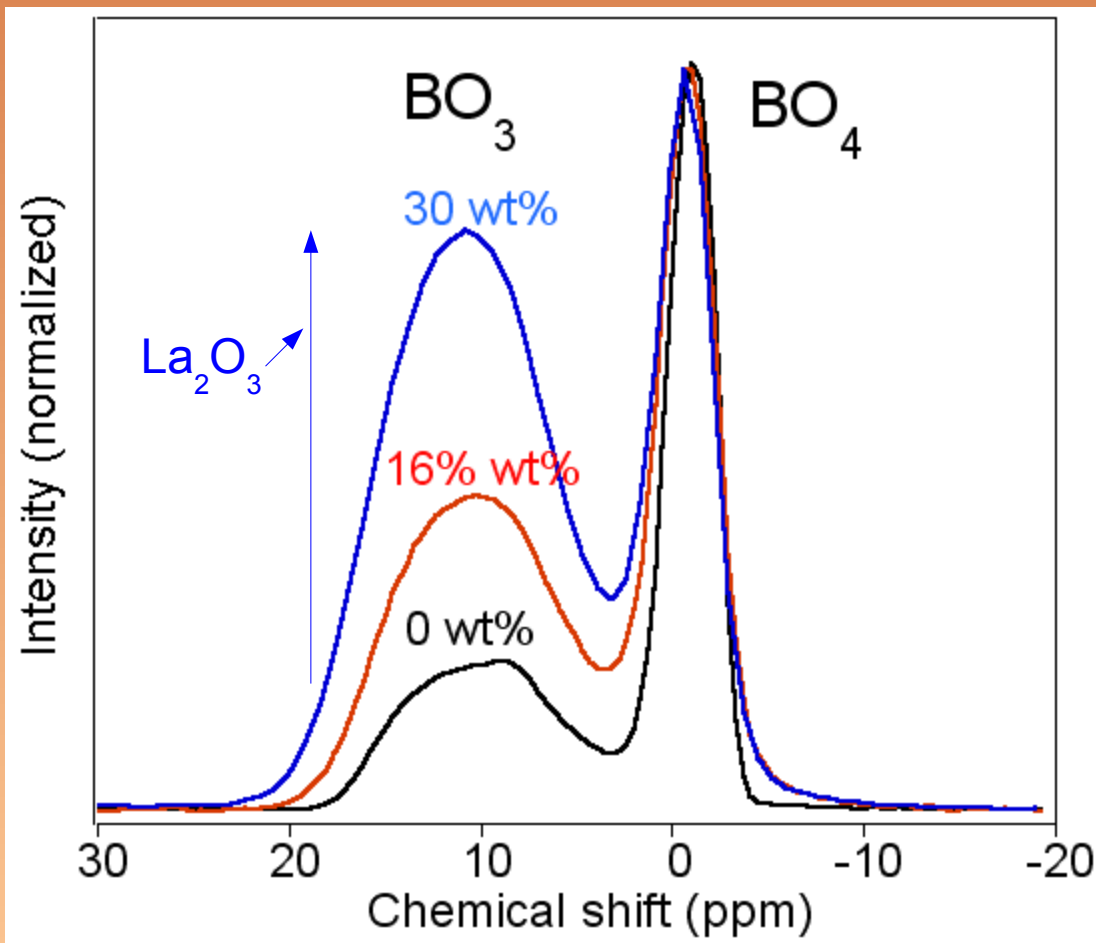
# Effect of $\text{Nd}_2\text{O}_3$ content on the aluminoborosilicate network



$\text{La}_2\text{O}_3$  incorporation has little effect on  $\text{AlO}_4$  units, that are charge compensated by  $\text{Na}^{+}$  ions

# Effect of $\text{Nd}_2\text{O}_3$ content on the aluminoborosilicate network

$^{11}\text{B}$  MAS NMR,  $B_0 = 11.75 \text{ T}$



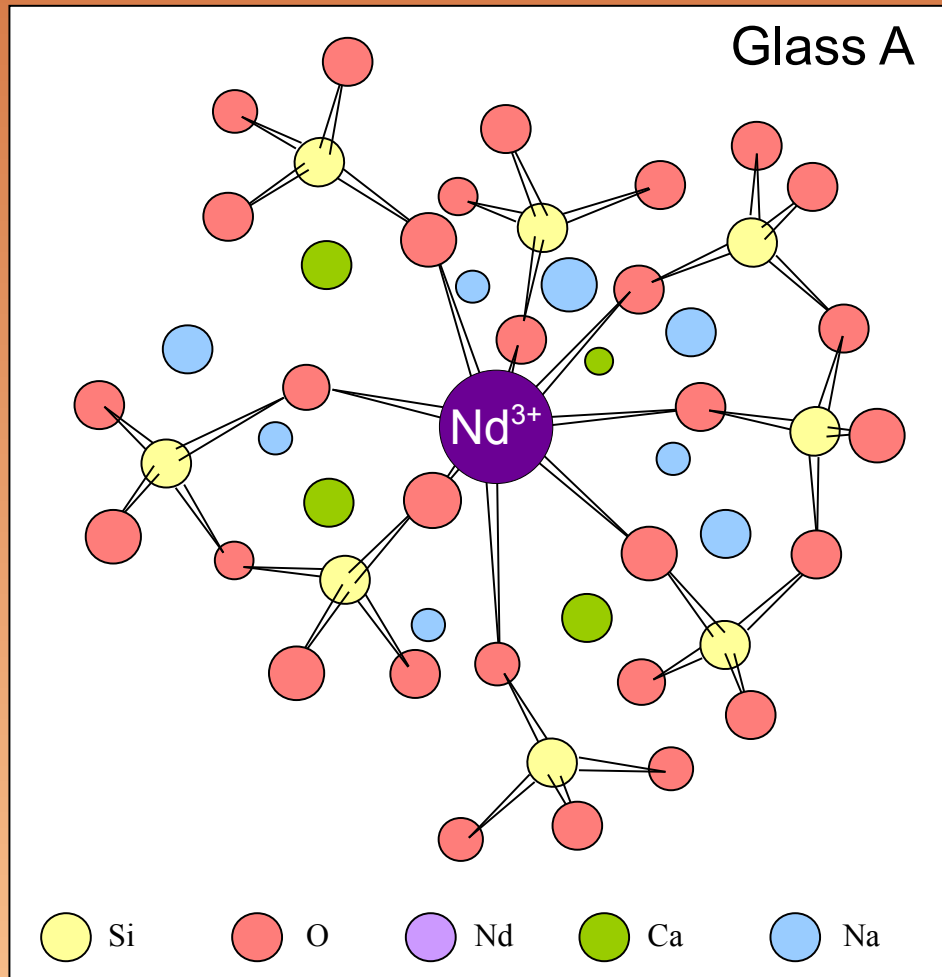
$\text{BO}_4$  fraction

60 %

42 %

30 %

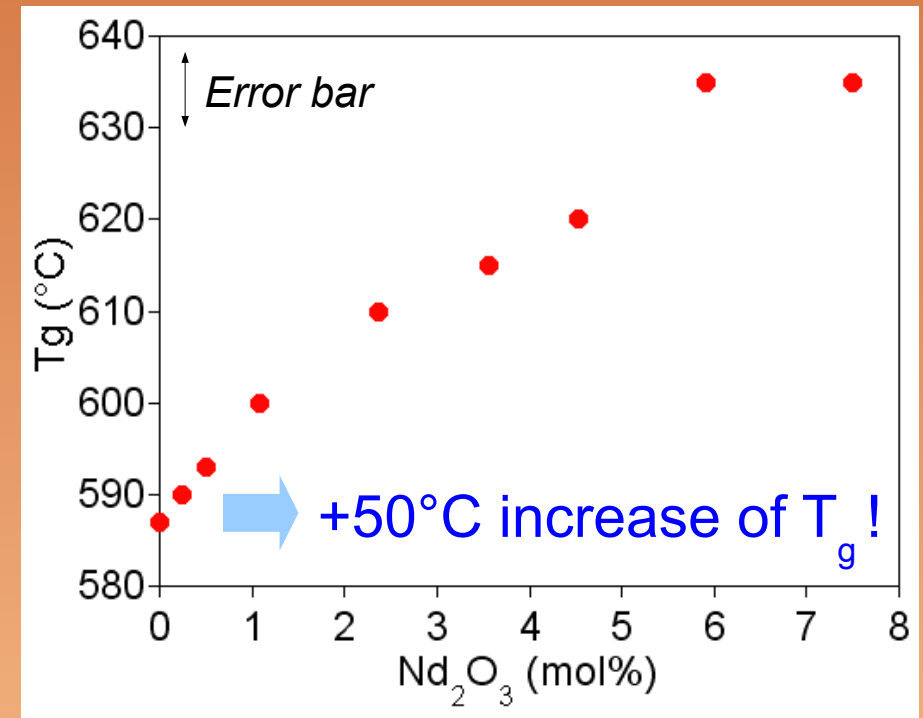
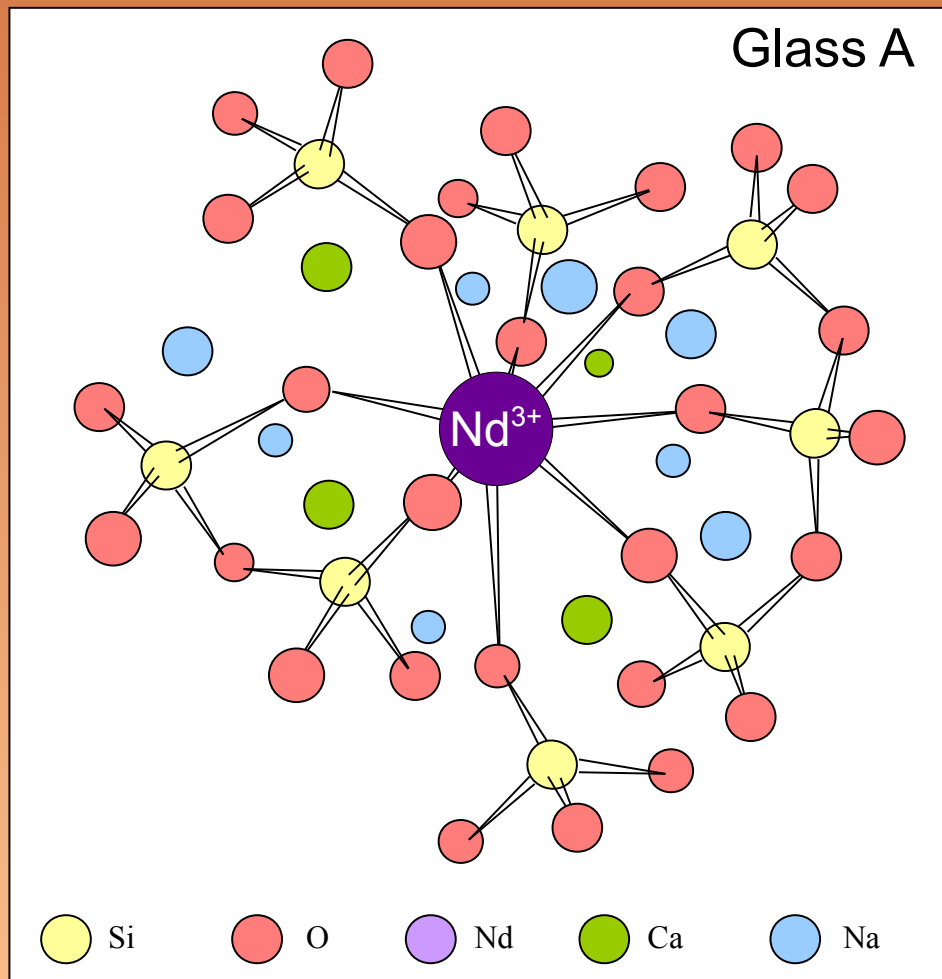
# Summary of $\text{Nd}_2\text{O}_3$ incorporation into the model glass



- Incorporation in depolymerized areas enriched in  $\text{NBO}/\text{Na}^+$  and  $\text{Ca}^{2+}$
- Depolymerization of the silicate network (creation of  $\text{NBO}$ ), principally by formation of  $\text{Q}^2$  units
- Destruction of  $\text{BO}_4$  units as  $\text{BO}_3 + \text{NBO}$
- No change of  $\text{AlO}_4$  units

Consistent with the formation of the  $\text{Nd}(\text{O})_7$  coordination sphere

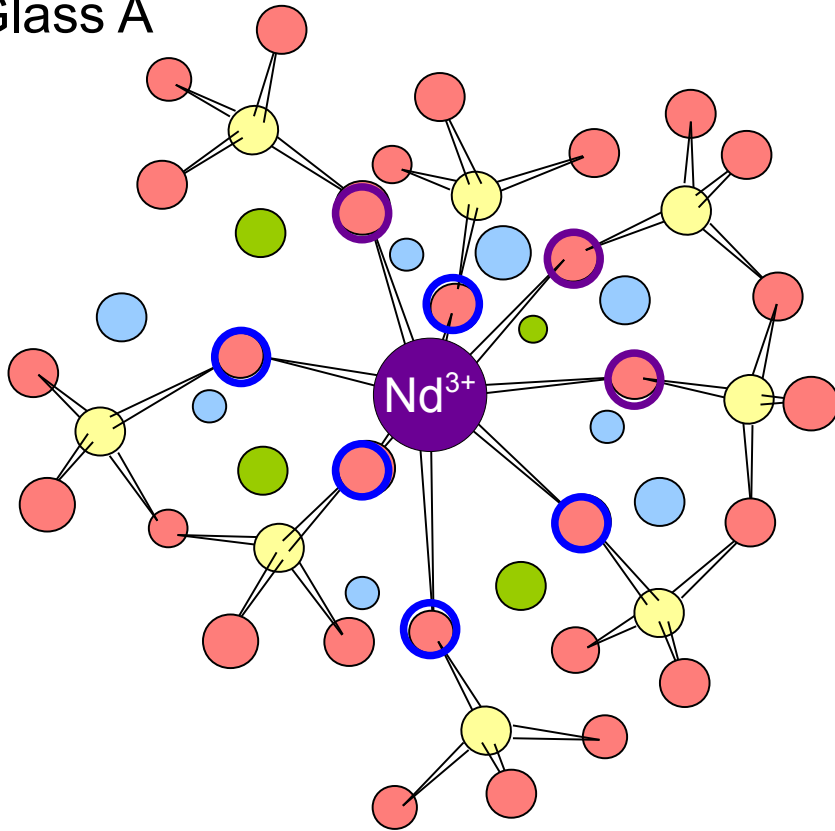
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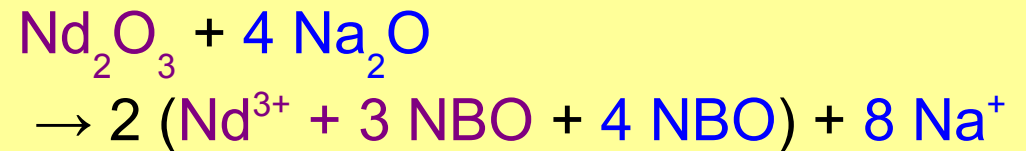
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# Summary of $\text{Nd}_2\text{O}_3$ incorporation into the model glass

Glass A

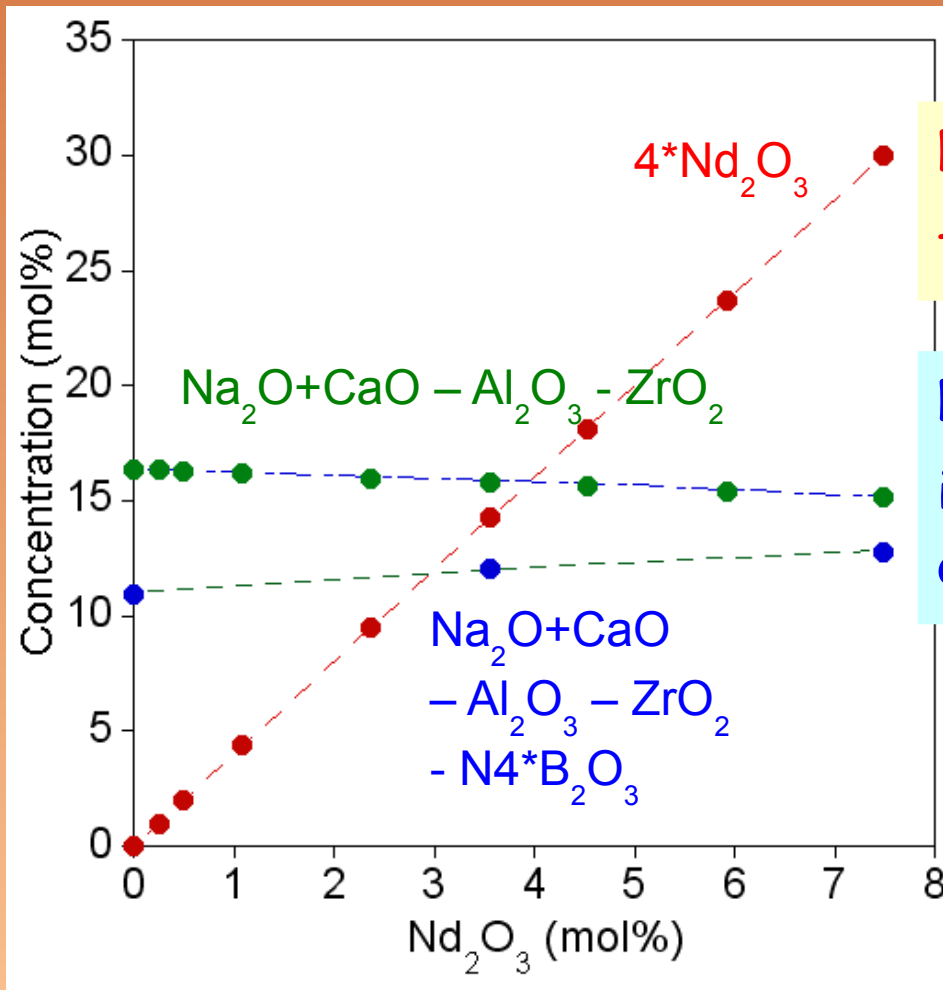


Si O Nd Ca Na



4 moles of  $\text{Na}_2\text{O}/\text{CaO}$  are  
necessary to incorporate 1  
mole  $\text{Nd}_2\text{O}_3$

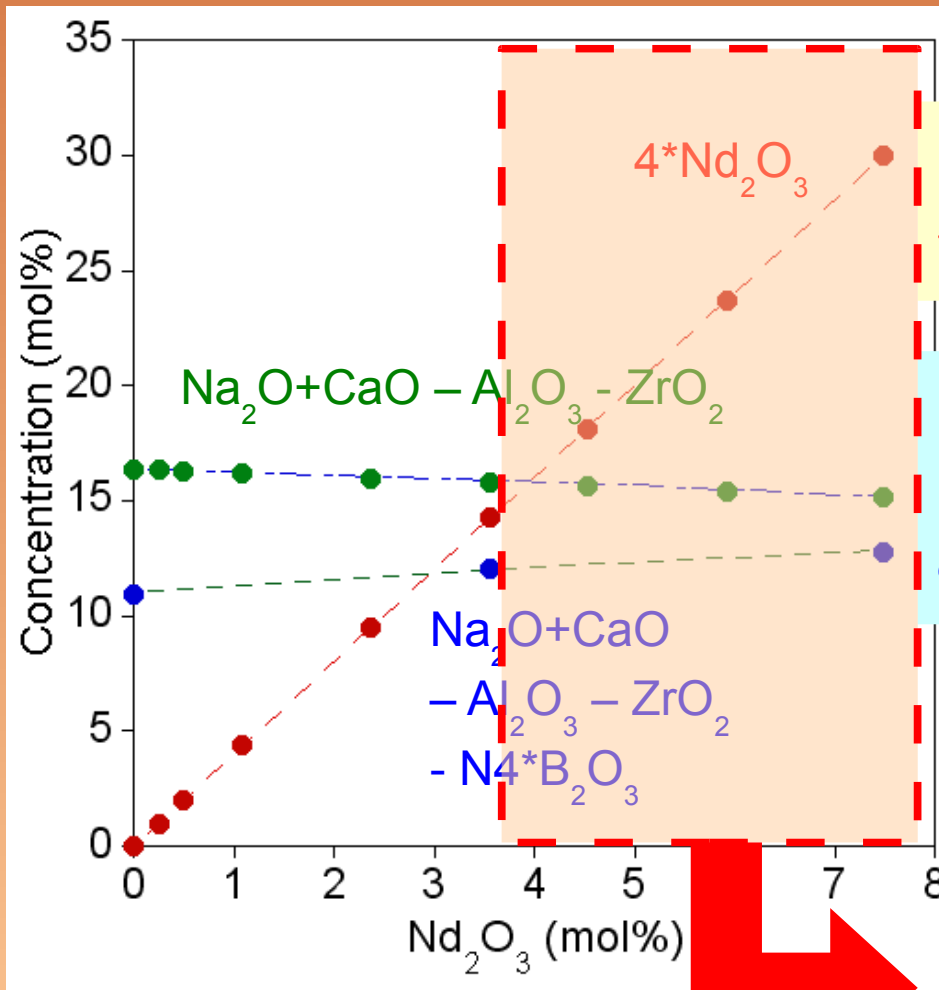
# Availability of charge compensating cations as a function of $\text{Nd}_2\text{O}_3$ content



Number of  $\text{Na}_2\text{O} + \text{CaO}$  necessary to form the  $\text{Nd}(\text{O})_7$  coordination sphere

Number of  $\text{Na}_2\text{O} + \text{CaO}$  available taking into account the charge compensation of  $(\text{AlO}_4)^-$ ,  $(\text{ZrO}_6)^{2-}$  and  $(\text{BO}_4)^-$  units

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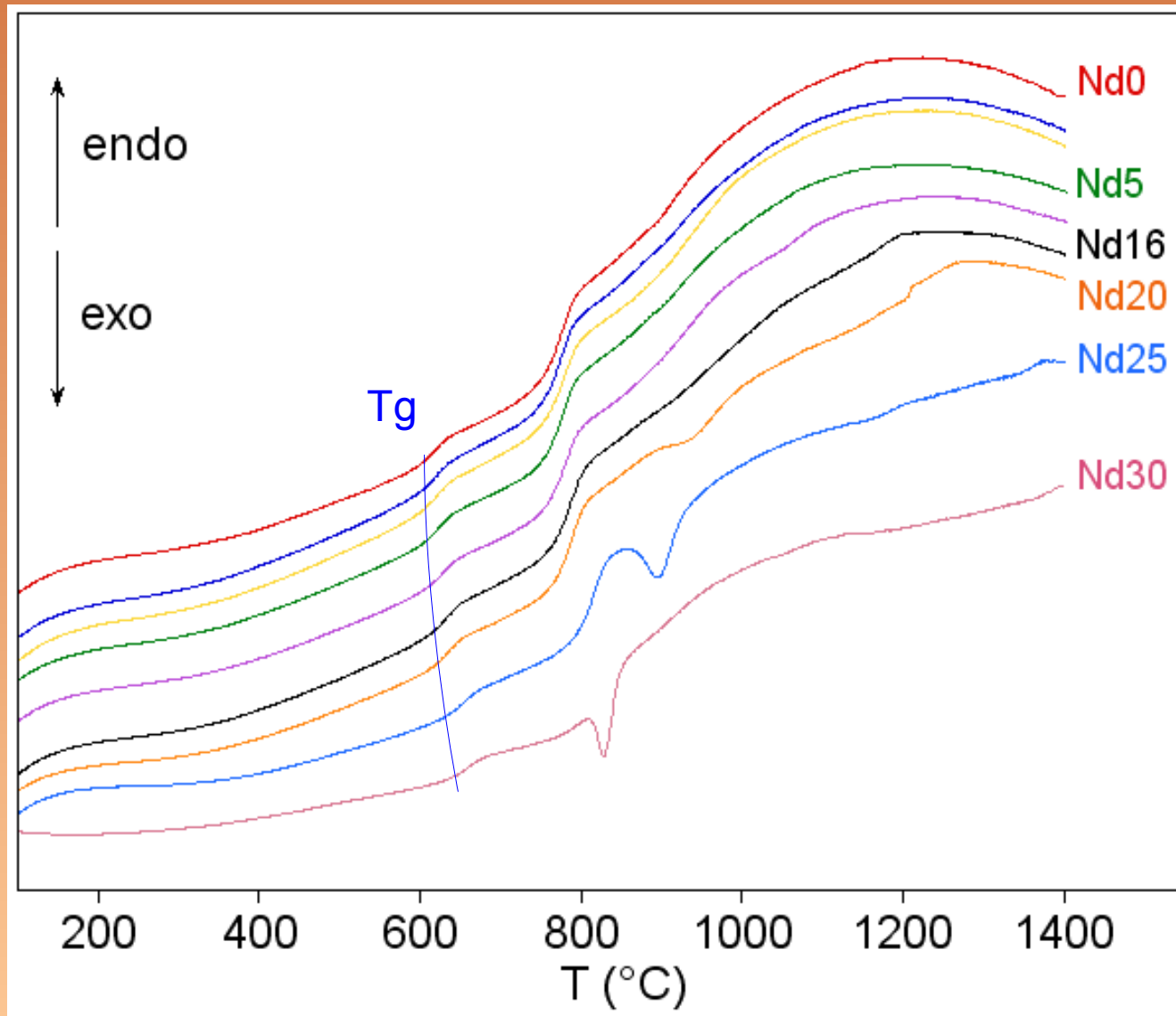
Number of  $\text{Na}_2\text{O} + \text{CaO}$  available taking into account the charge compensation of  $(\text{AlO}_4)^-$ ,  $(\text{ZrO}_6)^{2-}$  and  $(\text{BO}_4)^-$  units

On this range  $\text{Nd}^{3+}$  ions can not find enough NBO to incorporate as isolated  $\text{Nd}(\text{O})_7$



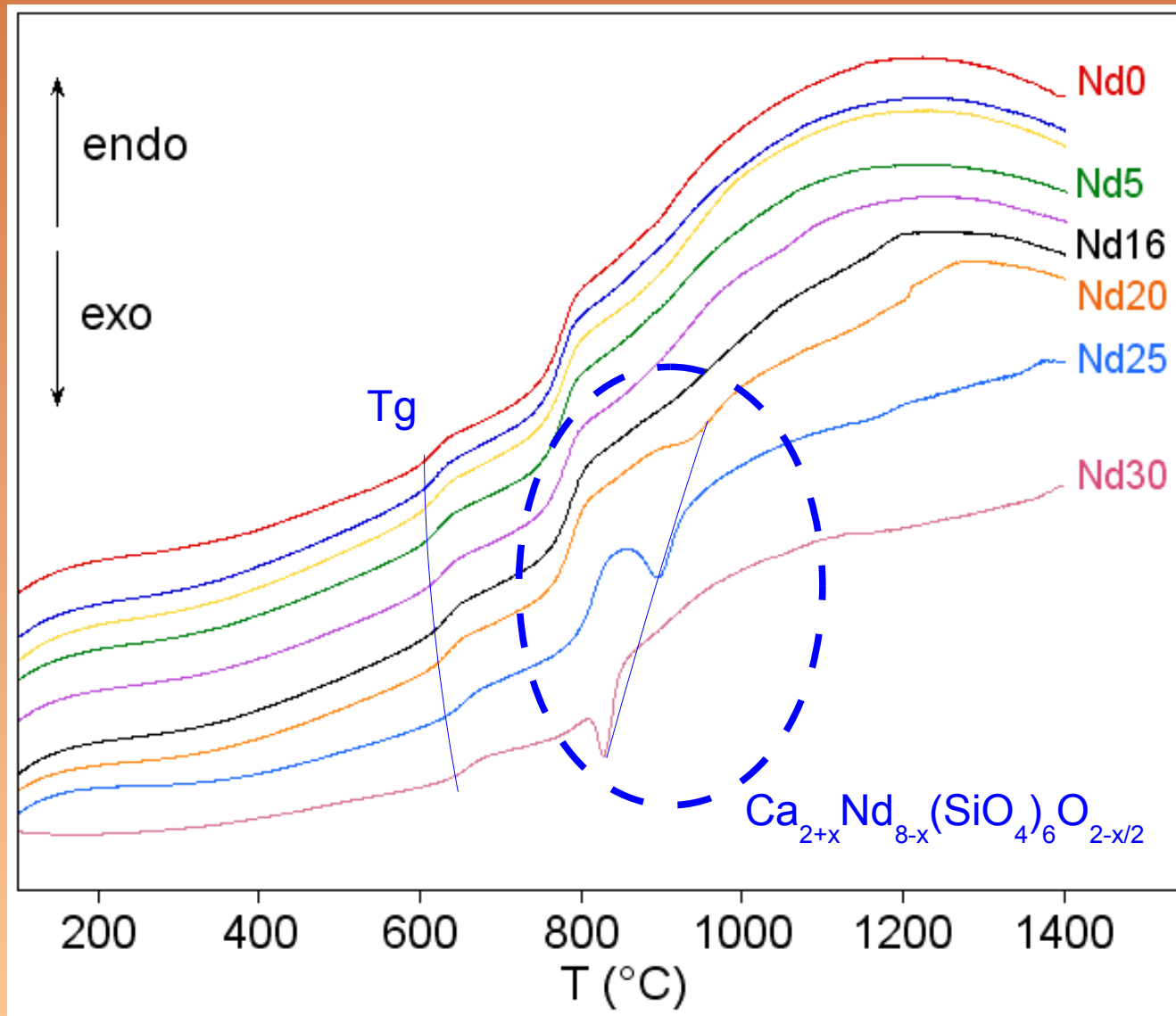
# Crystallization study

DTA at 10°C/min of glass powder, 80-125  $\mu\text{m}$



# Crystallization study

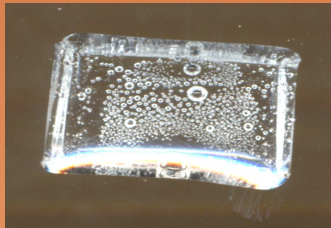
DTA at 10°C/min of glass powder, 80-125 µm



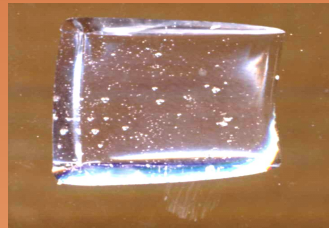
Crystallization of a  
Nd-silicate apatite  
beyond 4 mol% - 16  
wt% Nd<sub>2</sub>O<sub>3</sub>

# Crystallization study

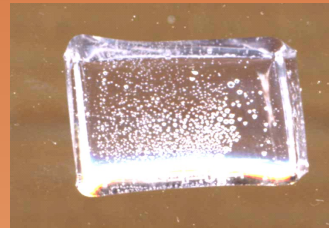
Slow cooling from the melt at 6°C/min



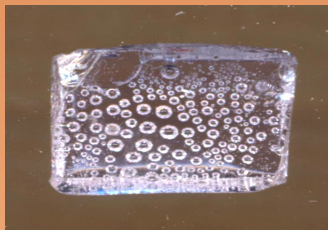
Nd0



Nd1,3



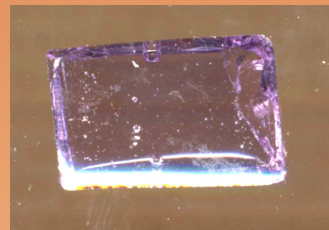
Nd2,5



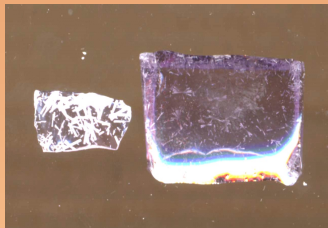
Nd5



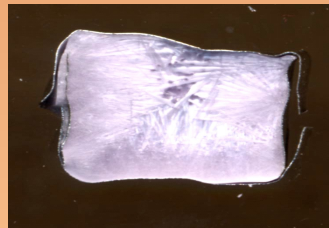
Nd10



Nd16



Nd20



Nd25



Nd30

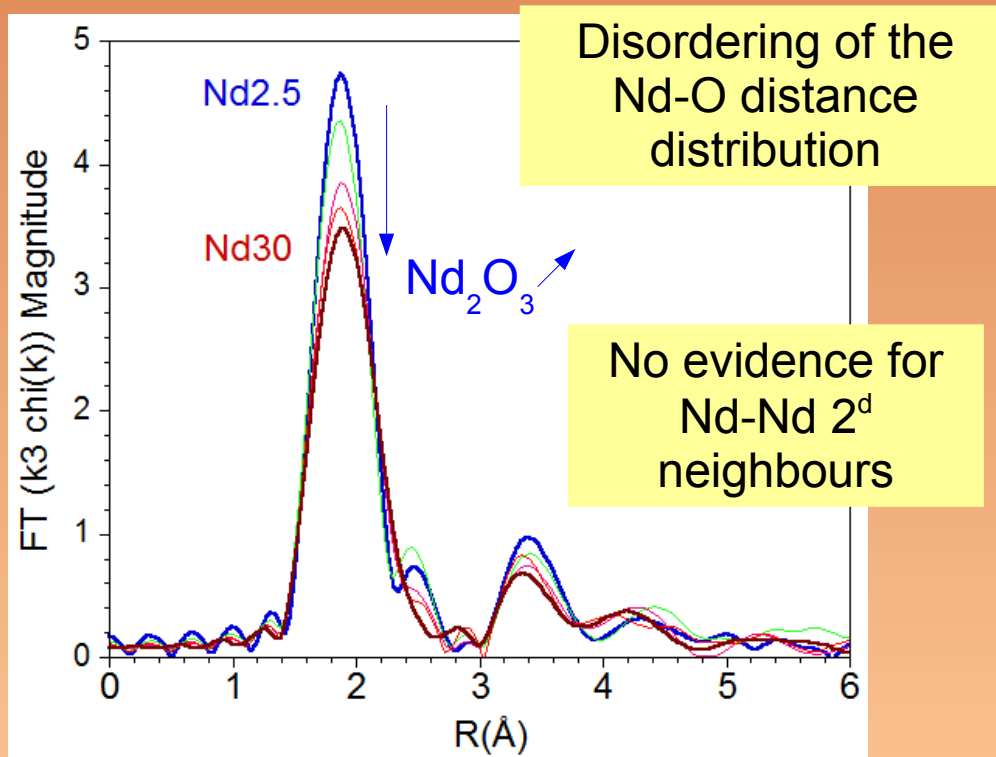
Crystallization of a  
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# Summary and composition changes

Beyond 4 mol% (16 wt%)  $\text{Nd}_2\text{O}_3$ ,  $\text{Nd}^{3+}$  ions :

- Are bonded to NBO's and to BO's
- And/or share their NBO's through clustering in the glass structure

No conclusive evidence from EXAFS and optical absorption spectra.  $\text{Nd}^{3+}$  luminescence decay study is under way.



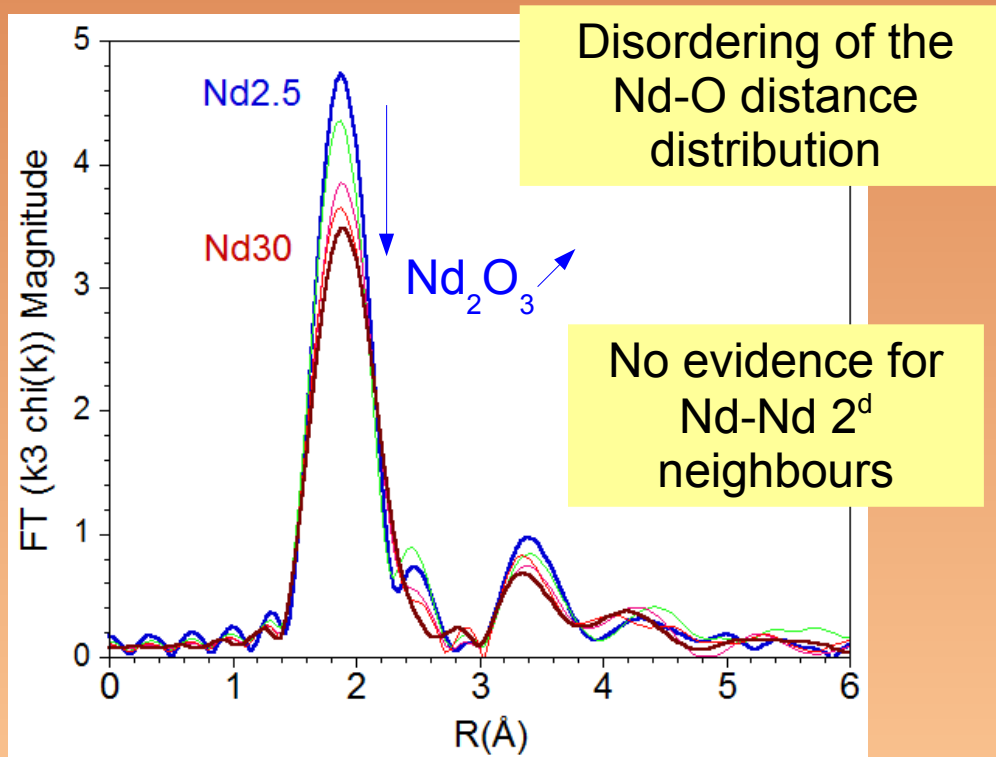
$\text{Nd L}_3$ -edge EXAFS at 77 K

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Nd L<sub>3</sub>-edge EXAFS at 77 K

**Relationship between this composition threshold and the crystallization of Nd-silicate apatite**

What happens when CaO and  $\text{Na}_2\text{O}$  are lacking

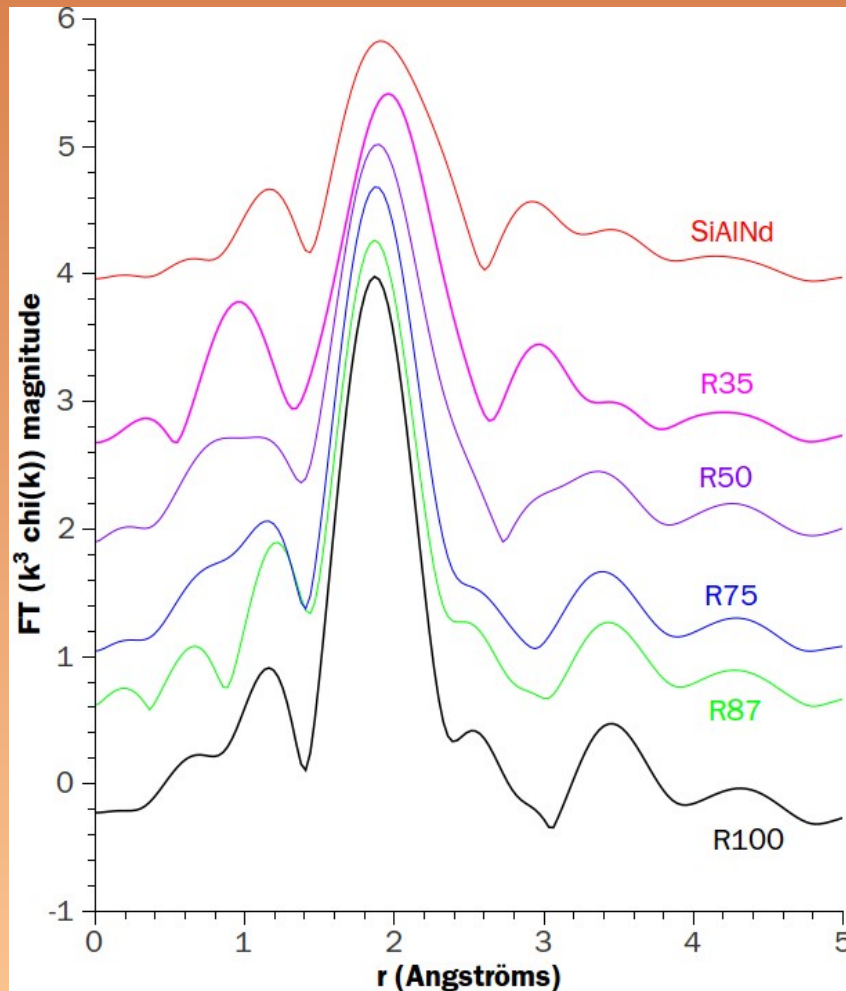
- In  $\text{Al}_2\text{O}_3$  enriched glass ?
- In  $\text{B}_2\text{O}_3$  enriched glass ?

O. Majérus et al., JNCS. 357 (2011) 2744-2751

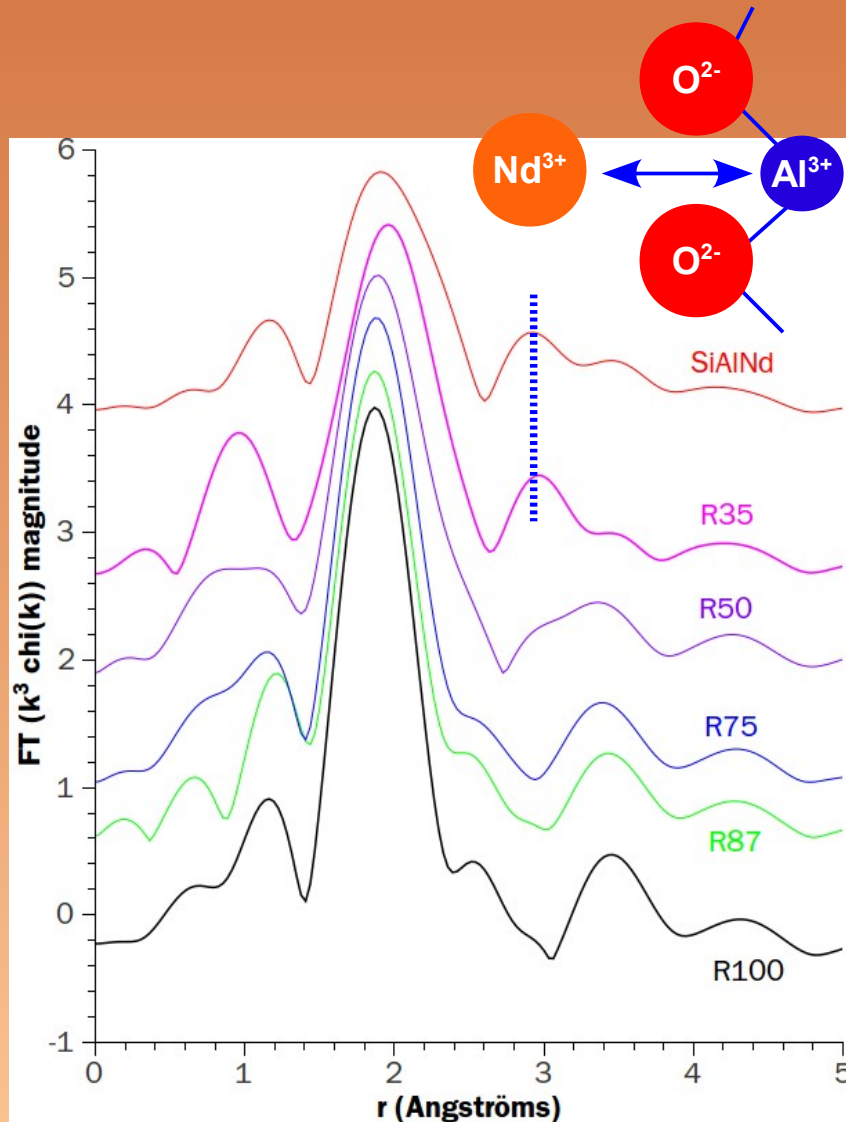
# Nd<sup>3+</sup> in peraluminous glasses

$$R = \frac{Na_2O + CaO}{Na_2O + CaO + Al_2O_3}$$

A glasses with R varying from 1 to 0.35  
(15.4 mol% Al<sub>2</sub>O<sub>3</sub>)



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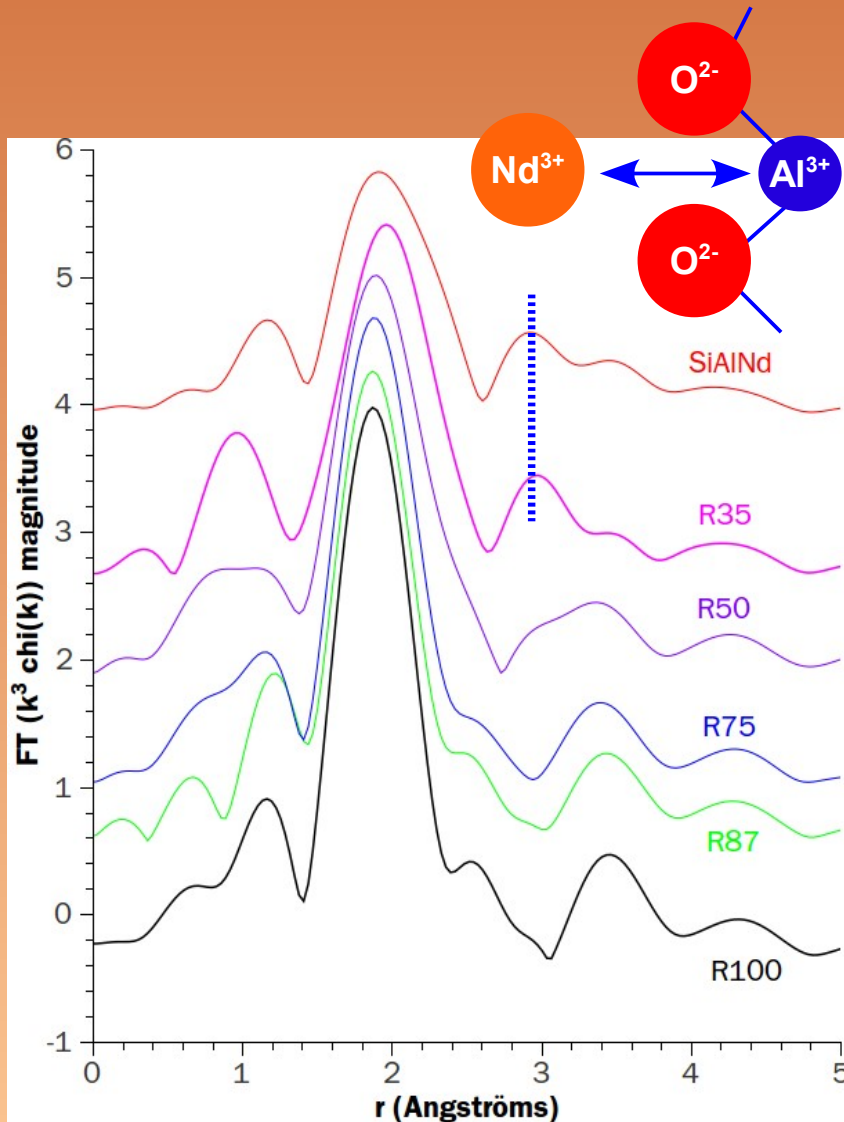


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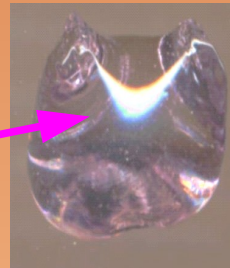


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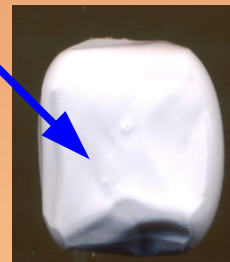


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When  $R < 0.5$   
Nd<sup>3+</sup> become charge  
compensators of (AlO<sub>4</sub>)<sup>-</sup> units  
Confirmed by <sup>27</sup>Al MAS NMR



Nd<sup>3+</sup> solubilization in the glass

*Slow cooling from  
the melt at  
6°C/min*



# Conclusions

Model **glass A** (mol%) :

<b>SiO<sub>2</sub></b>	<b>-</b>	<b>Al<sub>2</sub>O<sub>3</sub></b>	<b>-</b>	<b>B<sub>2</sub>O<sub>3</sub></b>	<b>-</b>	<b>Na<sub>2</sub>O</b>	<b>-</b>	<b>CaO</b>	<b>-</b>	<b>ZrO<sub>2</sub></b>	<b>-</b>	<b>Nd<sub>2</sub>O<sub>3</sub></b>
61.8		3.0		9.0		14.4		6.3		1.9		3.6

- A **well-defined 6-8 O coordination sphere** is inferred from the Nd<sup>3+</sup> spectroscopic data (optical spectroscopy and EXAFS)
- **Compensation of the excess negative charge of the Nd(O)<sub>7</sub> complex is provided by Na<sup>+</sup> and Ca<sup>2+</sup> ions.** Good agreement of the mean Nd-O distance and the BV-BL calculations considering the Nd(O)<sub>7</sub> model
- **Nd<sub>2</sub>O<sub>3</sub> incorporate within the depolymerized areas and create NBO's mostly as Q<sup>2</sup> units**
- There is an **obvious relationship between the composition threshold where modifier oxides are « lacking » to form the Nd(O)<sub>7</sub> complex and the crystallization of Nd-silicate apatite**
- Nd<sup>3+</sup> ions can play different structural roles in the aluminoborosilicate system with very different compositions to that of glass A (peraluminous, LaBS glasses, etc)

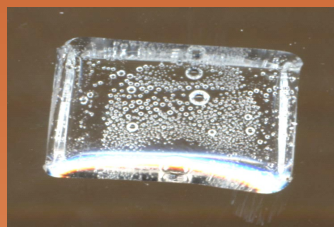
*Thank you for your kind attention !*

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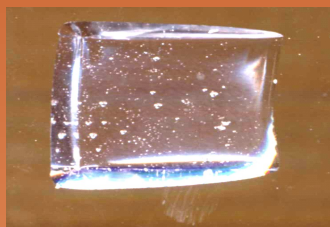


ParisTech

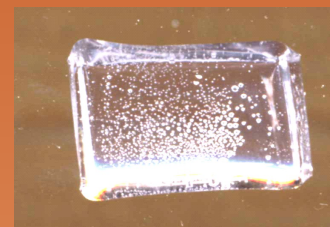




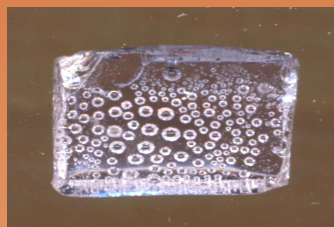
Nd0



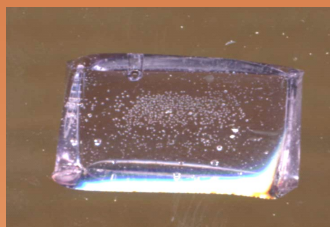
Nd1,3



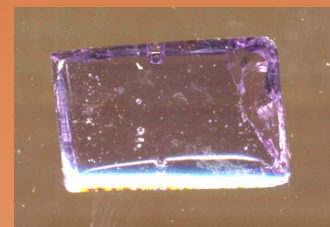
Nd2,5



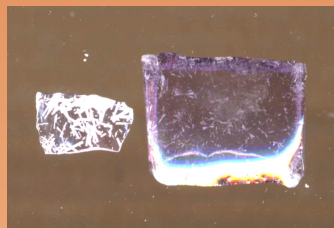
Nd5



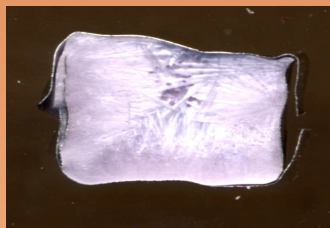
Nd10



Nd16



Nd20



Nd25



Nd30