

Effect of B_2O_3 addition on the structure and crystallisation properties of soda-lime borosilicate glasses containing lanthanides

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Thibault Charpentier, Jean-Luc Dussossoy

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CEA Iramis, SISMM

CEA/DEN/DTCD/SECM/LEDMC



ParisTech

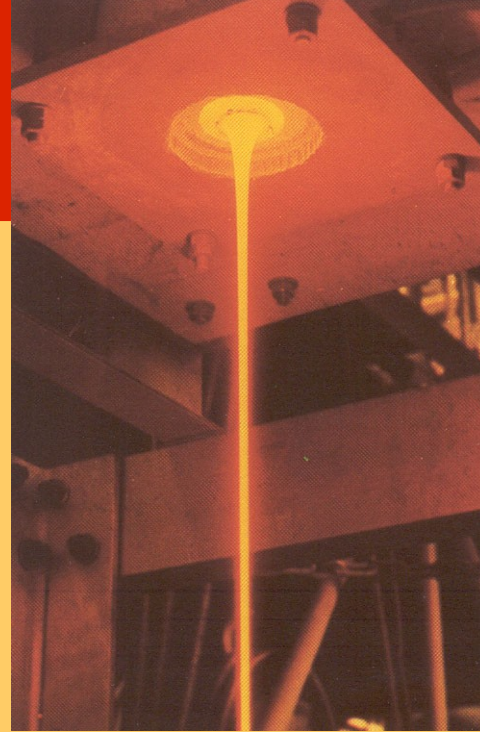


Introduction and outline

Context : composition – structure – crystallisation properties
relationships in nuclear waste containment glasses

Model **glass A** (mol %) :

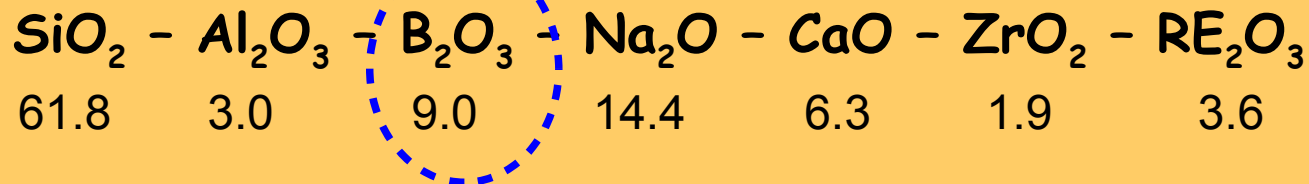
SiO₂	Al₂O₃	B₂O₃	Na₂O	CaO	ZrO₂	RE₂O₃
61.8	3.0	9.0	14.4	6.3	1.9	3.6



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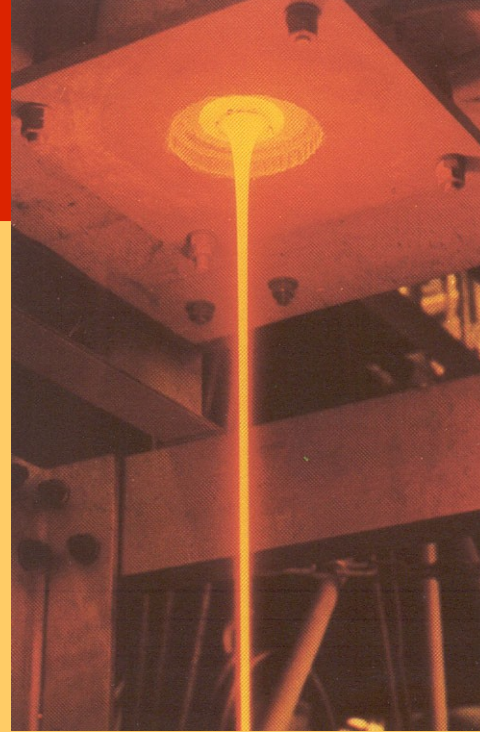
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Bx glass series with B_2O_3 content from 0 to 27 mol%

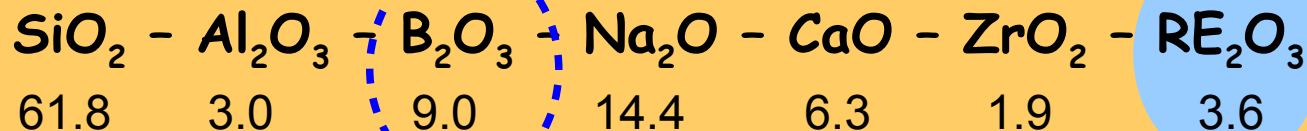
B_2O_3 is *added* to keep constant all other oxide molar ratios



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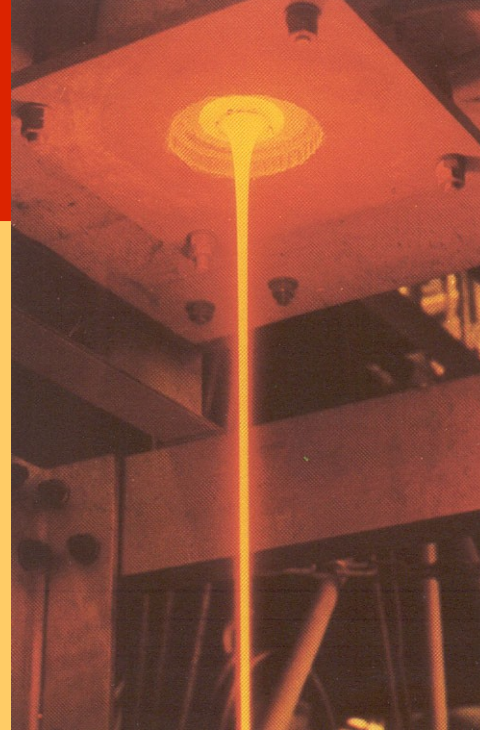
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Two
glass
series

RE = Nd

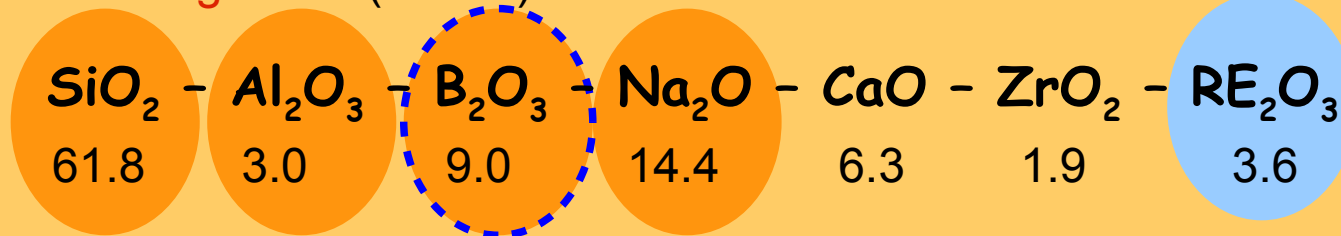
Optical absorption, Nd L_3 -edge EXAFS



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Model **glass A** (mol %) :



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B₂O₃ is *added* to keep constant all other oxide molar ratios

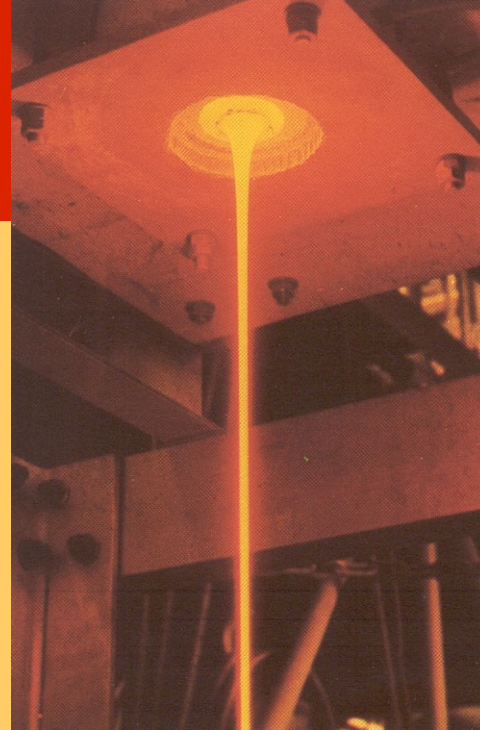
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RE = La + 0.15 mol% Nd

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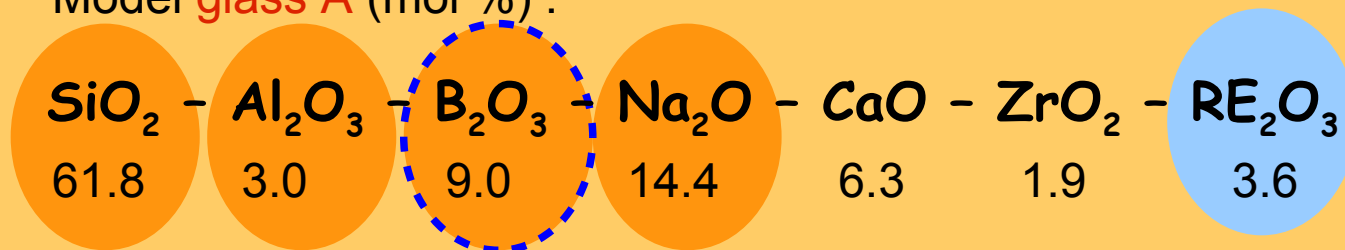
¹¹B, ²⁷Al, ²³Na NMR, Nd³⁺ luminescence



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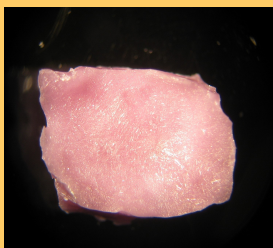
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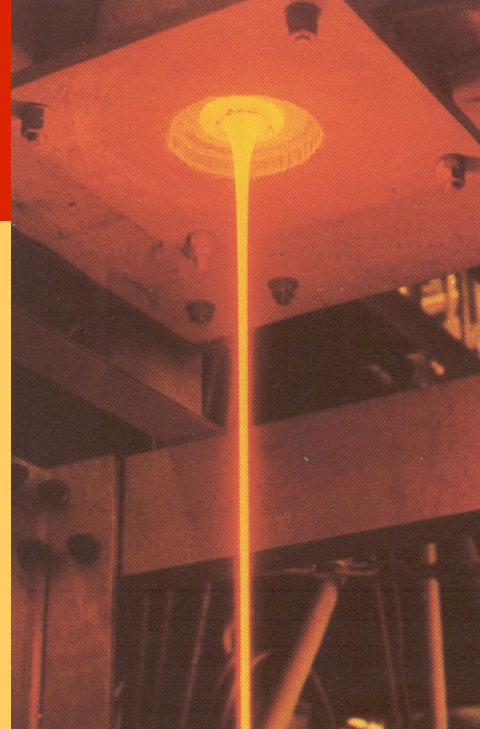
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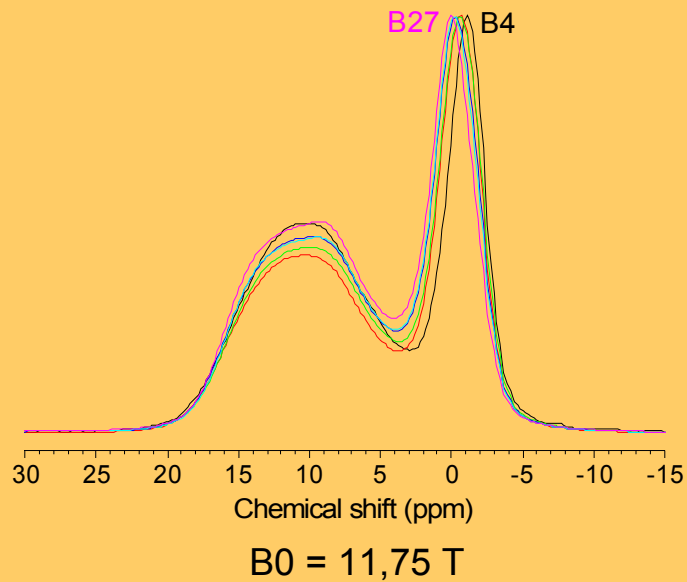
Impact of B_2O_3 content on the crystallisation of a RE-rich silicate phase near $\text{Ca}_2\text{Nd}_8(\text{SiO}_4)_6\text{O}_2$ composition



The glass network : ^{11}B and ^{23}Na MAS NMR

^{11}B MAS NMR

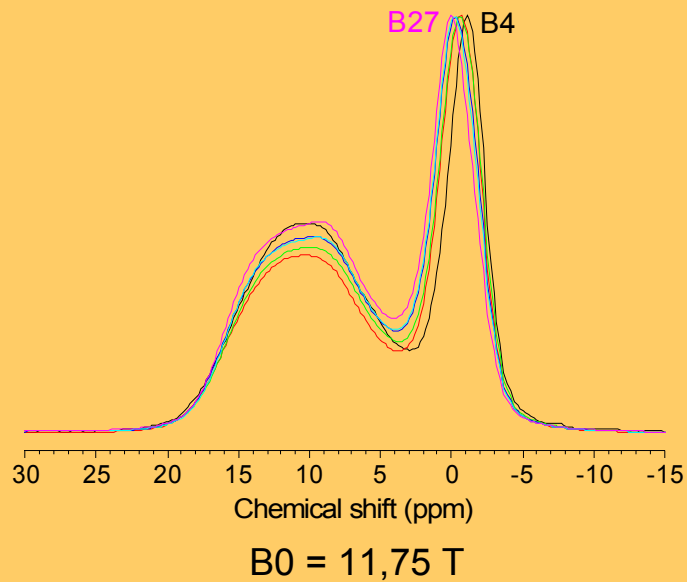
$N_4 = 40\%$ accross
the Bx series



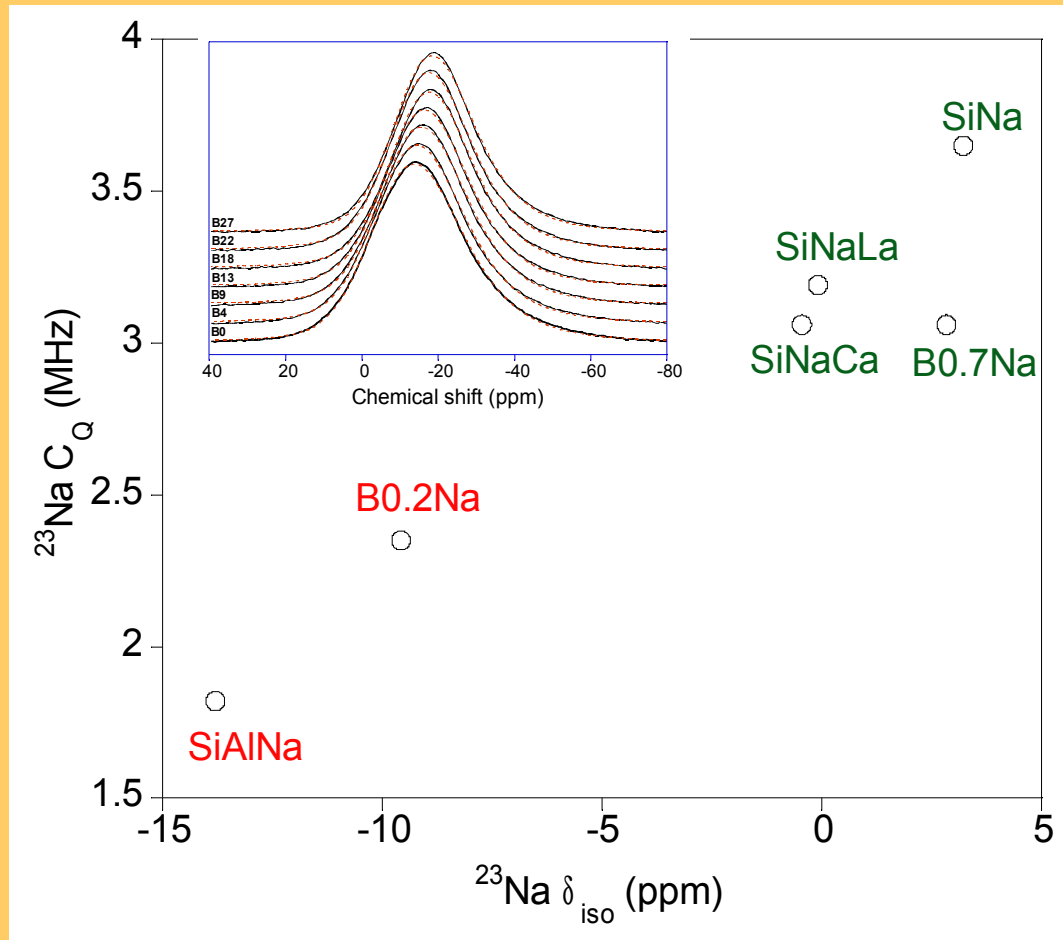
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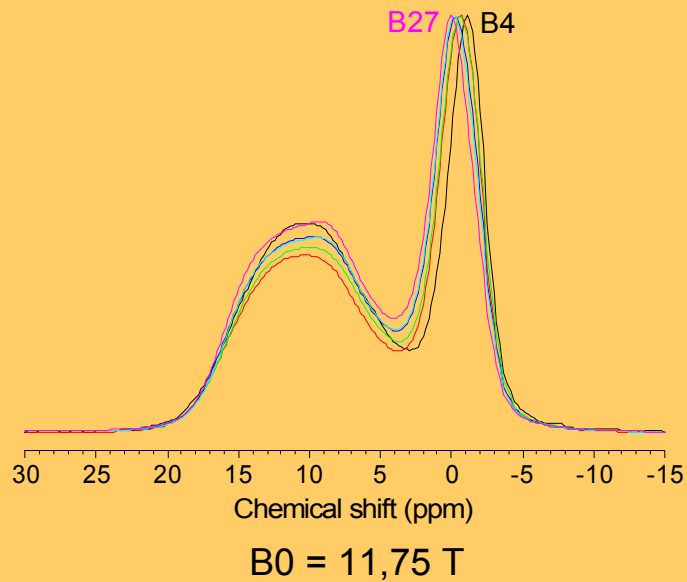
^{23}Na MAS NMR



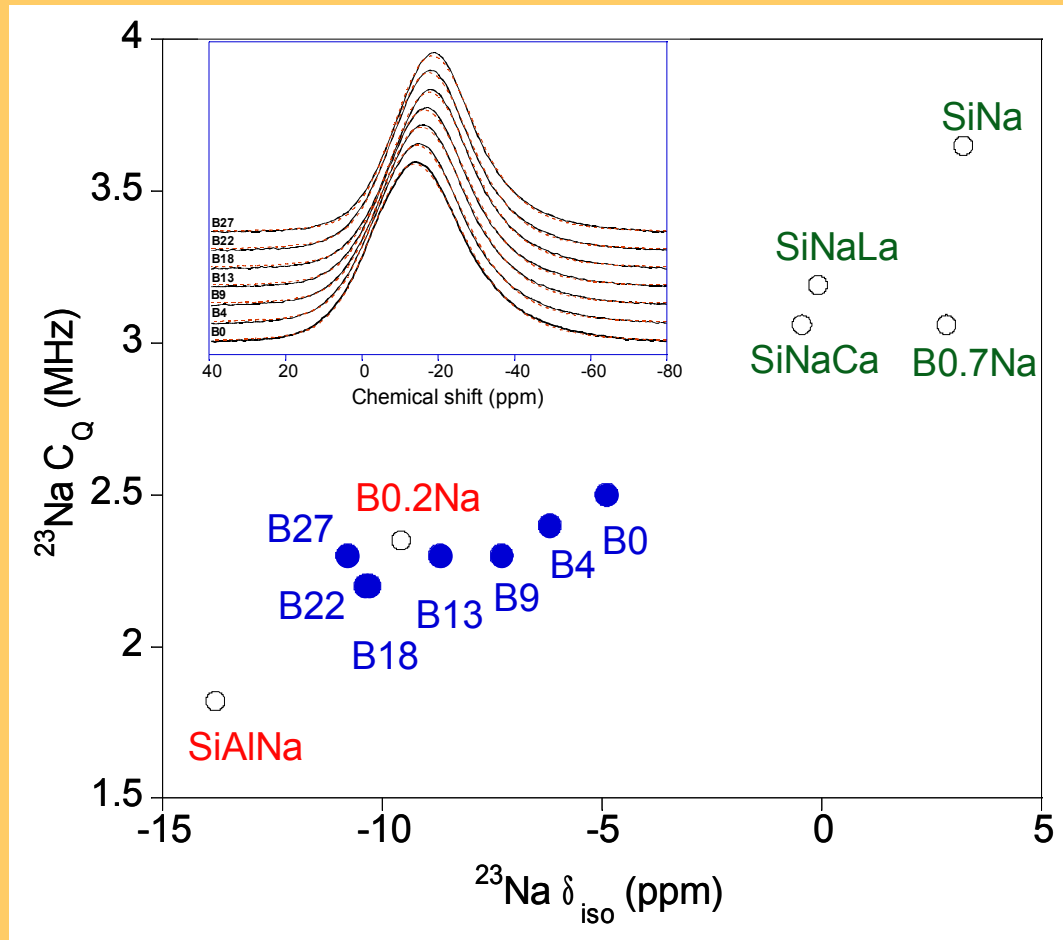
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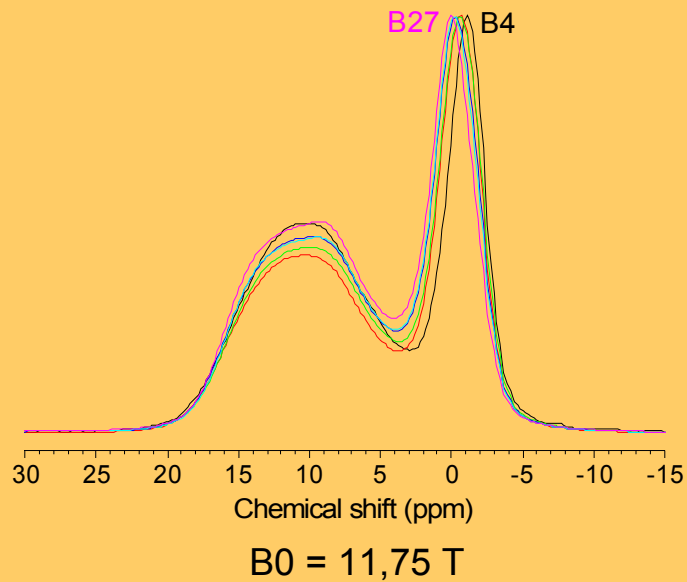
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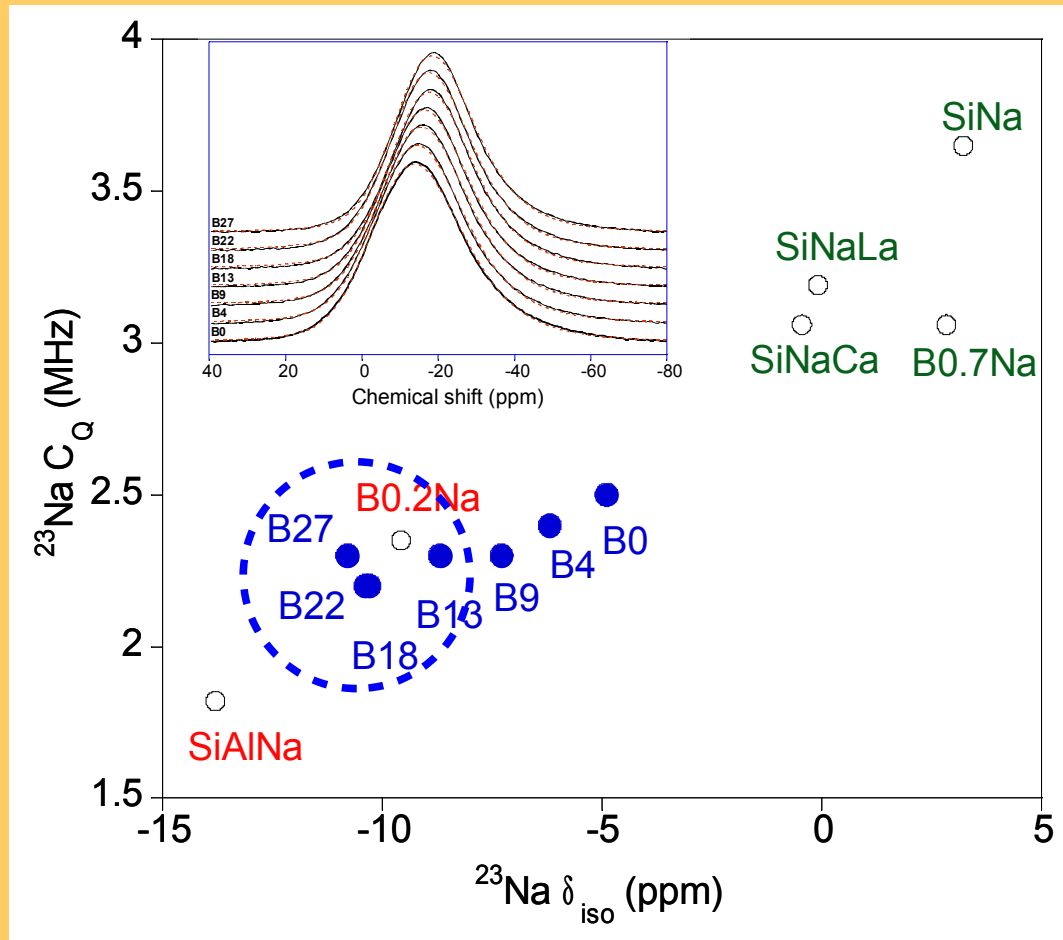
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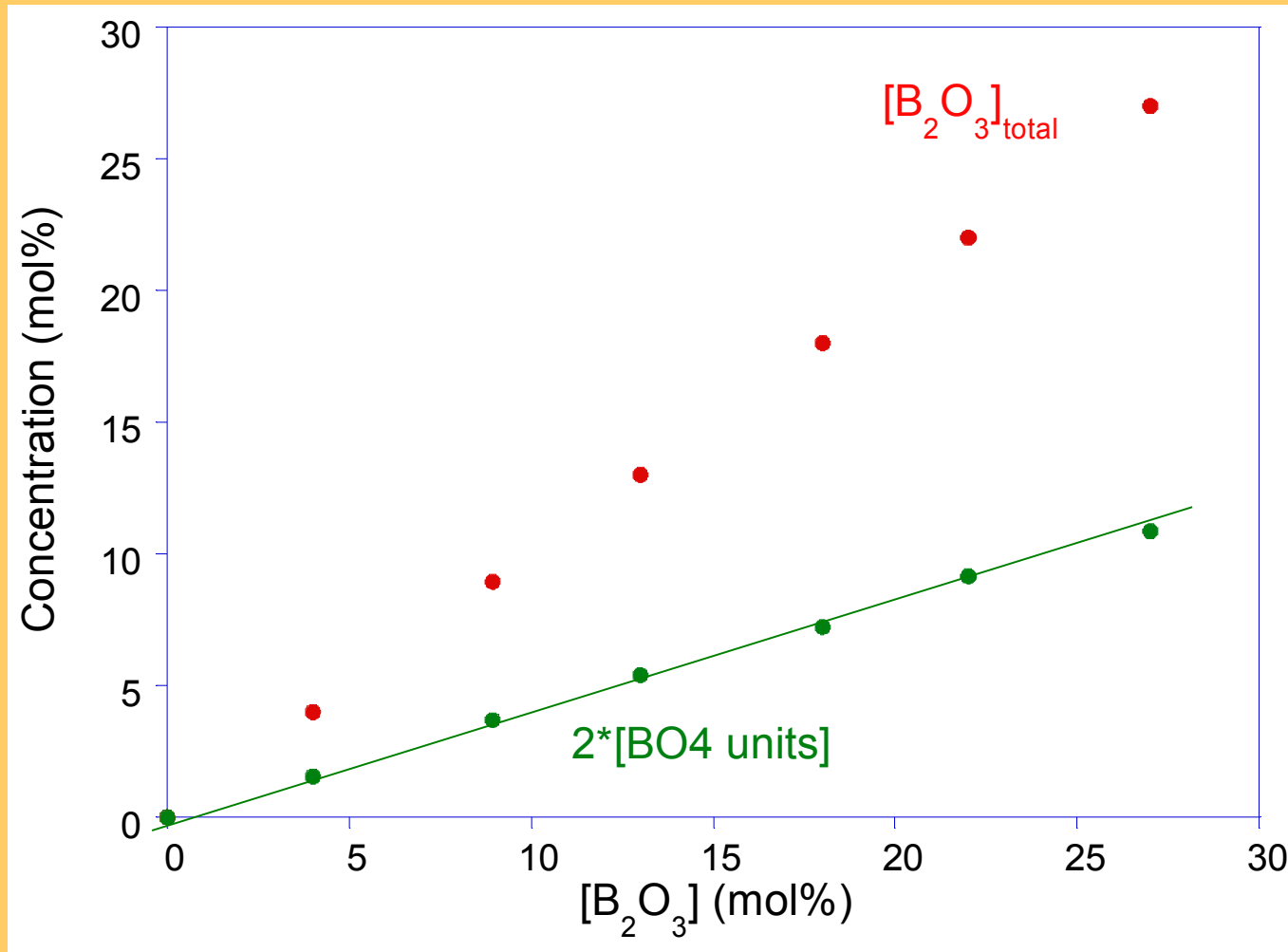
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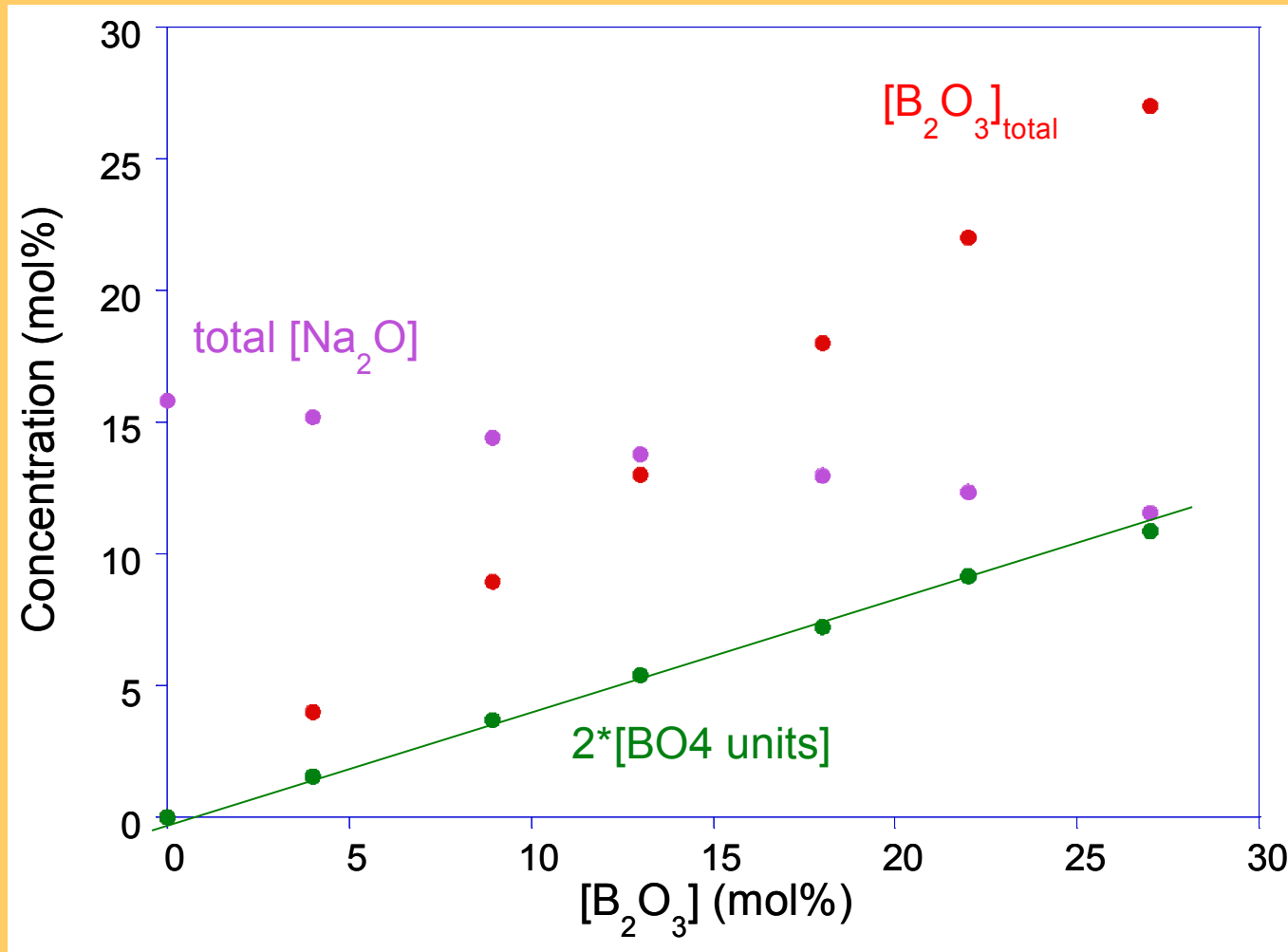
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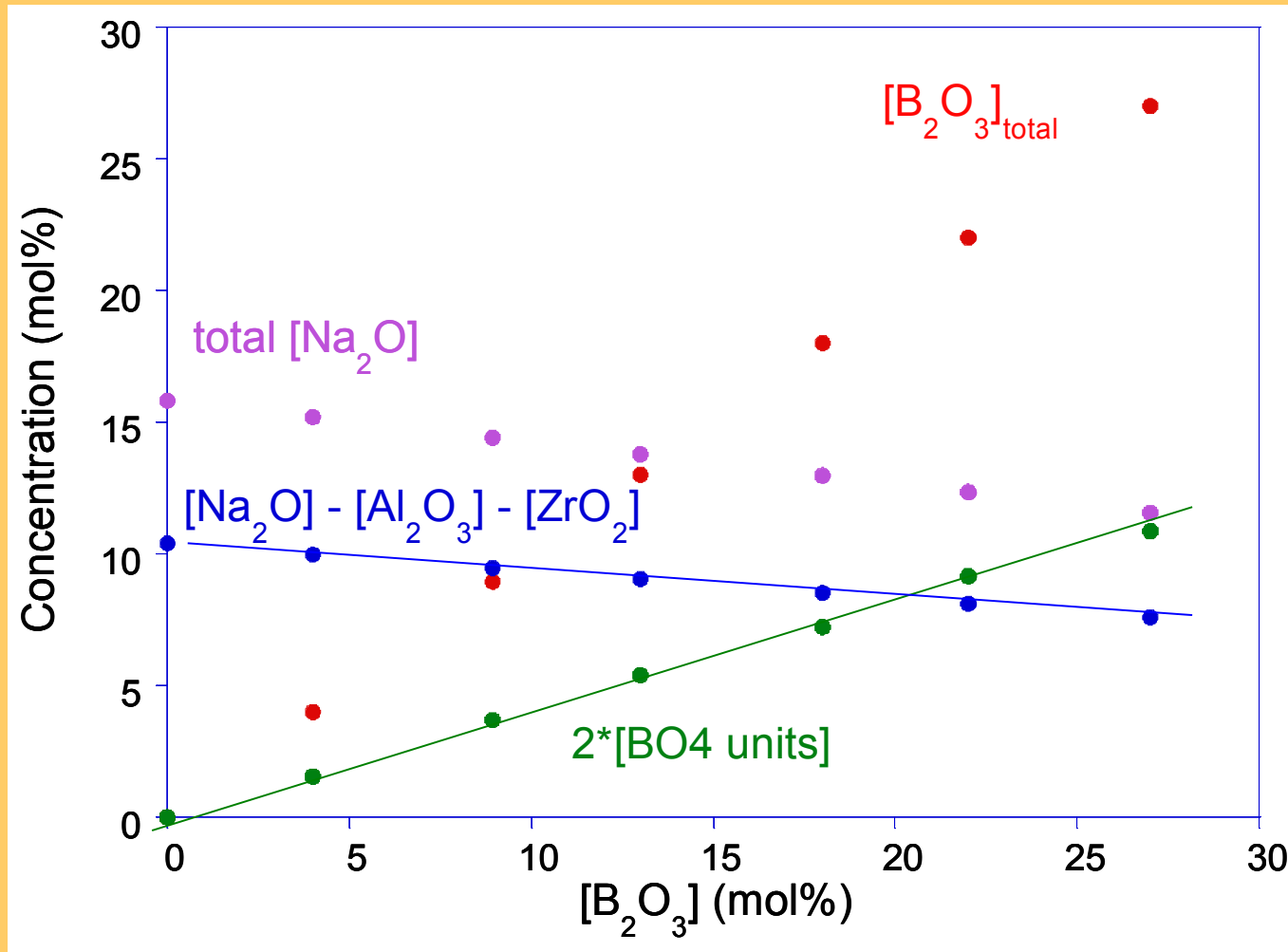
B_2O_3 modifies the structural role of Na^+ ... and of other cations



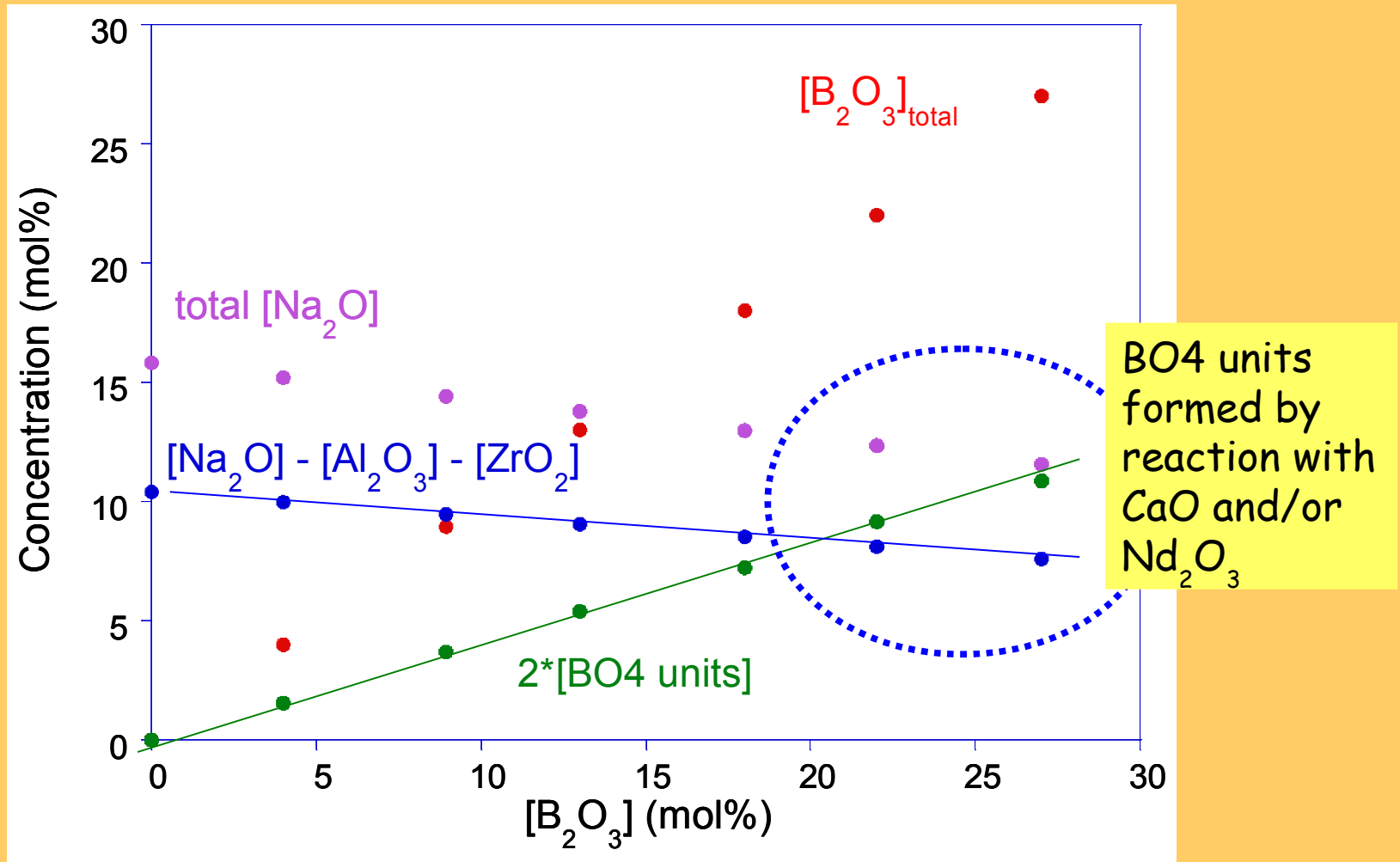
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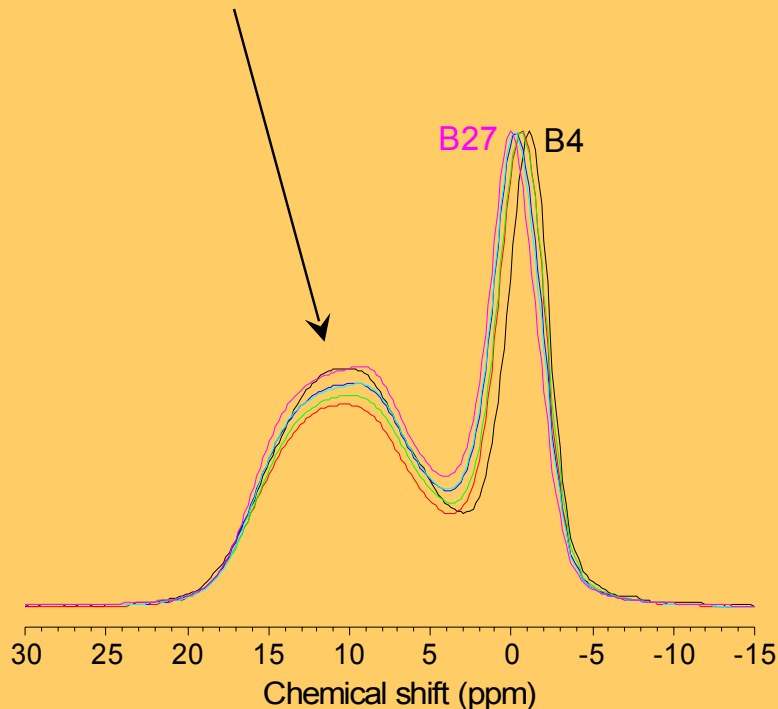


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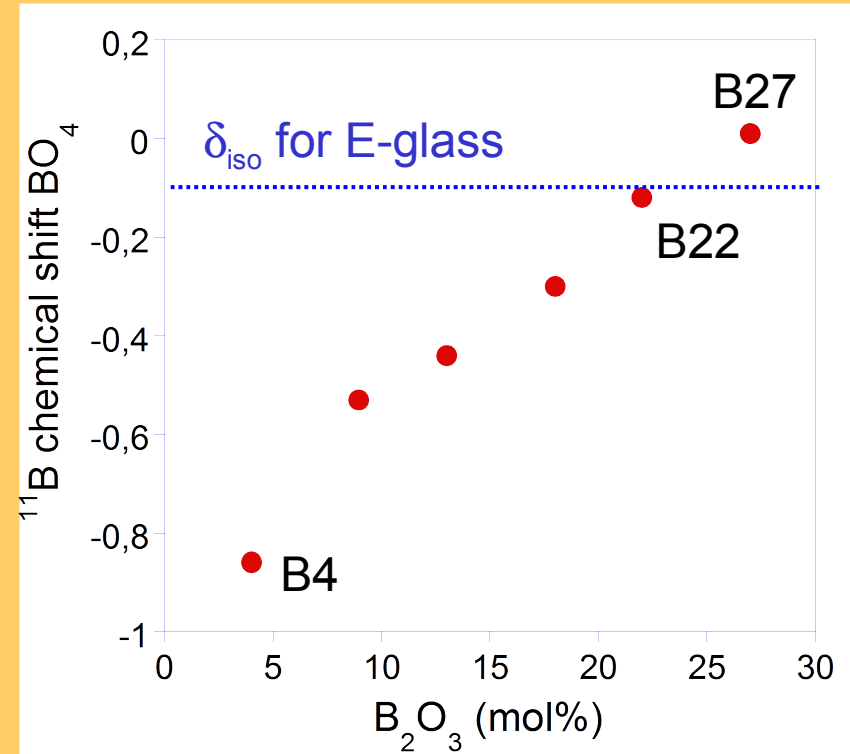
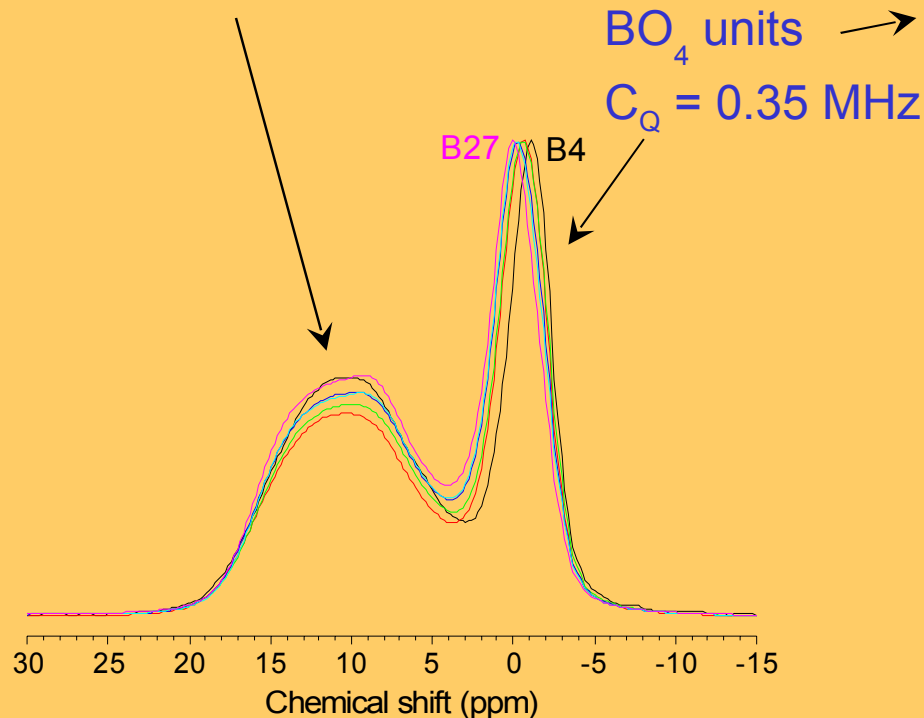
Glass network homogeneity : ^{11}B MAS NMR

Lorentzian broadening because
of interaction with Nd^{3+}
electronic spin (0.15 mol%) $\rightarrow$
no valuable fit for BO_3



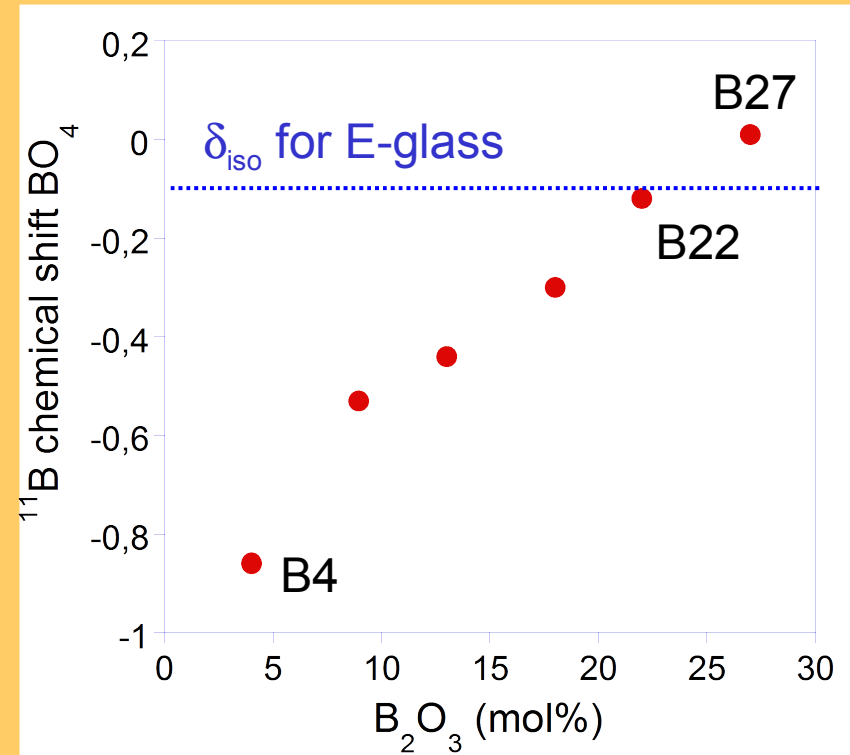
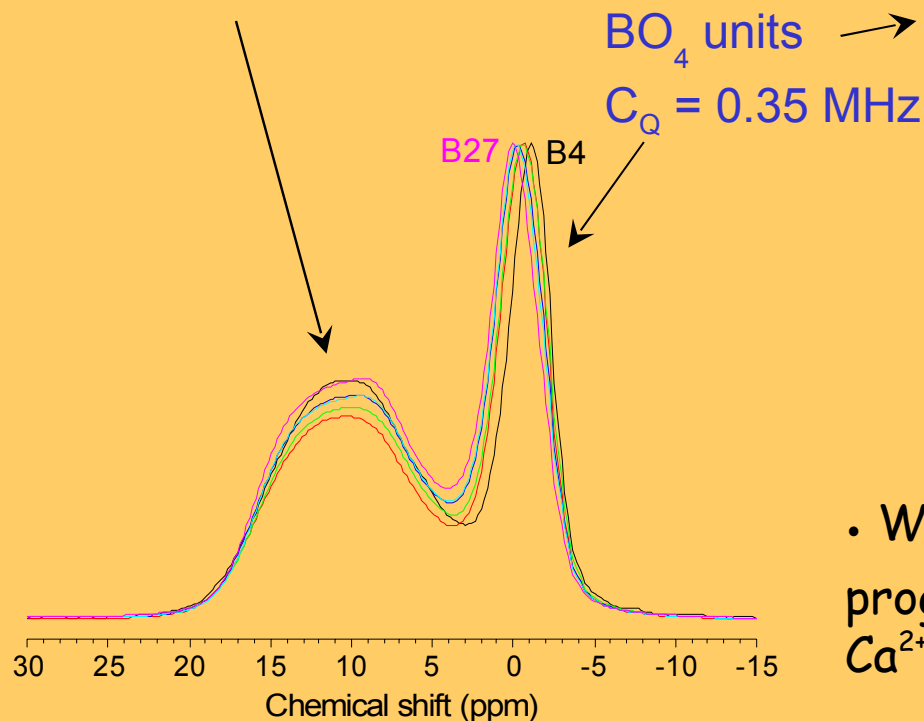
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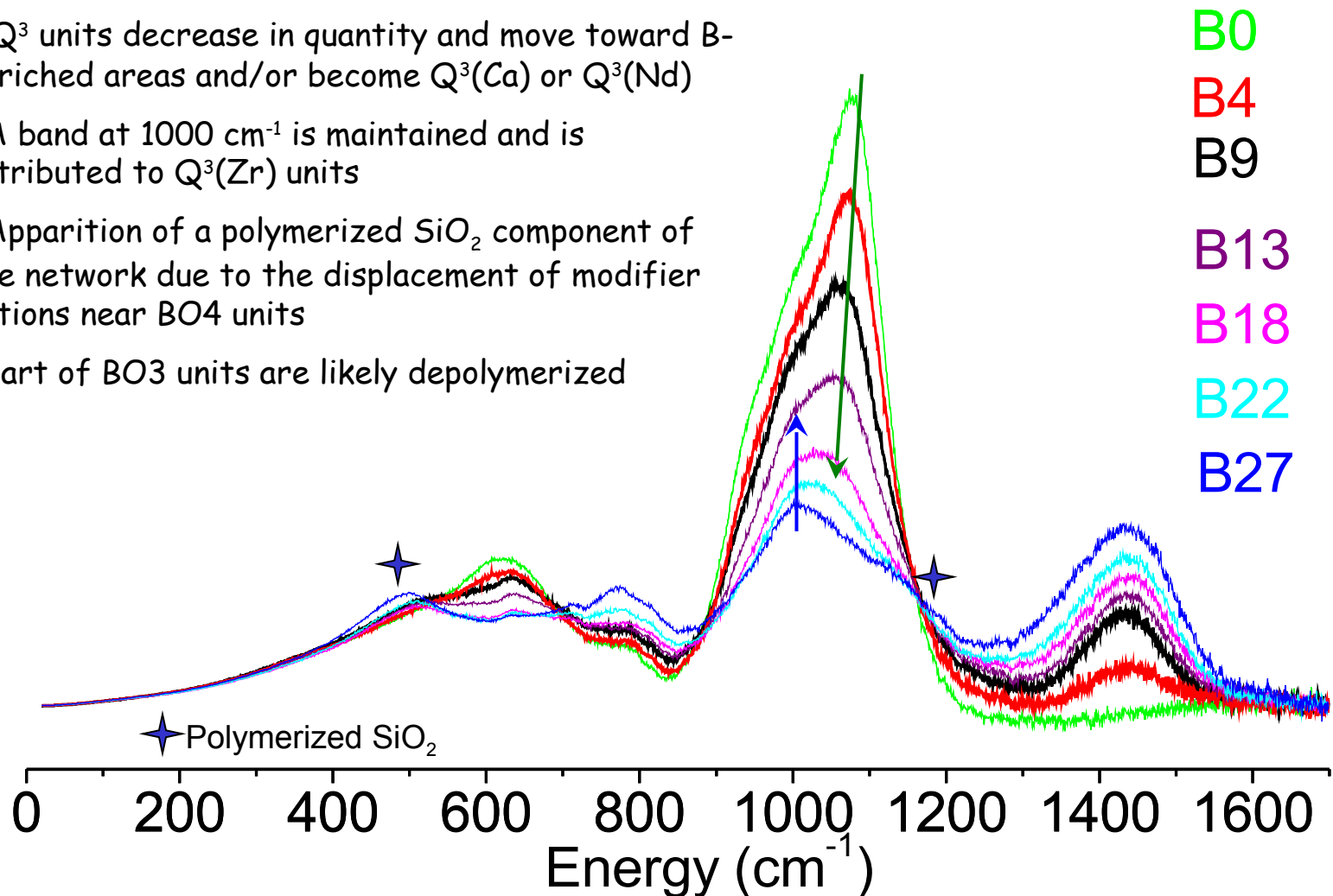
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- When adding B_2O_3 , new BO_4 units are progressively charge-compensated by Ca^{2+} and/or Nd^{3+}
- Progressive B-enrichment around BO_4 units is also possible

Glass network homogeneity : Raman spectroscopy

- Q^3 units decrease in quantity and move toward B-enriched areas and/or become $Q^3(\text{Ca})$ or $Q^3(\text{Nd})$
- A band at 1000 cm^{-1} is maintained and is attributed to $Q^3(\text{Zr})$ units
- Apparition of a polymerized SiO_2 component of the network due to the displacement of modifier cations near BO_4 units
- Part of BO_3 units are likely depolymerized



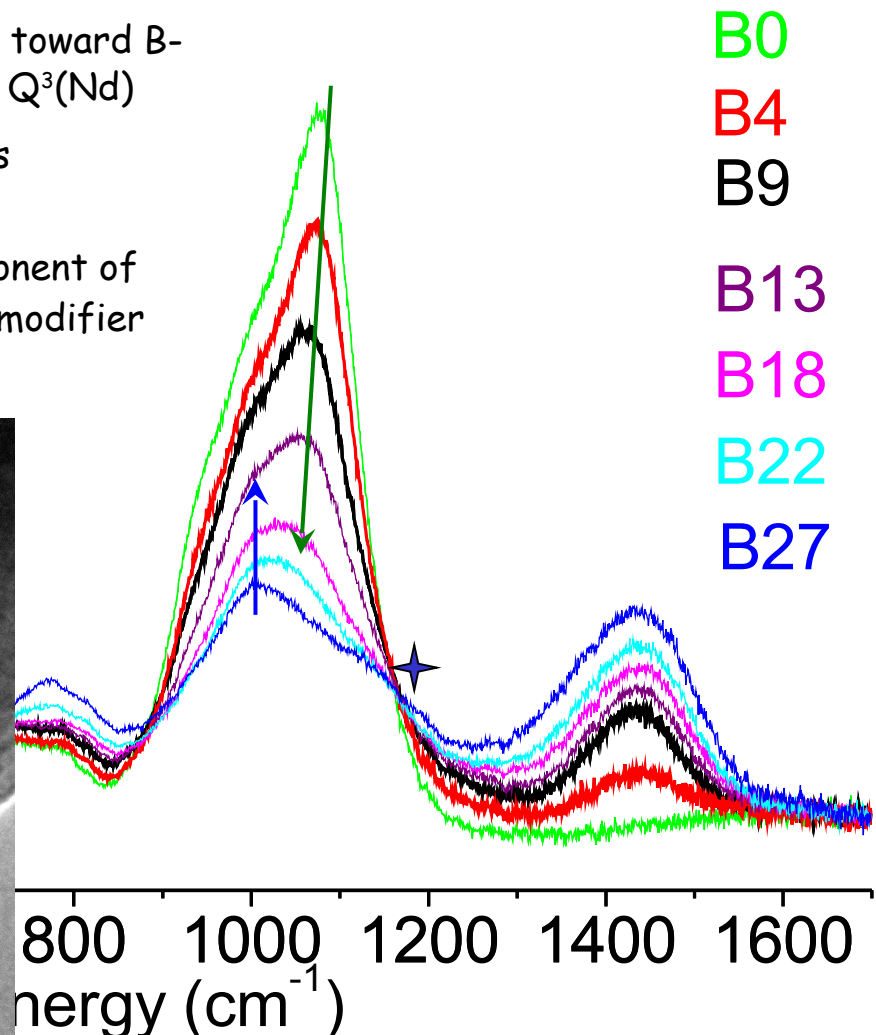
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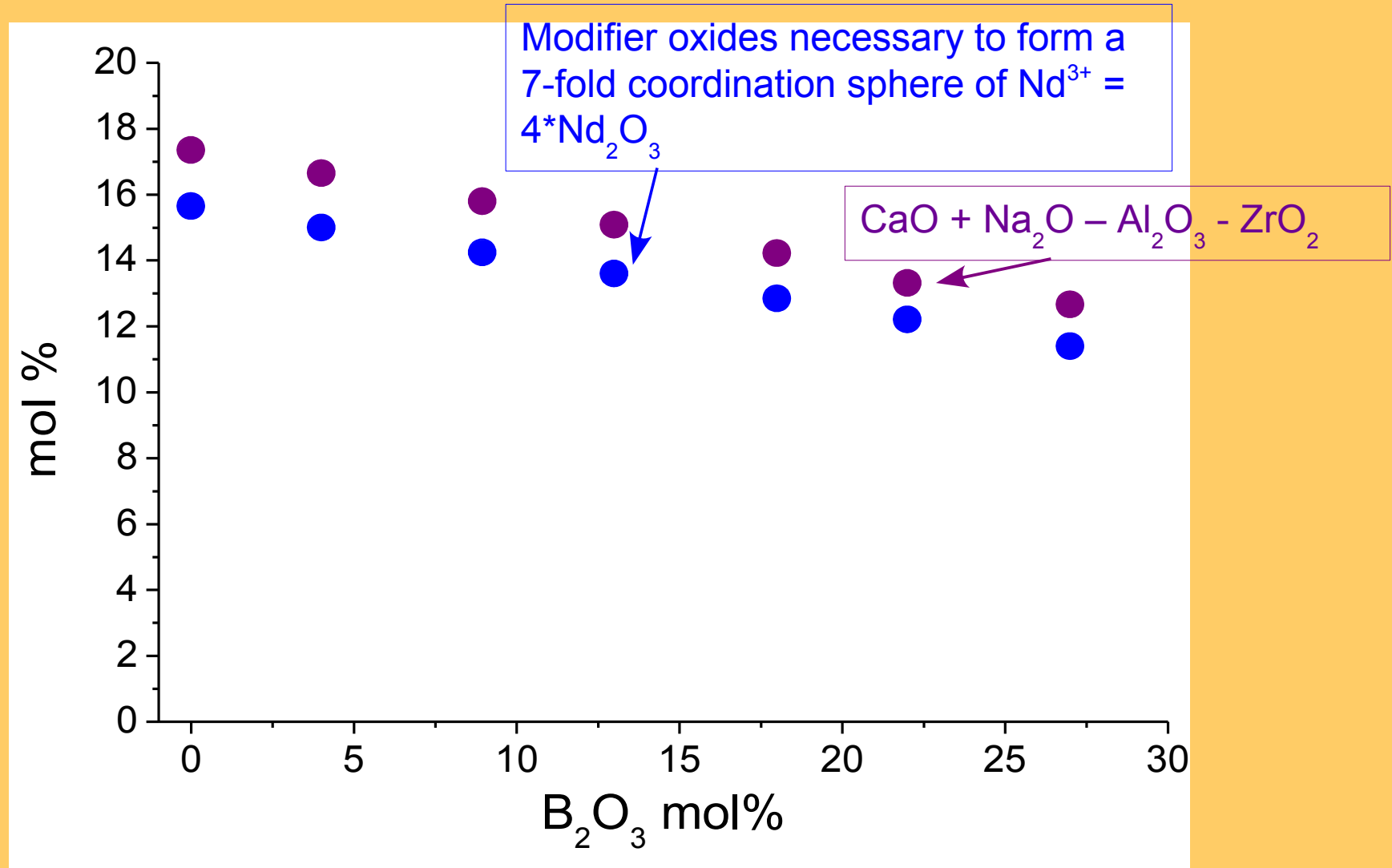
...but no phase separation at the TEM scale!

(B27 glass)

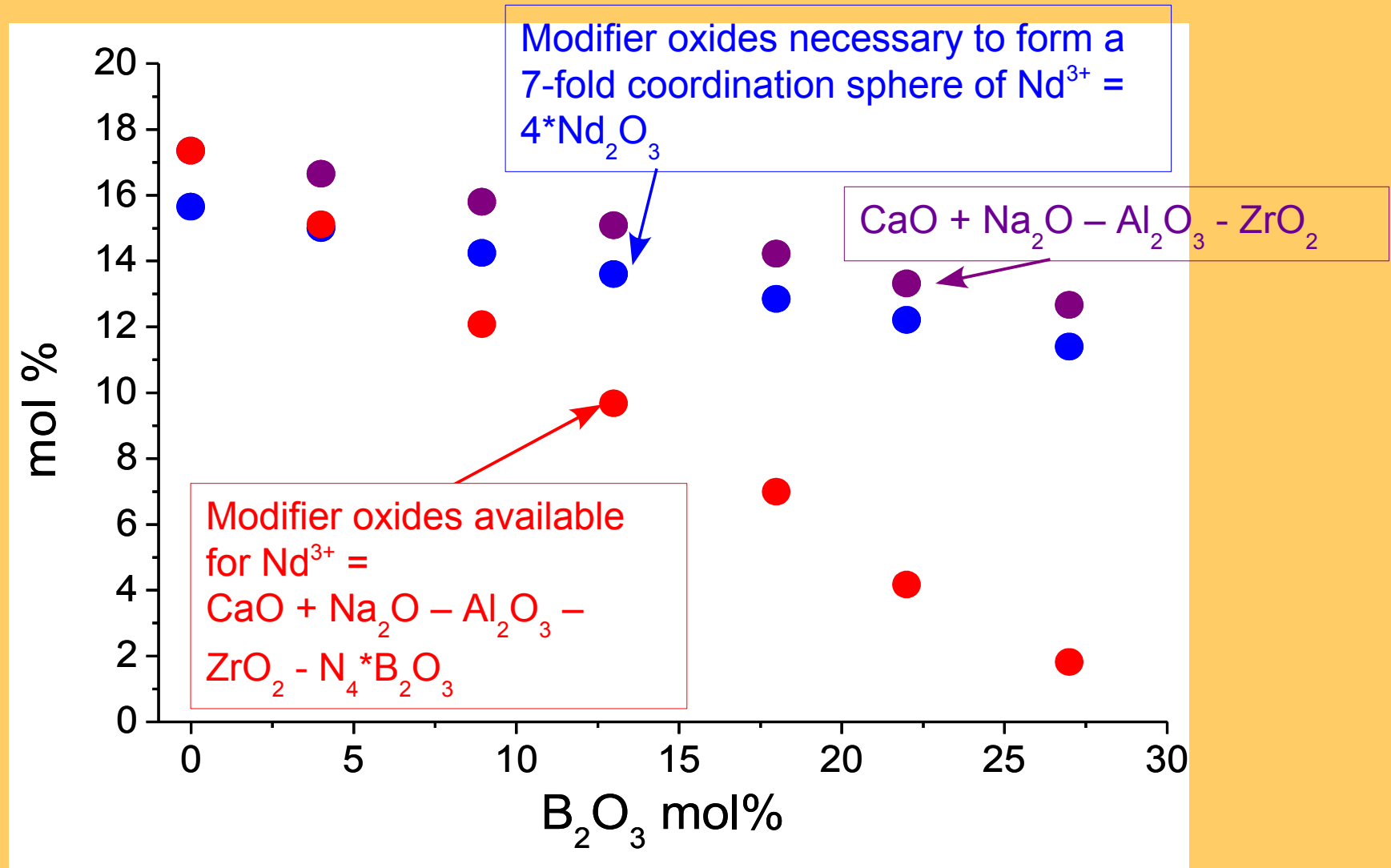
50 nm



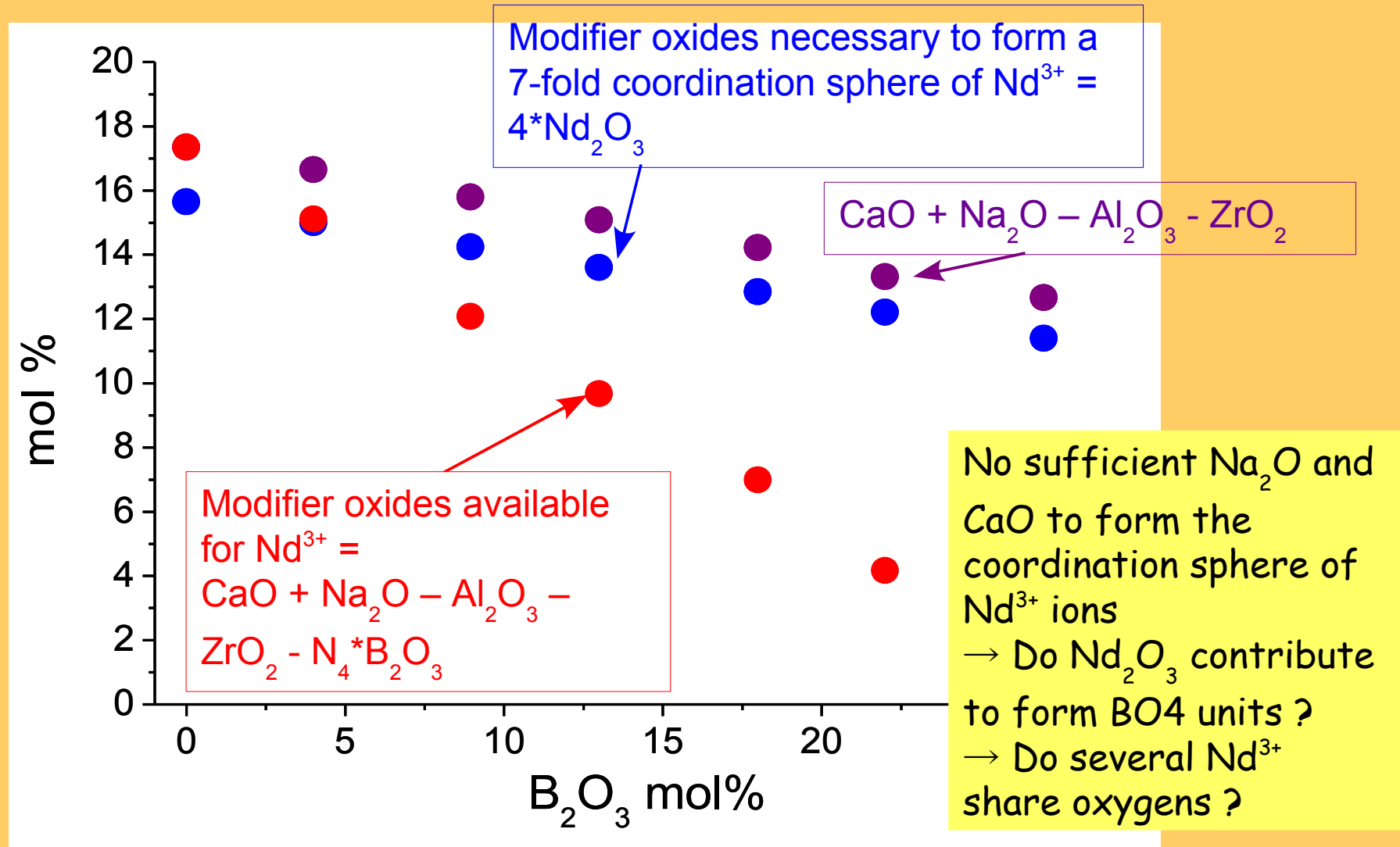
Nd³⁺ local environment : considerations from the composition



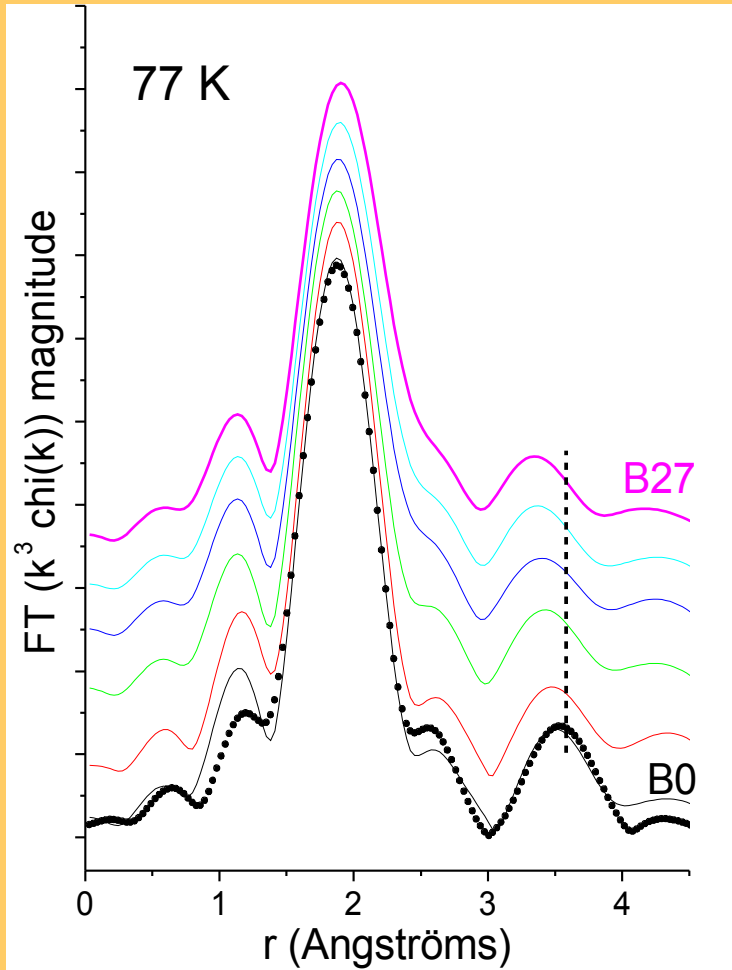
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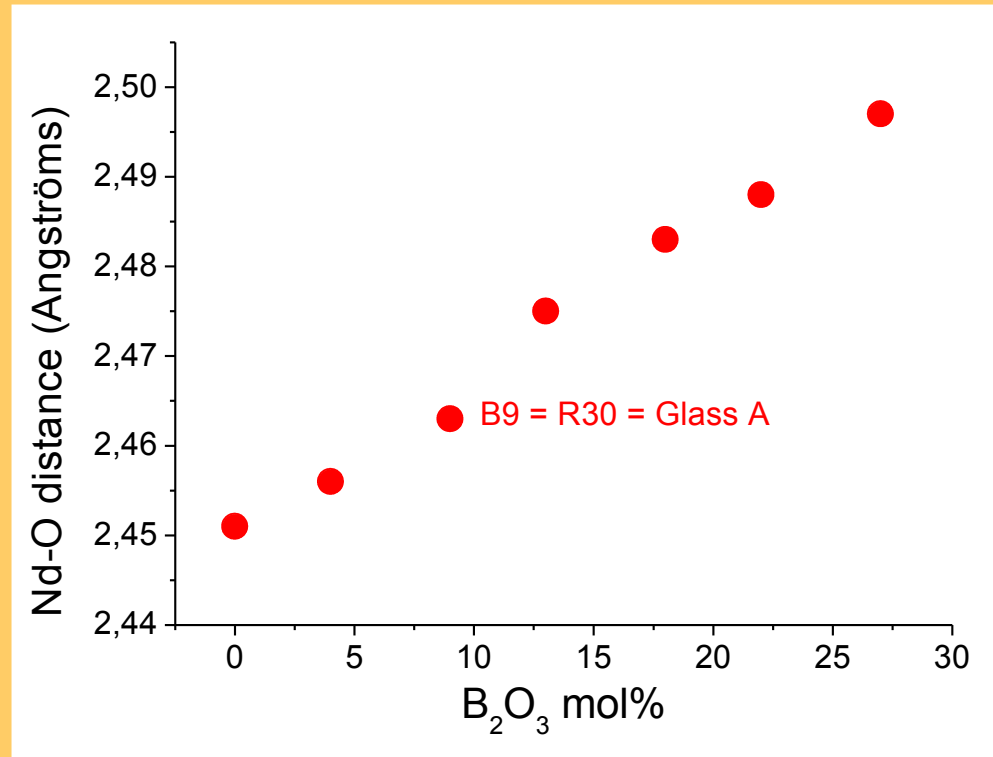
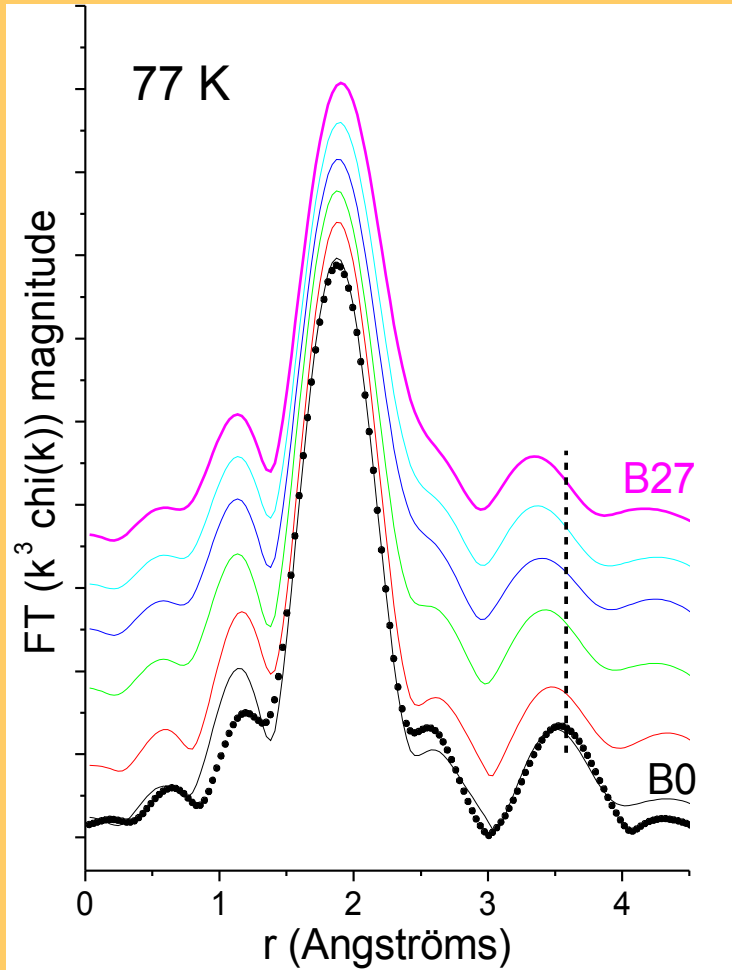
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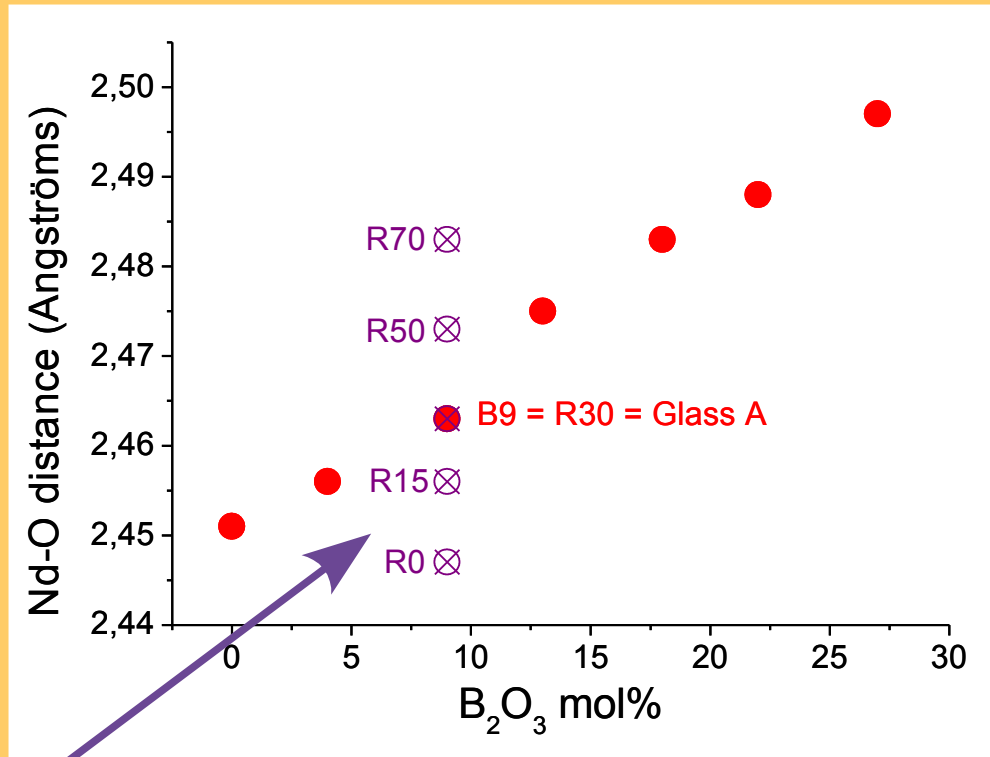
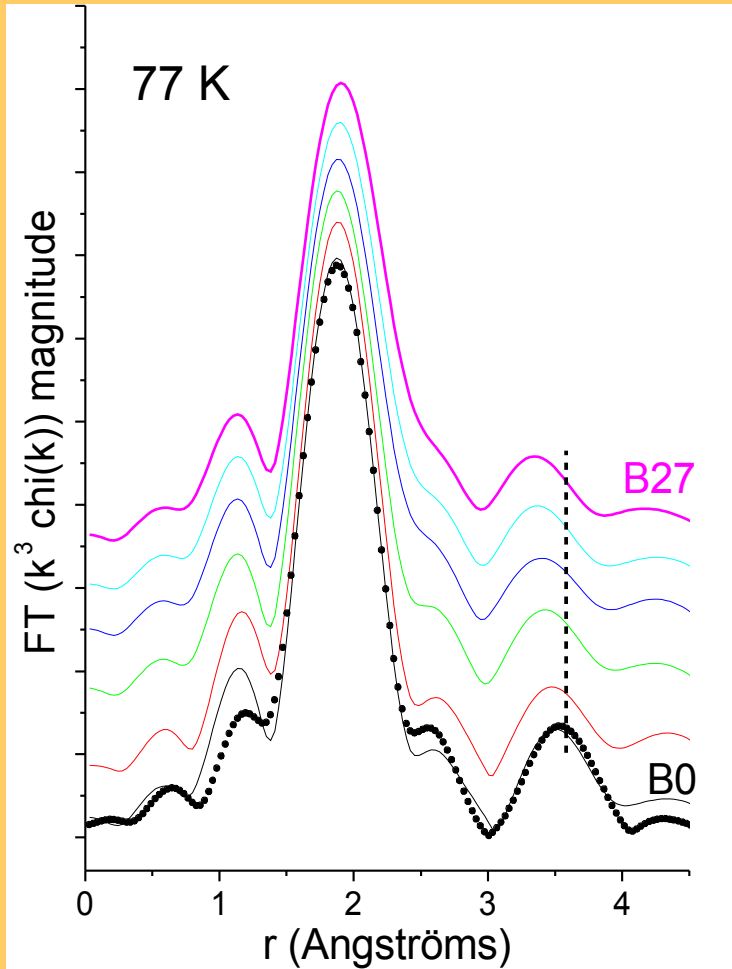
Nd³⁺ local environment : EXAFS at the Nd L₃-edge



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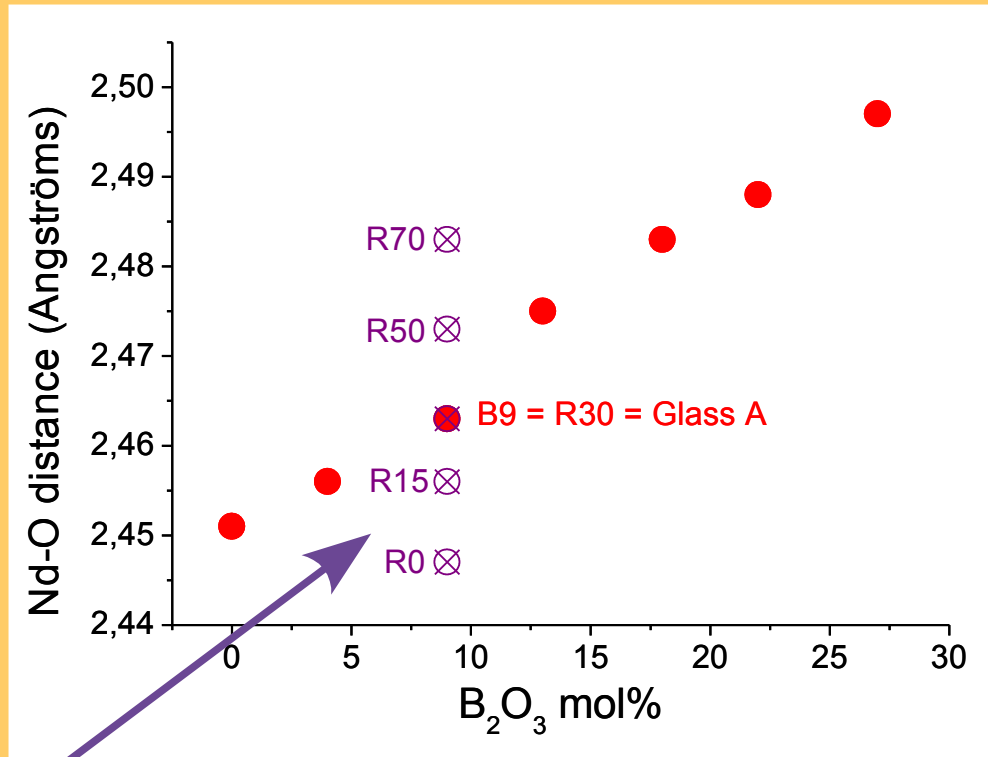
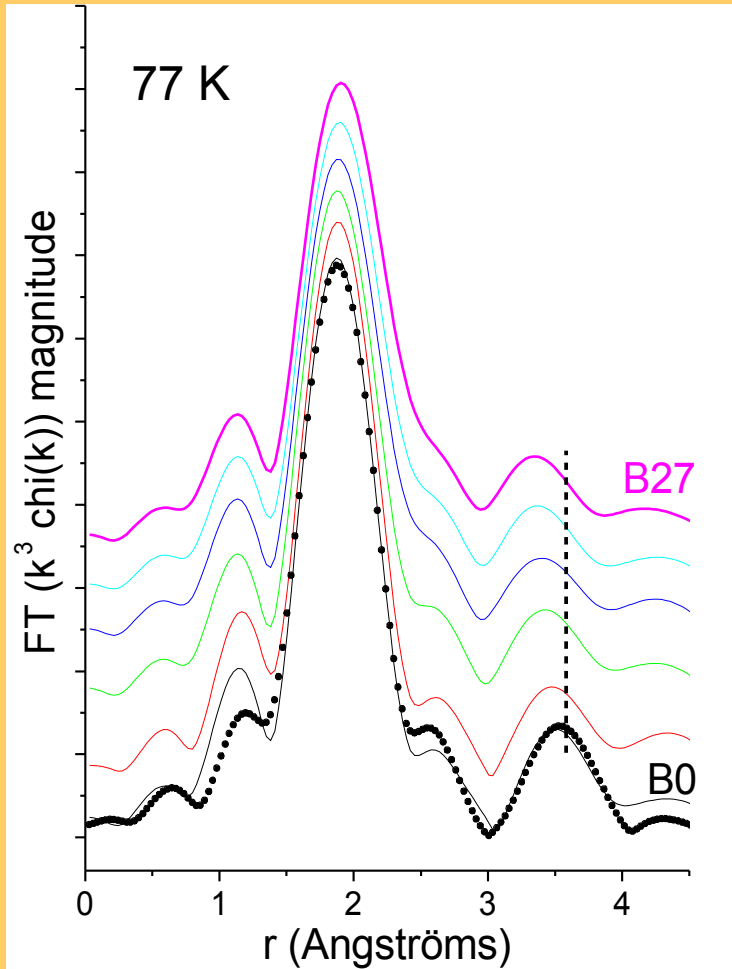


Nd³⁺ local environment : EXAFS at the Nd L₃-edge



Comparison with R glass series
 $R = \text{CaO} / (\text{CaO} + \text{Na}_2\text{O})$

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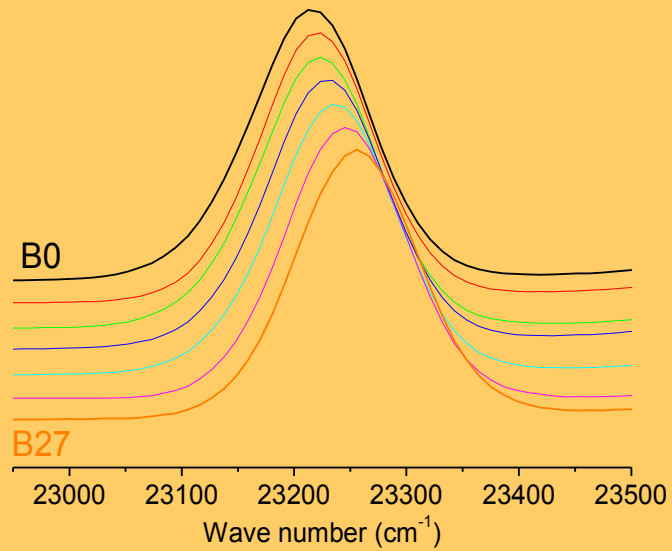


Comparison with R glass series
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Nd-O mean distance increases with B₂O₃ content on a wider range than that due to Ca/Na substitution

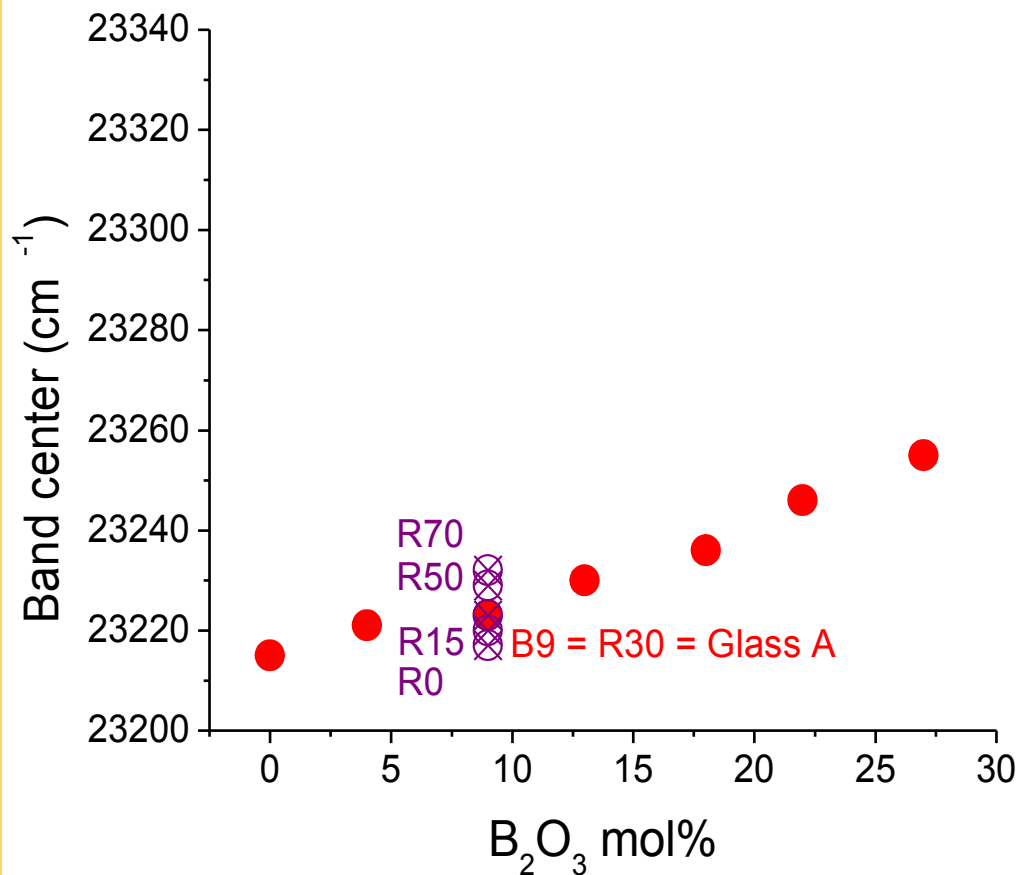
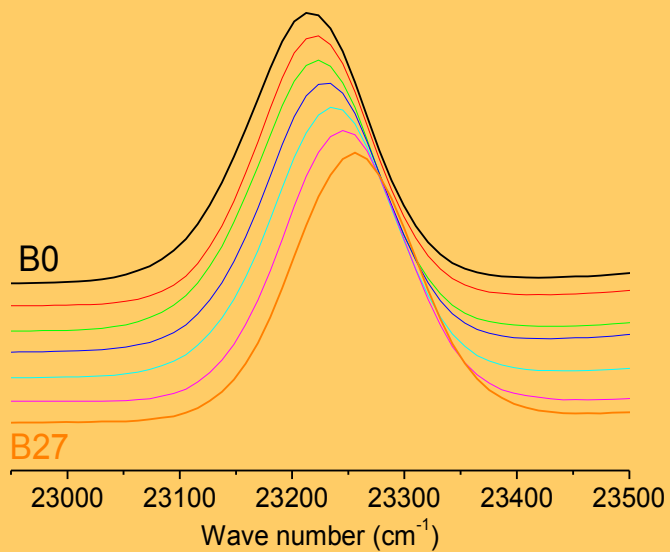
Nd³⁺ local environment : optical spectroscopy

$^4I_{9/2} \rightarrow ^2P_{1/2}$ transition at 10 K



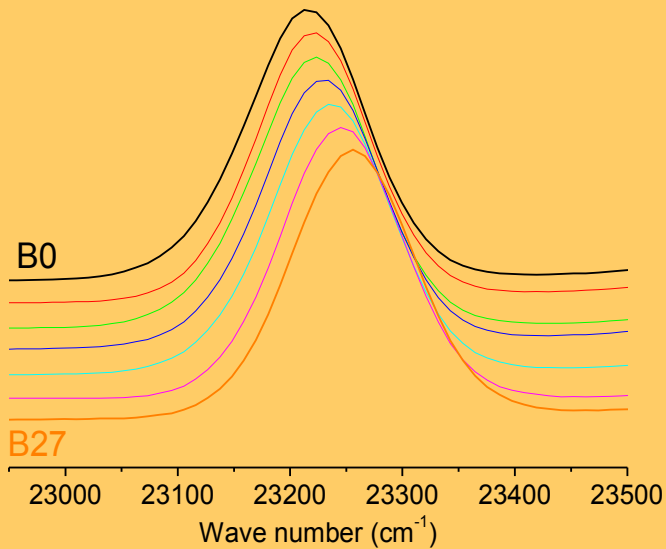
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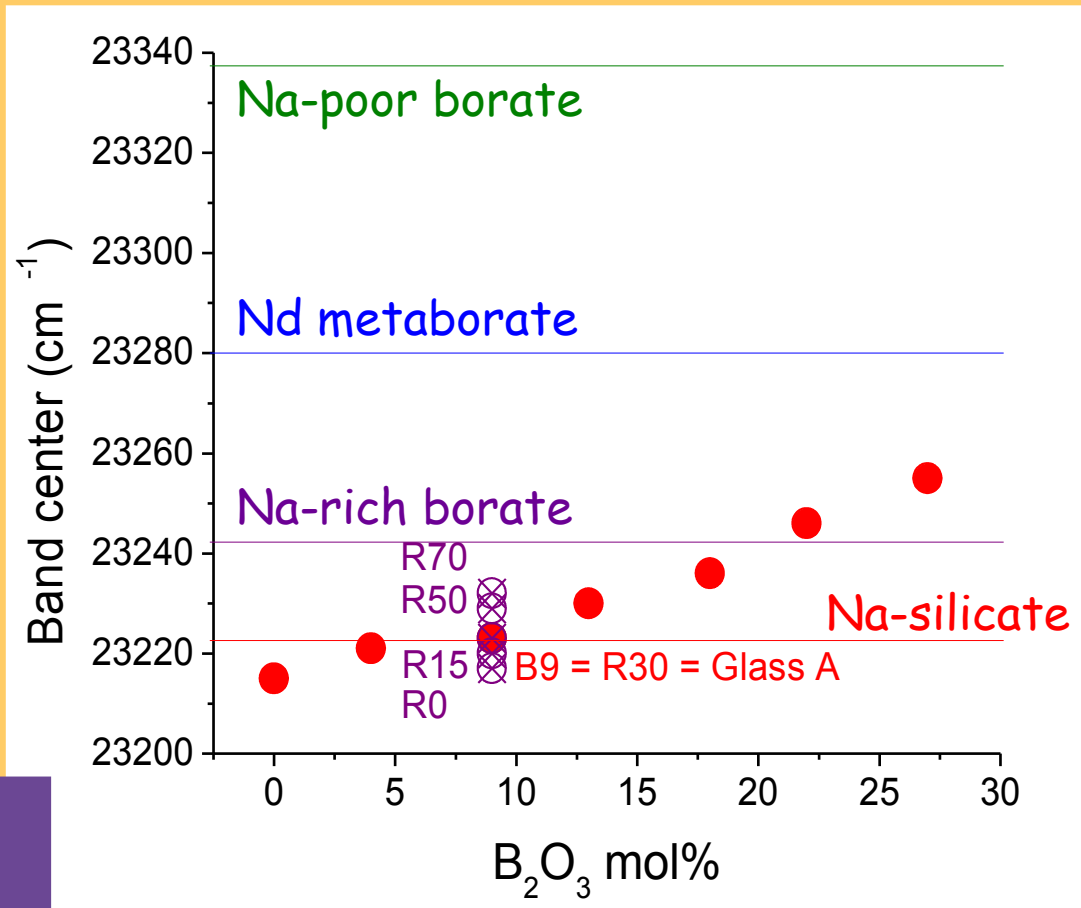


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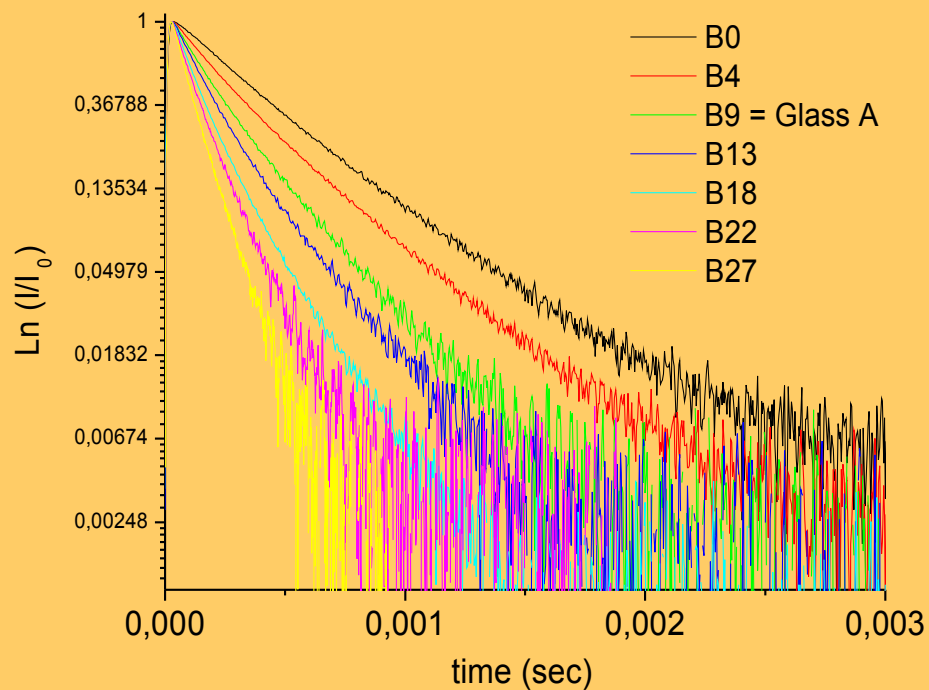
Comparison with R series and
with simple model glasses



Nephelauxetic shift toward high energy consistent with dN-O increasing
At high B₂O₃ content, band center close to that of Na-rich borate environment

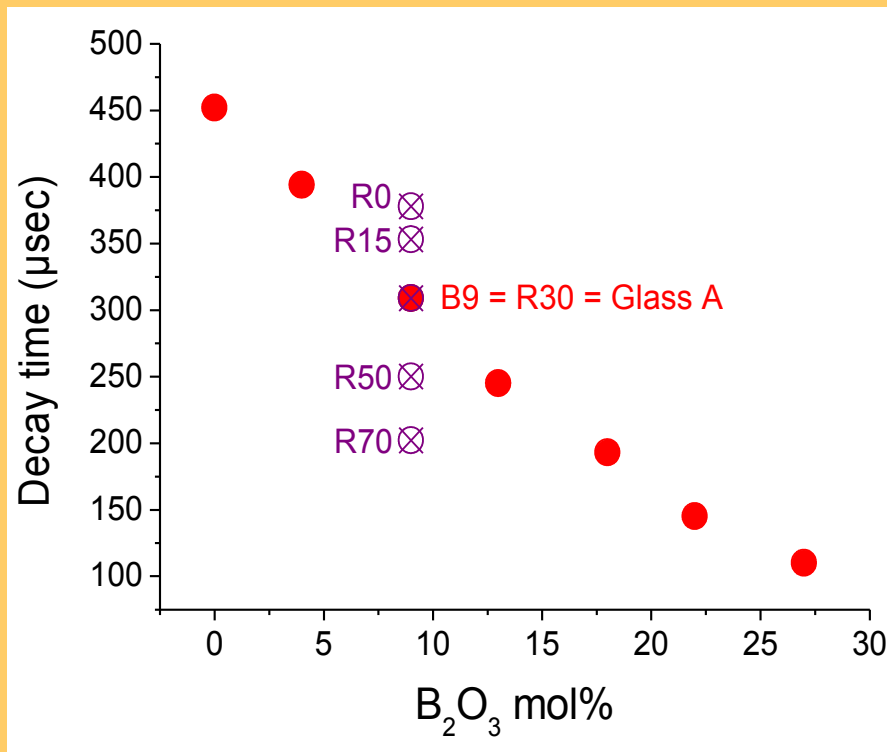
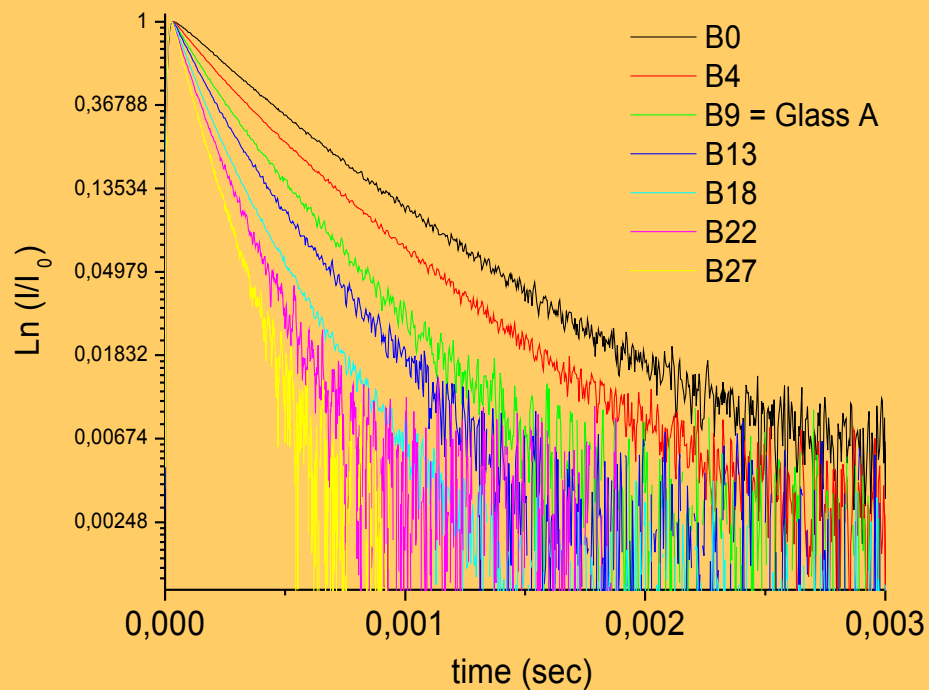
Nd³⁺ local environment : luminescence decays

Nd³⁺ global concentration $2.1 \cdot 10^{19}$ Nd.cm⁻³ for all glasses



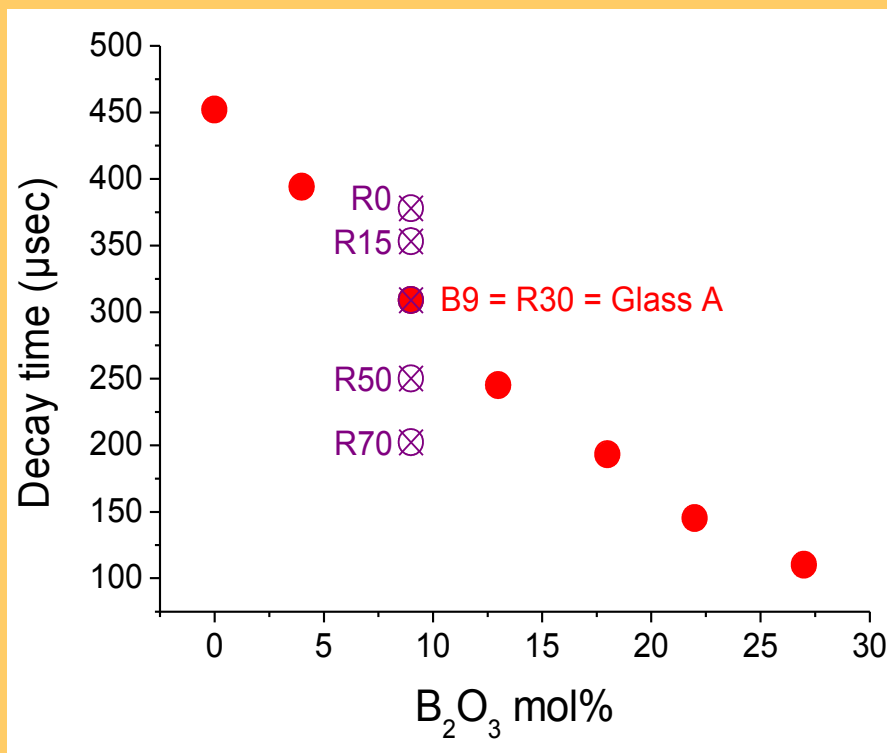
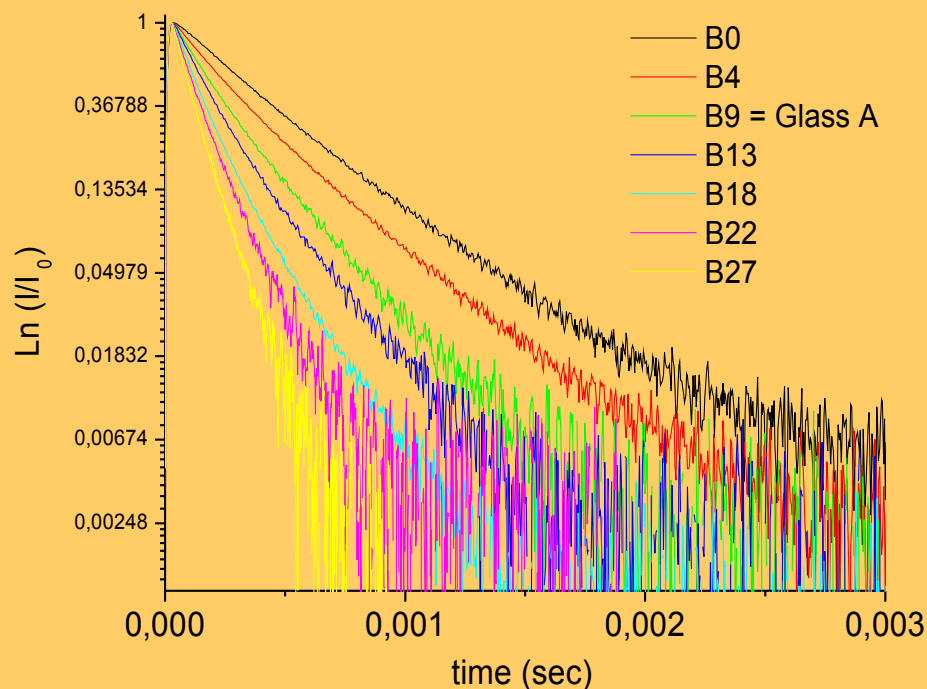
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$$\frac{1}{\tau} = \frac{1}{\tau_{rad}} + W_{phonons} + W_{migration} + W_{cross-relaxation}$$

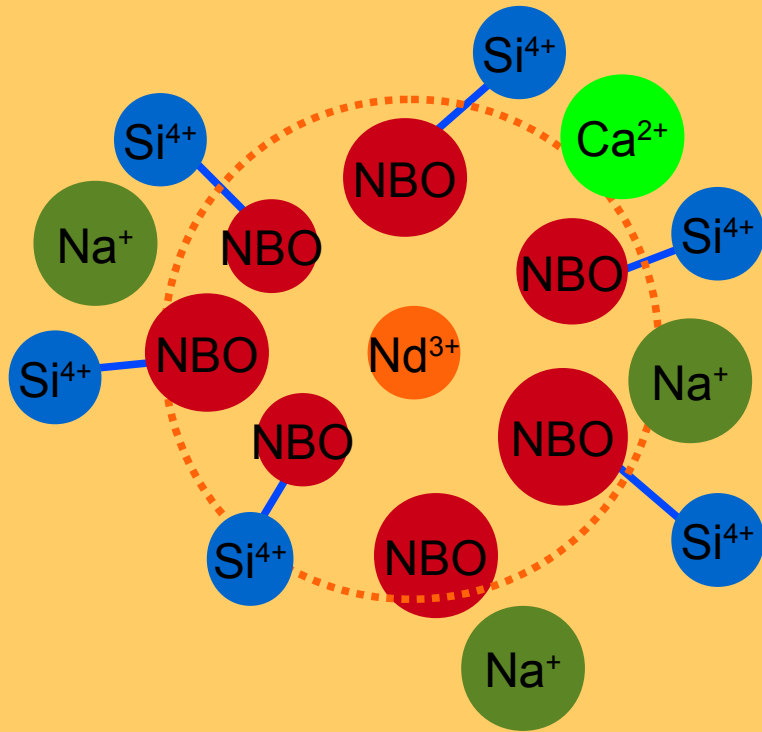
assumed identical in
Bx and R series

depend on local Nd
concentration

Nd³⁺ local concentration increases
with B₂O₃ content

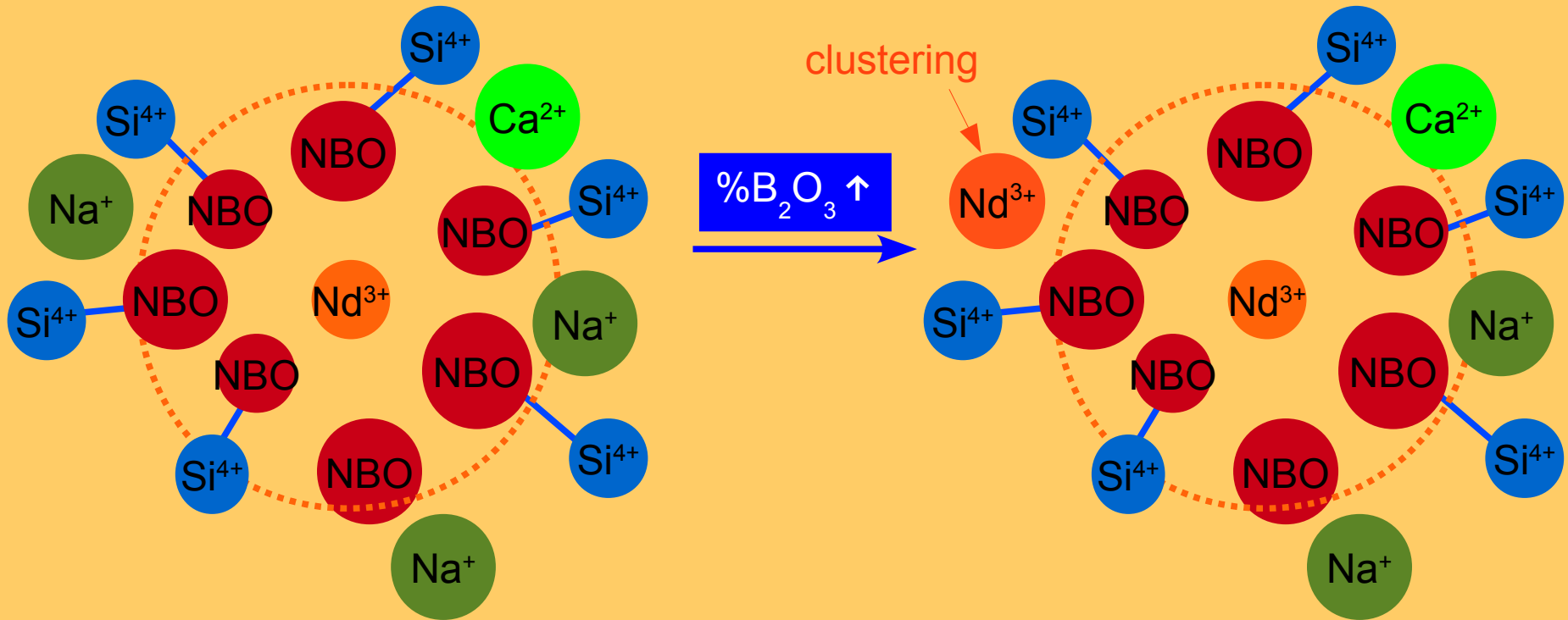
This effect is obvious even at low
B₂O₃ and is not due only to Ca/Na
substitution around Nd³⁺

Summary of spectroscopic results on Nd³⁺ local environment



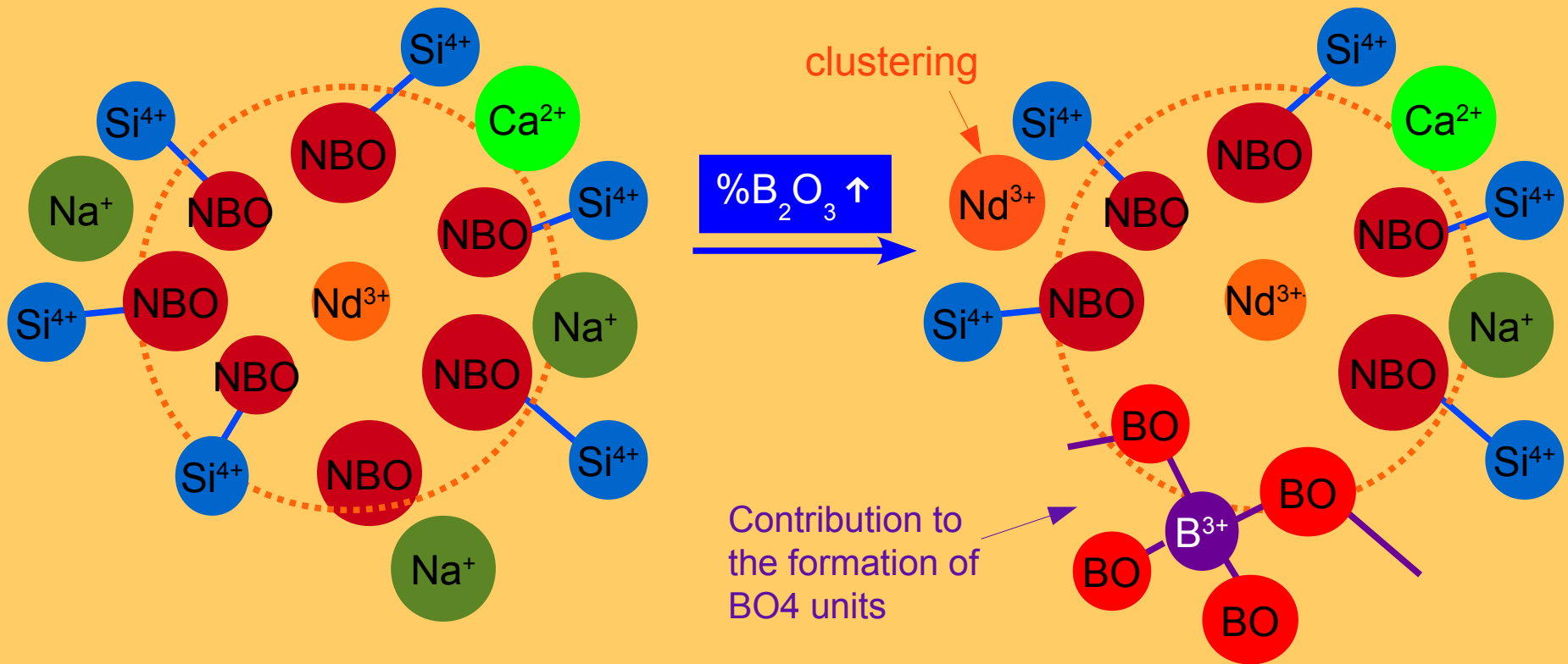
B0 glass : well-defined
~7-fold coordination
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Summary of spectroscopic results on Nd^{3+} local environment



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Summary of spectroscopic results on Nd³⁺ local environment



B0 glass : well-defined
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sphere

Spectroscopic data can not discriminate
between :

- Nd clustering and change of Ca/Na distribution around Nd
- Formation of BO₄ units within a B-enriched borosilicate phase

Implications on the glass thermal stability

Slow cooling at 1°C/min



← Nd^{3+} ions charge
compensation by other
species than Ca^{2+} and Na^{+} →



Crystallization tendency of
 $\text{Ca}_2\text{Nd}_8(\text{SiO}_4)_6\text{O}_2$ apatite decreases

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~~Nd clustering~~
~~Change of Ca/Na ratio around Nd~~

~~Nd_2O_3 reaction with B_2O_3 within a
B-enriched borosilicate phase ?~~

Implications on the glass thermal stability

Slow cooling at 1°C/min



← Nd³⁺ ions charge compensation by other species than Ca²⁺ and Na⁺ →
Crystallization tendency of Ca₂Nd₈(SiO₄)₆O₂ apatite decreases

← CaO and Nd₂O₃ contribute to form BO₄ units →

Implications on the glass thermal stability

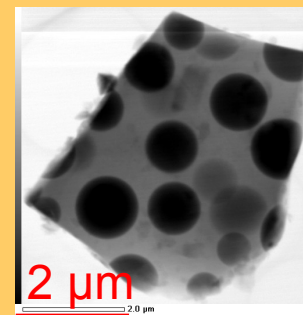
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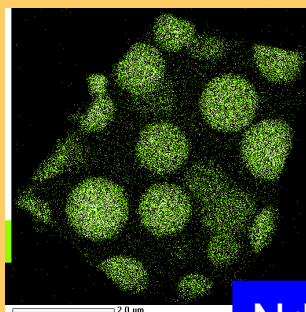
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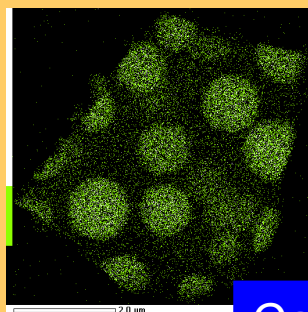
CaO and Nd_2O_3 contribute to
form BO_4 units



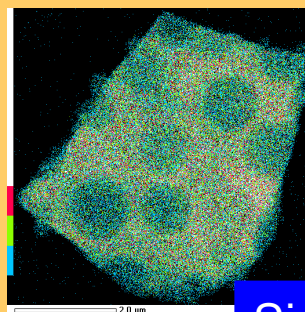
Phase
separation in
B18, B22 and
B27 after slow
cooling (1°C/min)



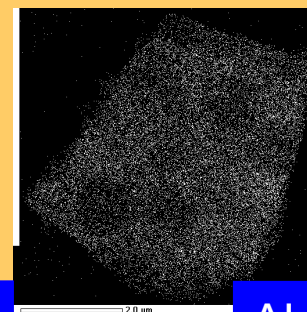
Nd



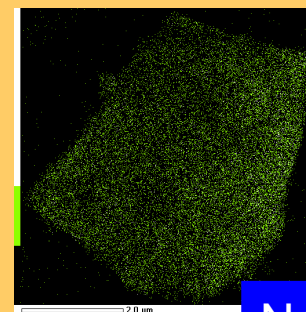
Ca



Si



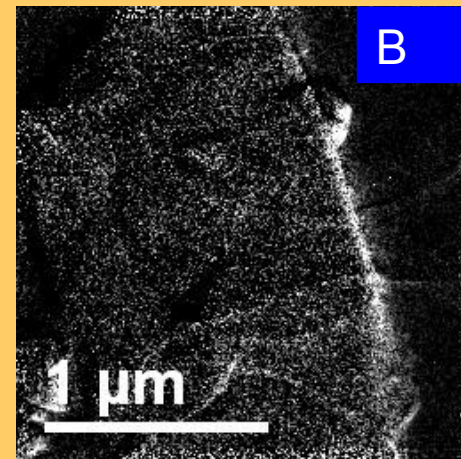
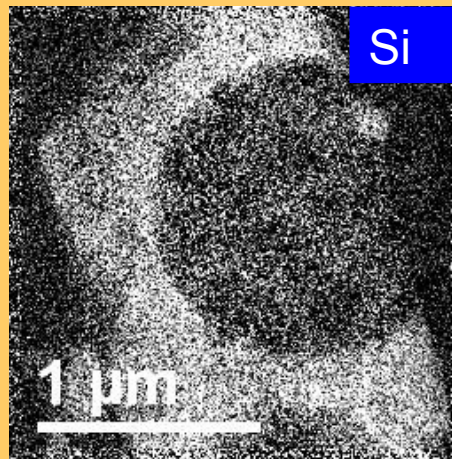
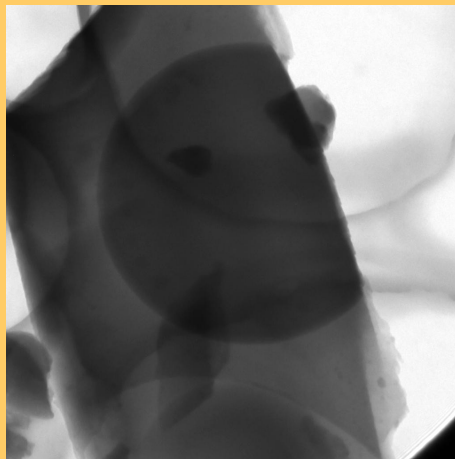
Al



Na

Implications on the glass thermal stability

EFTEM at the B and Si K-edges on B27 glass cooled at 1°C/min



Contrast at the Si K-edge, **not at the B K-edge !**
Si in excess is expelled outside the Nd- and Ca-rich phase

Conclusions

Glass A = composition range where CaO and Na_2O are not sufficient to charge-compensate all Al_2O_3 , ZrO_2 , B_2O_3 (as BO_4 units) and Nd_2O_3

Observation : when B_2O_3 content increases, the number of BO_4 units also increases (same proportion 40 %)

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→ At least at high B_2O_3 content, BO_4 units are formed by reaction of Nd_2O_3 with B_2O_3

→ At low B_2O_3 content, Ca/Na redistribution around Nd and Nd clustering are possible, but contradict observations about the crystallisation tendency. Thus there is likely a reaction of Nd_2O_3 with B_2O_3

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Observation : when B_2O_3 content increases, the number of BO_4 units also increases (same proportion 40 %)

→ At least at high B_2O_3 content, BO_4 units are formed by reaction of Nd_2O_3 with B_2O_3

→ At low B_2O_3 content, Ca/Na redistribution around Nd and Nd clustering are possible, but contradict observations about the crystallisation tendency. Thus there is likely a reaction of Nd_2O_3 with B_2O_3

A polymerized SiO_2 component appears at a very small scale (< TEM scale) Nd^{3+} and Ca^{2+} cluster within a borosilicate, Si-depleted, depolymerized phase. This is not exactly what could be inferred from extension of the Dell and Bray model (formation of a B-enriched depolymerized phase).

Questions and perspectives

- What medium-range structure for this complex inhomogeneous glass ?
 - What about the remaining boron in BO_3 units ? How are they accommodated in the silicate polymerized component ? Does CaO and Nd_2O_3 also depolymerize BO_3 units ?
 - Are there borate rings or superstructural units ?
- Is the reaction of Nd_2O_3 with B_2O_3 (forming BO_4 units and maybe depolymerized BO_3 units) only due to the lack of modifier ions in this particular composition range?
 - ...Or is it a preferential reaction occurring even at larger Na_2O , CaO contents ?

Need for « medium-range order sensitive tools » (simulations, NMR...) for the study of rare earth-rich alkali borosilicate glasses...

Thank you for your attention !

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