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# Facile and Reproducible Synthesis of Nanostructured Colloidal ZnO Nanoparticles from Zinc Acetylacetonate: Effect of Experimental Parameters and Mechanistic Investigations

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A facile and reproducible route to nanostructured colloidal ZnO nanoparticles was developed by controlled hydrolysis and condensation of zinc acetylacetonate in alkaline conditions. By reaction of an ethanolic solution of  $Zn(acac)_2$  with NaOH in a 1:2 molar ratio, after reflux, ZnO spherical nanoparticles were obtained that displayed a homogeneous size distribution; particle diameters ranged from 6 to 10 nm, as evidenced by transmission electron microscopy (TEM) analysis. The same reaction was carried out also in water, glycerol and 1,2-propanediol, to investigate the effect of the solvent viscosity and dielectric constant on the final features of the obtained material. Irrespective of the nature of the solvent, X-ray diffraction (XRD) analysis shows the formation of hexagonal ZnO, whereas the presence of residual unreacted  $Zn(acac)_2$  could be ruled out. Indeed, different particle sizes and very different morphologies were obtained. Also the reflux step was shown to be a key factor in avoiding the fast precipitation of a floc and achieving a pure compound, which

was isolated and thoroughly characterised. The composition of the obtained ZnO was determined by elemental analysis, X-ray photoelectron spectroscopy (XPS) and thermogravimetric analysis (TGA), showing the formation of pure ZnO. IR spectroscopy evidenced the presence of adsorbed organic ligands on the colloid surfaces. Diffuse reflectance infrared Fourier transform spectroscopy (DRIFT) revealed the presence of medium- to high-strength acidic sites on the ZnO surface. To gain a deeper insight into the formation mechanisms of these nanostructures, time-resolved UV/Vis and XAS studies were performed on the ethanol solution used for the synthesis of the oxide and also on the solid specimen, obtained after the refluxing step. No remarkable changes could be evidenced in the solution after the addition of an understoichiometric amount of NaOH, but the growth of the ZnO nanoparticles could be followed by UV/Vis spectra. (© Wiley-VCH Verlag GmbH & Co. KGaA, 69451 Weinheim, Germany, 2009)

## Introduction

Research in the field of ZnO nanostructures of different shapes and morphologies has witnessed an amazing development in the last 10 years.<sup>[1]</sup> This interest can be mainly traced back to the electrical and optical properties of ZnO<sup>[2]</sup> as well as to its outstanding electronic properties and functional performances,<sup>[3]</sup> which are in many cases related to defectivity.<sup>[4]</sup> Also its photocatalytic properties have been recently extensively reviewed.<sup>[5]</sup> Among the very different morphologies that ZnO can display (nanowires, nanotubes,

nanoparticles, nanoplatelets, nanowhiskers, etc.).<sup>[6]</sup> colloidal nanosized particles are particularly appealing for their functional properties and also because the spherical shape enhances the contribution of the intrinsic properties of ZnO with respect to that provided by the presence of defects. These suspensions can be approached by classical methods of colloid chemistry<sup>[7]</sup> as well as by solvo- and hydrothermal methods.<sup>[8]</sup> Different synthesis approaches to ZnO colloids have been reviewed by Spanhel.<sup>[9]</sup> Recently, Niederberger, Pinna et al. have developed an effective and easy nonaqueous route to prepare crystalline metal oxide nanoparticles, including ZnO ones.[10]

The outstanding functional properties of ZnO nanoparticles push the research towards the development of new, easy, low-cost and reproducible preparative routes, yielding pure nanostructured ZnO colloids.<sup>[1]</sup>

In the present work, we address the synthesis of ZnO by using a wet-synthesis route based on the use of zinc acetylacetonate as precursor and four different solvents as dispersing media. Zinc acetylacetonate has been extensively used as a volatile precursor for the vapour and MOCVD



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deposition of ZnO.<sup>[11]</sup> ZnO could also be obtained, starting with zinc acetylacetonate, by pyrolysis processes,<sup>[12]</sup> chemical spray deposition,<sup>[13]</sup> laser-induced deposition<sup>[14]</sup> or epitaxial growth.<sup>[15]</sup> Liu et al. have produced ZnO nanoparticles by degradation of a zinc acetylacetonate precursor in oleylamine,<sup>[16]</sup> whereas Chory et al. have grown ZnO nanoparticles from ethanol solutions using acetate, acetylacetonate and dimethylpropylenediamine as stabiliser agents.<sup>[17]</sup> Acetylacetonate has been used as a precursor also by Epifani, Iwasaki et al.<sup>[18,19]</sup>

However, to the best of our knowledge, this is the first study devoted to the synthesis of ZnO in these particular conditions and using different dispersing media. In this framework, the main goal of the present work was to develop a facile, reproducible, cost-effective, low-temperature and fast synthesis route for the production of monodispersed ZnO colloidal suspension. A related purpose was to investigate the effect of the chemicophysical nature of the dispersing medium on the formation and the evolution of the colloidal suspension. To this aim, time-resolved UV/Vis and EXAFS studies were also carried out.

## **Results and Discussion**

# Time-Resolved Studies on the Nucleation and Growth of ZnO Nanoparticles

Firstly, an attempt to follow the nucleation and growth of ZnO nanoparticles was carried out by time-resolved UV/ Vis and EXAFS measurements. Zinc acetylacetonate is a stable chelate complex, which, upon reaction with NaOH in ethanol under reflux, yields ZnO nanoparticles. The controlled particle growth during nucleation is essential in order to produce nanoparticles of reasonable size distribution. From the literature,<sup>[20]</sup> the reaction steps that occur when Zn(acac)<sub>2</sub> reacts with an ethanol solution of NaOH are



Figure 1. The  $k^3$ -weighted EXAFS function (left) and its Fourier transform (right) along with the theoretical models used in the fit of (a) solid anhydrous Zn(acac)<sub>2</sub> reference (1a), (b) solution intermediate with Zn(OAc)<sub>2</sub> treated with NaOH in ethanol (1b) and (c) Zn(acac)<sub>2</sub> in EtOH after addition of NaOH (1c).

known to be the decomposition of  $acac^{-}$  ligand and the subsequent formation of Zn–O–Zn bonded species during the reflux step. Moreover, few investigations have been carried out on the various equilibria between the zinc complexes in solution. Consequently, the identification of the main species that undergoes hydrolysis has been performed by EXAFS measurements. The present EXAFS investigation aims to confirm the presence of such Zn–O–Zn-type intermediates using the local structure of zinc, obtained from EXAFS data analysis. The experimental  $k^3$ -weighted EXAFS spectra of solid anhydrous Zn(acac)<sub>2</sub> (1a), Zn(acac)<sub>2</sub> dissolved in EtOH and treated with NaOH (1c), and Zn(OAc)<sub>2</sub> dissolved in EtOH and treated with NaOH (1b) are presented in Figure 1 along with their Fourier-transformed EXAFS spectra.

From the comparison of the  $k^3$ -weighted EXAFS spectra, the oscillations in k-space and the Fourier transform peaks for samples **1b** and **1c** look different. On the other hand, **1b** looks similar to **1a**, that is anhydrous  $Zn(acac)_2$ , which is trimeric (see Figure 2).



Figure 2. Molecular structure of trimeric zinc acetylacetonate used in the EXAFS modelling.

This observation gives an indication that the solution intermediate obtained from Zn(acac)<sub>2</sub> is different from that obtained by starting from Zn(OAc)<sub>2</sub>. However, compared to solid anhydrous Zn(acac)<sub>2</sub>, the spectra of the liquid solution (1c) show a slight increase in intensity of the peak at around 1.9 Å and a decrease in intensity of the peak at around 3.2 Å. In order to gain accurate insight into the type of the reaction species in solution, the structure of the starting solid material has to be known precisely. The experimental EXAFS spectra of the anhydrous Zn(acac)<sub>2</sub> were fitted with theoretical EXAFS phase and amplitude functions calculated using FEFF6 for the crystal structure of anhydrous Zn(acac)<sub>2</sub>.<sup>[21]</sup> The fit to the experimental data in R-space (Fourier transformed EXAFS) using the theoretically calculated FEFF paths of trimeric Zn(acac)<sub>2</sub> as a twosite model was performed by the iteration of the change in half path length  $\Delta R$ , Debye-Waller-like factor  $\sigma^2$  and energy shift  $E_0$  for each path, with fixed values for path degeneracy. In the fitting model, the distorted octahedral arrangement of oxygen around the Zn2 site and a distorted trigonal bipyramidal coordination of five oxygen atoms around Zn1 were considered (see Figure 2). The Zn-Zn distances were constrained for both Zn1 and Zn2 (see Exp. Section) centres to be equal. Within the error limits, the obtained fits were statistically satisfactory and the resulting model structure, presented in Table 1, was used as a yard stick to evaluate the changes that might be evidenced in the zinc species present in ethanol solution. When the experimental

EXAFS spectrum of 1c was compared with the solid reference 1a, no considerable changes were observed in the position of the FT peaks. The monitoring of the reaction of an ethanol solution of Zn(acac)<sub>2</sub> with NaOH and the possible reaction mechanism involving different intermediates has been reported earlier in the literature.<sup>[20]</sup> In the above-mentioned study, gas liquid chromatography (GLC) and highpressure liquid chromatography (HPLC) were used to monitor the products of C-C bond cleavage in the ligands and as an inference the local structure around zinc was postulated in the mechanism. In this work, a direct probe of the local structure at the Zn K-edge using EXAFS was used to obtain the local structure around zinc. A strong FT peak at about 1.9 Å and less intense peaks at 2.7 and 3.2 Å are seen in the FT EXAFS spectra of 1c, similar to the Zn- $(acac)_2$  reference. However when this model was used in the fitting, it resulted in a statistically poorer fit. Similarly poor fits were obtained when the models consisting of either  $Zn_4O(CH_3COO)_6$  or  $Zn(OH)_2$  were used. However, the model consisting of four Zn-O neighbours and two Zn-C neighbours resulted in a statistically good fit. The model and the fitting parameters obtained as results are presented in Table 1. Therefore, from the EXAFS investigations, no specific structure could be assigned to the intermediate formed in solution and only the immediate local environment around zinc could be ascertained. This is due to the fact that at a given time, during dissolution of the stable chelate complex, or during NaOH addition, different species coexist in solution and what is obtained as an EXAFS spectrum is actually an average of different local structure contributions. This limitation of EXAFS is well known. Also, neither a polymeric nor cluster model structure of  $Zn_4O(COOCH_3)_6$  species reported while using zinc acetate precursor for ZnO synthesis could be evidenced by the EXAFS fitting procedure because the inclusion of higher Zn-Zn coordination paths resulted in poorer fits. In summary, it can be concluded that the local structure around zinc is not found to change dramatically upon dissolution in EtOH. Also, upon subsequent reflux no significant changes were observed. When NaOH was added in a stoichiometric ratio, the following observations were made: the immediate Zn-O and Zn-C coordination remain almost intact; and Zn-Zn coordination similar to that found in the intermediate obtained with zinc acetate precursor could not be detected in solution even after NaOH addition.

In Figure 3, the UV/Vis spectra of the suspension collected at different time intervals (0, 50, 80, 110, 230, 260, 290 min) after the addition of NaOH at a molar ratio of  $OH^{-}/Zn^{2+}$ equal to 0.016 at 80 °C are plotted. As can be seen in Figure 3, formation of ZnO starts 110 min after the addition of NaOH. A shift at higher wavelength with increasing time is observed, which can be traced back to the growth of the nanoparticles. Nanoparticles having a diameter of 4 nm were detected at 260 min, whereas at 290 min they had grown to 7 nm. These values are slightly higher than those detected by TEM and XRD (vide infra), being in the range 6–10 nm, but still in the range of the standard deviation.

Sample	Path	Degeneracy <sup>[a]</sup>	Amplitude <sup>[a]</sup>	$\sigma^2$ (Å <sup>2</sup> )	$E_0$ (eV)	$R_{\rm eff}$ (Å)	$\Delta R$ (Å)	<i>R</i> (Å)
1a site 1	Zn1↔O	1.0	0.67	0.0054	-0.189	1.971	-0.023	1.948
	Znl↔O	1.0	0.67	0.0054	-0.189	2.002	-0.005	1.997
	Zn1↔O	1.0	0.67	0.0054	-0.189	2.008	-0.010	1.998
	Znl↔O	1.0	0.67	0.0038	-0.189	2.054	0.026	2.081
	Zn1↔O	1.0	0.67	0.0038	-0.189	2.096	0.035	2.131
	Zn1⇔Zn2	1.0	0.67	0.0074	-0.189	3.259	-0.023	3.236
1a site 2	Zn2↔O	2.0	0.33	0.0030	-0.189	2.014	-0.045	1.969
	Zn2↔O	2.0	0.33	0.0038	-0.189	2.108	-0.123	1.985
	Zn2↔O	2.0	0.33	0.0038	-0.189	2.167	-0.106	2.060
	Zn2⇔Zn1	2.0	0.33	0.0074	-0.189	3.259	-0.023	3.236
1b	Zn↔O	1.0	1.0	0.0259	6.778	1.936	0.243	2.179
	Zn↔O	3.0	1.0	0.0057	6.778	1.946	0.036	1.982
	Zn⇔Zn	3.0	1.0	0.0146	6.778	3.162	0.080	3.242
1c	Zn↔O	2.0	1.0	0.0035	6.172	1.999	0.064	2.064
	Zn↔O	1.0	1.0	0.0030	6.172	2.005	-0.042	1.964
	Zn↔O	1.0	1.0	0.0030	6.172	2.011	-0.042	1.969
	Zn⇔C	1.0	1.0	0.0040	6.172	2.959	-0.011	2.948
	Zn⇔C	1.0	1.0	0.0040	6.172	2.981	-0.011	2.970

Table 1. The parameters obtained from the fitting procedure for different paths for the samples 1a, 1b and 1c.

[a] Parameters were fixed during the iteration;  $E_0$ : passive electron reduction factor,  $\sigma^2$ : Debye–Waller-like factor,  $R_{\text{eff}}$ : initial path length,  $\Delta R$ : change in path length, R: final path length.



Figure 3. UV/Vis spectra after: (a) 290, (b) 260, (c) 230, (d) 110, (e) 80, (f) 50, (g) 0 min of reflux time.

# Synthesis and Characterisation of Nanocrystalline ZnO in Different Dispersing Media

#### Preparation without Heating: Effect of the Heating Step

ZnO suspensions were prepared starting from  $Zn(acac)_2$ in four different solvents (ethanol, water, glycerol and 1,2propanediol) and by heating at 80 °C for two hours.

The composition, the microstructure and the morphology of all obtained samples were investigated. For the specimen prepared starting from ethanol suspension, the effect of the reflux step was also taken into account.

It was found that longer heating times did not affect the average size of the nanoparticles, as particles heated for 2 or 6 hours practically displayed the same dimension. Furthermore, it is worth noting that different replicates of the same experiment lead to identical results both in terms of composition and structure as well as of average particle size, thus confirming the reproducibility of the route. On the other hand, the absence or presence of the heating step was proven to dramatically affect the evolution of the systems in ethanol. In fact, suspensions prepared without heating, but just stirring the mixture of  $Zn(acac)_2$  and NaOH at room temp., yielded a precipitate that was a mixture of ZnO (*wurtzite*) and a further Zn-based compound with reflections at lower angles and a high amount of organic residual (as revealed by XPS and IR). In this case IR also revealed the band of acetylacetonate and the presence of OH groups, although the XRD pattern was completely different from that of Zn(acac)<sub>2</sub> (see Figure S1 in Supporting Information).

SEM micrographs of the ZnO powders obtained without the inclusion of a thermal step (see Figure 4) show a very peculiar morphology characterised by a combination of microrods and large, more compact, aggregates.



Figure 4. SEM micrographs of ZnO powders before thermal treatment collected at high (a) and low (b) magnification level.

The EDX spectra (see Figure 5) of these systems revealed that the C signal corresponding to the undecomposed acetylacetonate groups, as confirmed by IR and XPS analysis, is still present in the material. Furthermore, EDX mapping investigations (data not shown) evidence that organic residuals are predominantly located on the microrods, while the compact macroaggregates are predominantly composed of ZnO matrixes.



Figure 5. EDX spectrum of ZnO powders before thermal treatment.

The decomposition of acetylacetonate to acetate, as suggested by Inubushi et al.,<sup>[20]</sup> was ruled out on the basis of the XRD pattern (see Figure S1).

#### Preparation with Heating Step

In a first approximation, we can consider the diffusion as inversely related to the viscosity by the well known Stokes– Einstein–Sutherland equation.<sup>[22]</sup> It is expected that, other conditions being equal, an increase of the viscosity of the dispersing medium could lead to a decrease of the diffusion coefficient and thus affect the growth rate of the colloidal particles.

One of the aims of the present study was to point out whether the different viscosities and dielectric constants of the solvents could either slow down or accelerate the growth rate of the colloidal particles and affect the distribution of their sizes.

The growth mechanism of ZnO nanocrystals in suspension, typically obtained by reaction of zinc acetate with sodium hydroxide, has been the topic of several extended investigations,<sup>[9b,23]</sup> which have pointed out that the growth is affected not only by the composition of the suspension, but also, as expected, by the temperature of the suspension. Although there are reports on non-Ostwald ripening growth mechanisms,<sup>[23a,24]</sup> the almost universally accepted mechanism for growth of semiconductor nanocrystals is the diffusion-limited Ostwald ripening.<sup>[25,26]</sup> In the past, it has been reported<sup>[25c,23f]</sup> that the growth of ZnO in water, on the basis of the dependence of average diameter on time ( $d \approx t^{1/3}$ ), follows the common accepted Ostwald ripening path. However, more recent contributions,<sup>[23a]</sup> based on the evaluation of the dependence of the growth kinetics on temperature and composition, invalidate this oversimplified approach and show instead a dependence which is different from that expected in an exclusively diffusion-limited scenario.

In particular, more recent reports have evidenced that the growth of ZnO colloids can be explained as determined by the occurrence of different events and can be interpreted as an intermediate between the two limiting models, that is diffusion-limited Ostwald ripening and kinetically controlled growth.

In our case, the molar ratio between zinc acetylacetonate and NaOH was kept constant (1:2), as well as the reaction temperature (80 °C). The only difference between the four routes is the nature of the suspension media used, which are characterised by different viscosity, dielectric constant and  $pK_a$ , as summarised in Table 2.

The powders isolated by filtering and drying the suspensions of ZnO were characterised by TEM and XRD, providing information on structure and crystallite size. As mentioned, different batches always present the same XRD patterns, thus confirming the reproducibility of the preparative route.

The solids precipitated after the reflux show the typical XRD patterns of hexagonal *wurtzite* zinc oxide,<sup>[27]</sup> whereas no reflections ascribed to residual zinc acetylacetonate were detected. In Figure 6, the diffractograms of the four solids are plotted. The broadened reflections can be ascribed to the presence of nanosized domains.<sup>[28]</sup>

The most striking result is the broadening of the peaks in the samples obtained with 1,2-propanediol and ethanol. This is evidenced also by the (200), (112) and (201) peaks merging into a single broad signal. The widening of the diffraction peaks is indicative of a reduction in crystallite size. The average sizes of the crystallites were quantified by the Scherrer equation<sup>[28]</sup> (see Table 2 and also Table S1 in Supporting Information). For example, the size of crystallites in the direction normal to the (101) planes was 33 and 22 nm for the samples obtained by water and glycerol,

Table 2. Dielectric constants, viscosities and  $pK_a$  values for the different solvents used for synthesis of nanostructured ZnO and corresponding average diameters as determined by TEM and XRD analyses.

Sample	3	η (cp)	pK <sub>a</sub>	Average diameter (TEM) (nm)	Average diameter (XRD) (nm)
ZnO_EtOH	25.0 (25 °C)	0.504 (70 °C) 1.2 (20 °C)	15.9	$6\pm 2$	8 ± 2
ZnO_Gly	42.5 (25 °C)	629 (30 °C) 1490 (20 °C)	14.15	>200	21±9
ZnO_Prop	37.0 (20 °C)	3.02 (80 °C) 19.9 (20 °C)	14.22	$6\pm 2$	$7 \pm 1$
ZnO_H <sub>2</sub> O	78.5 (25 °C)	0.3547 (80 °C) 1.002 (20 °C)	15.75	20–200	31±9



Figure 6. X-ray diffraction pattern of nanostructured ZnO prepared with (a) water, (b) ethanol, (c) 1,2-propanediol and (d) glycerol. The indexing and, on the bottom, the bars correspond to the hexagonal structure of ZnO.

whereas it was much decreased in the case of samples prepared with ethanol and 1,2-propanediol, that is 8 and 7 nm, respectively (vide infra). These values are in agreement with those obtained by TEM, thus suggesting that each fine ZnO particle consists of a single wurtzite crystallite. It is worth noting that along the direction normal to the (110) planes the crystallite sizes for the samples obtained in ethanol and 1,2-propanediol were 7 and 6 nm, respectively, in good agreement with those reported above for the (101) planes. These results suggest a rather isotropic crystallite shape. On the other hand, as far as crystallite size is concerned, samples prepared in water or glycerol showed a certain degree of anisotropy. If the crystallite size along the normal to the (110) planes was compared to that assessed on the basis of (101) peaks, a sharp decrease was noted, with values of 23 nm for the sample obtained with water and 17 nm for that produced in glycerol.

These differences in the size of the crystallites observed in the different dispersing media were ascribed to the different growth mechanisms occurring in alcoholic and aqueous environments as a consequence of the different chemicophysical features of the different solvents used.

TEM measurements confirmed XRD results. All samples synthesised in ethanol feature a particle dimension ranging from 6 to 9 nm, whereas the average diameters obtained in the other dispersing media are reported in Table 2. Primary particles do not coalesce, but just aggregate, and show a narrow size distribution, that is a low polydispersity.

Figure 7 shows the different morphologies obtained in the different solvents. In particular, Figure 7 (a) shows an example of particles obtained in EtOH. For each agglomerate, selected area electron diffractions (SAED) were collected in order to investigate the crystalline nature of samples. SAED were integrated using the ProcessDiffraction program.<sup>[29]</sup>



Figure 7. TEM micrographs of ZnO\_EtOH nanoparticles (a), ZnO\_Prop nanoparticles (b), ZnO\_H<sub>2</sub>O showing needle-like structures (c) and star-like morphology typical of ZnO\_Gly (d).

All SAED were indexed as hexagonal zinc oxide, space group *P63mc* (Figure 8, a), in agreement with XRD data. EDXS analyses show as principal elements zinc and oxygen (Figure 8, b), whereas the presence of carbon and copper are due to the TEM grid used for sample preparation.



Figure 8. (a) SAED of ZnO\_EtOH nanoparticles and (b) EDXS of ZnO\_EtOH nanoparticles.

To obtain statistical information about the size distribution, image analysis was performed. Because of the tendency of nanoparticles to agglomerate, it is impossible to perform image analysis on micrographs like Figure 7a, therefore dark field images were used for this purpose. The dark field allows only some particles to be highlighted, to avoid the problem of particle superimposition in images of agglomerates and also to improve the image contrast. Obviously the information about the majority of particles is lost, but the lower particle superimposition allows the analysis to be performed. The average diameter of ethanol samples ranges between 6 and 9 nm and the distributions are quite close to the mean value.

Concerning the samples synthesised in 1,2-propanediol (see Figure 7, b), the average particle diameter ranges from 6 to 10 nm, in agreement with XRD data and the distributions are quite close to the mean values also in this case.

All the collected SAED were indexed as hexagonal ZnO. EDXS spectra (data not shown) of the samples confirm zinc and oxygen as the main components. Traces of sodium were also detected. In these samples, the presence of sodium was explained by the fact that the Na<sup>+</sup> counterions are attracted by the hydroxy ions around the nanocrystals forming a kind of virtual capping layer.

The samples synthesised in water indeed display a needlelike morphology (Figure 7, c) with thickness in the order of 20–100 nm and length more than 500 nm. Electron diffraction shows that the crystalline phase is hexagonal zinc oxide (see Figure S2 of Supporting Information).

Concerning the sample synthesised in glycerol, it features a star-like structure with particle dimensions larger than 200 nm (Figure 7, d). Electron diffraction shows that the crystalline phase is still hexagonal zinc oxide. EDXS analysis again displays zinc, sodium and oxygen as the main components (data not shown).

The Scherrer formula applied to XRD data revealed an average crystallite size of  $(21 \pm 9)$  nm, suggesting that the larger agglomerates observed in TEM micrographs are likely constituted by smaller crystallites retaining their integrity.

The morphology of the heated samples is very different from that observed in the samples prepared at room temperature. As can be observed in Figure 9 (a, b) and in the corresponding EDX spectrum (Figure 10), the heating step process also results in a dramatic change in the sample morphology with respect to those obtained at room temperature.



Figure 9. SEM micrographs of ZnO powders after reflux thermal treatment collected at high (a) and low (b) magnification level.

SEM pictures of ZnO powders obtained after the reflux thermal treatment at 80 °C, collected at high and low magnification level (Figure 9), confirm the dramatic influence



Figure 10. EDX spectrum of ZnO powders after reflux thermal treatment.

of thermal procedure on the system morphology. In fact, inspection of Figure 9 shows the formation of a highly porous material characterised by small nanoaggregates, as confirmed also by TEM investigation. The EDX compositional analysis of these systems (see Figure 10) proves both the complete disappearance of the organic residuals, found in the untreated powders, and the high purity of these ZnO nanoaggregates.

XPS measurements were performed on ZnO obtained from 1,2-propanediol, water, ethanol and glycerol to get information about the composition and the chemical environment of zinc and oxygen atoms. O1s peaks were deconvoluted using commercial software. The BE values for the Zn2p<sub>3/2</sub> and O1s regions, along with the O/Zn atomic ratios, are summarised in Table 3.

Table 3. BE (eV) values for  $Zn2p_{3/2}$  and O1s and atomic percentage ratios O/Zn relative to the four ZnO samples synthesised from different media. Values of BE were corrected considering charging effects.

BE (eV)	EtOH	1,2-Propanediol	H <sub>2</sub> O	Glycerol
Zn 2p3/2 (eV)	1021.3	1021.1	1021.1	1021.6
Ols	529.8	530.0	529.4	530.4
	530.8	531.5	530.9	531.7
	532.4	532.9	532.2	532.9
O/Zn (atomic%)	1.1	2.0	1.1	1.3

In Figure 11 (a, b), the Zn2p region and the deconvolution of O1s are reported for ZnO obtained from ethanol. For the other samples, see the Supporting Information (Figures S3, S4, S5).

For ZnO, tabulated BE values for  $Zn2p_{3/2}$  are between 1021.1 and 1022.4 eV while for  $Zn(OH)_2$  the value is 1022.7 eV.<sup>[30,32,34,35]</sup> The experimental values (see Table 3) for the four samples are in the range 1021.1–1021.6 eV, values similar to zinc oxide, whereas they are quite different from those reported for  $Zn(OH)_2$ . However, the deconvolution of O1s in three components reveals the presence of OH species at the surface. Peak components corresponding



Figure 11. (a) Zn2p region for ZnO\_EtOH; (b) deconvolution of O1s region for ZnO\_EtOH.

to OH were at 530.8, 531.5, 530.9 and 531.7 eV for samples obtained from ethanol, 1,2-propanediol, water and glycerol, respectively.

The presence of OH groups is confirmed also by FTIR analysis, which reveals for all the samples the typical broad band at 3390-3370 cm<sup>-1</sup> of the O–H stretching (Figure 12).



Figure 12. FTIR spectra for: (1)  $ZnO_H_2O$ , (2)  $ZnO_Prop$ , (3)  $ZnO_EtOH$ , (4)  $ZnO_Gly$ .

As can be seen in Table 3, the O/Zn% atomic ratios are quite close to the stoichiometric ratio (1), except in the case of 1,2-propanediol.

The atomic percentage of carbon is between 35% and 40% for samples in ethanol, 1,2-propanediol and glycerol, while for water the percentage is lower, 20%, as confirmed by elemental analysis. These are typical values for adventitious carbon but, as evidenced by FTIR, the presence of organic ligand 2,4-pentanedionate from the precursor must be taken into account.

The specific surface area of the ZnO nanoparticles synthesised in EtOH, as determined by BET measurements, was  $(64 \pm 1) \text{ m}^2/\text{g}$ , with an average pore diameter of 78 Å.

#### **Surface Acidities**

Concerning the ZnO prepared in ethanol, the surface acidic active sites were investigated (with DRIFT spectroscopy) by chemisorbing pyridine. The DRIFT spectra, recorded after exposure of the ZnO prepared in ethanol to a pyridine/argon mixture at room temperature, and successively to an argon stream for different periods of time, are shown in Figure 13. Inspection of the spectral region of the ring-stretching modes reveals the presence of several contributions at 1441, 1580, 1589 and 1611 cm<sup>-1</sup> (Figure 13, a). Comparison with literature data suggests the existence of different kinds of pyridine–ZnO interactions. The contributions at lower wavenumber are consistent with the pyridine H-bound to the surface hydroxy groups (1441, 1580 and 1589 cm<sup>-1</sup>). The peak at higher wavenumber is ascrib-



Figure 13. DRIFT spectra obtained at room temp. after exposing the ZnO prepared in ethanol to pyridine + Ar (gray line) and successively to Ar for 7 min (black line), 15 min (dotted line) and 25 min (dashed line). (a) Pyridine stretching modes spectral region. (b) O–H stretching spectral region.

able to pyridine interacting with Lewis acidic sites distributed on the ZnO surface.<sup>[31]</sup>

After evacuation with Ar flow at room temp. the bands due to H-bound pyridine progressively decrease, while the contribution at 1611 cm<sup>-1</sup> remains almost unchanged, suggesting that the pyridine coordinated to Lewis acidic sites is very strongly tied up. This confirms the presence of medium- to high-strength acidic sites on the ZnO surface.

Parallel to the above spectral changes, pyridine interaction brings about a severe perturbation of the surface hydroxy groups by creating a rather broad adsorption  $3686 \text{ cm}^{-1}$  (Figure 13, b). The behaviour as a function of evacuation time suggests that the pyridine interacting by H bond is mainly responsible for the perturbation.

### Hydrolysis and Condensation towards ZnO

As far as the initial formation of the first ZnO "seeds" is concerned, as already mentioned, neither XAFS nor UV/ Vis provided evidence of the formation of polynuclear species.

In the quoted paper by Pinna, Niederberger et al.,[10d] the authors prepare ZnO nanoparticles starting from zinc acetylacetonate and benzylamine. Also in the reported study, as in the present case, a nonaqueous route is used to produce nanostructured zinc oxide. In the paper by Pinna et al., the zinc acetylacetonate is nucleophilically attacked by benzylamine, and a reaction mechanism based on the C-C cleavage of acetylacetonate is proposed on the basis of GC-MS results. In the present case, the nucleophilic attack is instead carried out by NaOH. Beside the similarities (nonaqueous medium, nucleophilic attack, use of zinc acetylacetonate as precursors), there are however some relevant differences such as the solvothermal reaction conditions applied by Niederberger et al.<sup>[10d]</sup> as well as the use of benzylamine as solvent, which means it cannot be inferred that the same mechanism occurs also in our case. In the study presented in this paper, the role of the dispersing medium seems to be pivotal.

Although it is not yet clear how, the nature of the solvent strongly affects the size and the morphology of the particles. In fact, the average particle diameter obtained in the different dispersing media (see Table 3) is small (about 8 nm) in EtOH and 1,2-propanediol and larger in the case of water and glycerol (31 and 21 nm, respectively). Moreover, whereas in the case of the former two media, spherical and well separated nanoparticles are formed, in the case of water needle-like crystals consisting of crystallite of about 21 nm, with an overall size of about 200 nm, are observed. In the case of glycerol, star-like particles with an average crystallite size of 21 nm can be seen.

As far as the nucleation and growth fashion and particle size in the different media are concerned, a rationalisation relating the chemicophysical nature of the different media on the mechanism of hydrolysis and condensation is quite challenging and did not provide any satisfactory conclusion. In Table 2, three relevant chemicophysical properties  $(pK_a)$ , dielectric constant and viscosity at room and final temperatures) are reported for the four media. As a matter of fact, no relationship among viscosity, acidity, dielectric constant and crystallite size could be observed, which would be expected in the case of an oversimplified growth model based only on diffusion. On the basis of these considerations, it becomes quite difficult to single out which solvent characteristic (other conditions being equal) is actually responsible for determining the different morphologies, sizes and size distributions observed, although it can be hypothesised that a major role is played by the dielectric constant.

## Conclusions

In the present work we describe an easy, reproducible, cost-effective and fast synthesis route for the production of ZnO colloids. The synthesis was carried out starting from  $Zn(acac)_2$  in four different solvents to investigate the effects of the physicochemical properties of the dispersing medium on the nucleation and growth of the oxide nanostructures.

Formation of hexagonal ZnO nanostructures was observed in all cases, whereas the different nanoscopic architectures and morphologies were evidenced as a function of the different media. On the surface of the nanoparticles, the presence of acetylacetonate residuals was detected. The heating step was evidenced to play an important role in affording the formation of pure ZnO. The investigation of the kinetics of the process by time-resolved EXAFS was very difficult due to the coexistence of different species and different equilibria in solution, delivering only an averaged picture.

Moreover, this study has evidenced that the combination of complementary analytical methods with different selectivity, sensitivity, time and spatial resolution is mandatory to collect information on the process to obtain ZnO nanoparticles. This study has also outlined that the examined system is too complicated to unambiguously determine the sequence of chemical events involved in the process. As a final remark, the proper combination of methods delivering complementary and mutually integrating information is fundamental to avoid errors and misinterpretation of data arising from separate experiments.

## **Experimental Section**

**Chemicals:** Zinc 2,4-pentanedionate purchased from Aldrich was anhydrified under vacuum on  $P_2O_5$  for 48 h. Sodium hydroxide, ethanol, 1,2-propanediol and 1,2,3-propanetriol (glycerol) were purchased from Aldrich, Milan, Italy and used without further purification. Anhydrous KBr was purchased from Merck, Germany.

**Chemicophysical Characterisation of Zn(acac)**<sub>2</sub>: FTIR (KBr):  $\tilde{v} = 3000$  (w), 2966 (w), 2919 (w), 1591 [v(C=O), s], 1520 [v(C=C), s], 1444 [δ(CH<sub>3</sub>), sh], 1403 [δ(CH<sub>3</sub>), s], 1367 (s), 1259 (C–CH<sub>3</sub> and C=C, s), 1195 [δ(CH<sub>3</sub>), w], 1018 (m), 926 (m), 806 (w), 763 [δ(C–H), m], 670 [v(Zn–O), w], 553 [v(Zn–O), w], 418 [v(Zn–O), m] cm<sup>-1</sup>. <sup>1</sup>H NMR (CD<sub>3</sub>CD<sub>2</sub>OD, 298 K):  $\delta = 1.92$  (s, CH<sub>3</sub>), 5.32 (s, CH) ppm. <sup>13</sup>C NMR (CD<sub>3</sub>CD<sub>2</sub>OD, 298 K):  $\delta = 28.24$  (CH<sub>3</sub>), 100.58

(CH), 190.78 (C=O) ppm. XPS analysis (BE in eV and atomic percentages): C1s: 284.8 eV (52.2%); O1s 531.5 eV (32.7%); Zn2p 1021.6 eV (15.1%). XRD: see Supporting Information, Figure S1.  $C_{10}H_{14}O_4Zn$  (281.64): calcd. C 45.5, H 5.35; found C 45.12, H 5.51.

#### Procedures

Synthesis of ZnO in Ethanol (ZnO\_EtOH):  $Zn(acac)_2$  (2.13 g, 8.1 mmol) was dissolved in ethanol (100 mL) whilst stirring and heating. NaOH (2 equiv.) was added from a solution of NaOH (0.44 M) in ethanol. The resulting suspension was stirred at 80 °C for 6 h, then cooled to room temperature. After cooling, the fine precipitate was collected by centrifugation of the suspension at 4500 rpm for 7 min and then washed three times with ethanol. The precipitate was dried under vacuum for 24 h. ZnO\_EtOH: found C 1.00; H 0.45.

Synthesis of ZnO in 1,2-Propanediol (ZnO\_Prop):  $Zn(acac)_2$  (2.10 g, 8.0 mmol) was dissolved in 1,2-propanediol (100 mL) whilst stirring. NaOH (2 equiv.) was added from a solution of NaOH (0.44 M) in ethanol. The resulting suspension was stirred at 80 °C for 6 h, then cooled to room temperature. After cooling, the fine precipitate was collected by centrifugation of the suspension at 4000 rpm for 15 min and then washed three times with ethanol. The precipitate was dried under vacuum for 24 h. ZnO\_Prop: found C 4.89, H 0.83.

Synthesis of ZnO in Glycerol (ZnO\_Gly):  $Zn(acac)_2$  (2.12 g, 8.04 mmol) was dissolved in glycerol (100 mL) whilst stirring and heating. NaOH (2 equiv.) was added from a solution of NaOH (0.44 m) in ethanol. The resulting suspension was stirred at 80 °C for 6 h, then cooled to room temperature. After cooling, the fine precipitate was collected by centrifugation of the suspension at 4500 rpm for 15 min and then washed three times with ethanol. The precipitate was dried under vacuum for 24 h. ZnO\_Gly: found C 2.37; H 0.66.

Synthesis of ZnO in Water (ZnO\_H<sub>2</sub>O):  $Zn(acac)_2$  (2.12 g, 8.04 mmol) was suspended in distilled water (100 mL) whilst stirring and heating. NaOH (2 equiv.) was added from a solution of NaOH (0.44 M) in water. The resulting suspension was stirred at 80 °C for 6 h, then cooled to room temperature. After cooling, the fine precipitate was collected by centrifugation of the suspension at 3500 rpm for 7 min and then washed three times with ethanol. The precipitate was dried under vacuum for 24 h. ZnO\_H<sub>2</sub>O: found C 0.41, H traces.

**FTIR Data of ZnO\_EtOH:** 3350 [v(OH), br], 1554 [v(C=O), vw], 1524 [v(C=C), 2,4-pentanedionate, vw], 1065 (w), 865 (w), 435 [v(Zn-O), s] cm<sup>-1</sup>.<sup>[19]</sup>

**ZnO\_Gly:** 3370 [v(OH), br], 2931 [v(C–H), w], 2881 [v(C–H), w], 2842 (w), 1930 (br, w), 1463 [v(C–CH<sub>3</sub>), 2,4-pentanedionate, w), 1380 [ $\delta$ (CH<sub>3</sub>), 2,4-pentanedionate, w), 1120 (w), 1062 (w), 991 (w), 879 (w), 504 (s), 420 [v(Zn–O), s] cm<sup>-1.[19]</sup>

**ZnO\_Prop:** 3390 [v(OH), br], 1643 (vw), 1380 [ $\omega$ , $\delta$ (CH<sub>3</sub>), 2,4-pentanedionate, br, w], 1132 (vw), 1043 (w), 919 (vw), 842 (vw), 450 [v(Zn–O), s] cm<sup>-1</sup>.<sup>[19]</sup>

ZnO\_H<sub>2</sub>O: 3400 [v(OH), br], 450 [v(Zn–O), s] cm<sup>-1</sup>.<sup>[19]</sup>

#### **Characterisation Methods**

**FTIR Analysis:** FTIR experiments were performed with a NEXUS 870 FTIR (NICOLET), operating in the transmission range 400–4000 cm<sup>-1</sup>, collecting 64 scans with a spectral resolution of 4 cm<sup>-1</sup>. The measurements were recorded by dispersing the powders in anhydrous KBr.

XPS Analysis: The powder materials were investigated by XPS with a Perkin–Elmer  $\Phi$  5600ci instrument using standard Al- $K_{\alpha}$  radiation (1486.6 eV) operating at 350 W. The working pressure was  $\leq 5 \times 10^{-8}$  Pa, about  $10^{-11}$  Torr. The calibration was based on the binding energy (BE) of the Au4f7/2 line at 83.9 eV with respect to the Fermi level. The standard deviation for the BE values was 0.15 eV. The reported BE values were corrected for the BE's charging effects, assigning the BE value of 284.6 eV to the C1s line of carbon.<sup>[32,34]</sup> Survey scans were obtained in the 0-1350 eV range (pass energy 187.5 eV, 1.0 eV/step, 25 ms/step). Detailed scans (58.7 eV pass energy, 0.1 eV/step, 50-150 ms/step) were recorded for the O1s, C1s, Zn2p, ZnLMM and Na1s regions. The atomic composition, after a Shirley-type background subtraction,<sup>[33]</sup> was evaluated using sensitivity factors supplied by Perkin-Elmer.<sup>[34]</sup> Charge effects were partially compensated by using a charge neutraliser (flood gun). Peak assignment was carried out according to literature data.[32,34]

**Elemental Analysis:** Elemental analyses were obtained in the Microanalysis Laboratory of the Department of Chemistry of the University of Padova using a Fisons EA 1108 instrument.

**Thermogravimetric Analysis:** The thermogravimetric analyses (TGA) were performed in air on a LabSys Setarm SDT 2960 instrument in the temperature range 20–800 °C using a heating rate of 10 °C/min.

UV/Vis Absorption Analysis: The UV/Vis absorption analyses were carried out using a Cary 5 UV spectrophotometer in the spectral range 340-520 nm (reported 320-440 nm) and a quartz cuvette with a cell thickness of 1 cm. The concentration of the analysed Zn(acac)<sub>2</sub> suspension in ethanol was 0.008 M.

**X-ray Diffraction:** The XRD data were collected with a Bruker D8 Advance Diffractometer equipped with a Göbel mirror. The angular accuracy was 0.001° and the angular resolution was better than 0.01°. The average crystallite sizes were calculated by means of the Scherrer formula from the most intense peaks of each spectrum. Further wide-angle X-ray diffraction patterns were recorded with a Philips X'Pert PRO diffractometer operating in the  $\theta$ - $\theta$  geometry, equipped with a graphite monochromator on the diffracted beam (Cu- $K_a$  radiation).

TEM Analysis: Transmission electron microscopy (TEM) analysis was carried out using a Philips CM12 microscope operated at 120 keV and equipped with an energy dispersive X-ray spectrometer (EDXS). TEM samples were prepared starting from ZnO powder, ultrasonically suspended in ethanol for 5 min. For each sample a 100-µl suspension drop was put on a TEM carbon-coated copper grid. After solvent evaporation ZnO powders remained dispersed on the carbon film of the TEM grid. Samples were therefore observed in order to investigate microstructure and size distribution of particles. Several ZnO powders which differ for the solvent used during synthesis were analysed. In particular, six samples synthesised in ethanol, two in propylene glycol, one in water and one in glycerol were analysed. An attempt to observe the ZnO particles directly from synthesis solutions was carried out for three different solutions. However ZnO nanoparticles could not be clearly distinguished because of the sample heterogeneity and large amount of reactants.

**SEM Analysis:** The morphology of four ZnO samples, prepared from different solvents as described above, and their composition were analysed using scanning electron microscopy (SEM) and EDX spectroscopy. The precipitates obtained before heating the suspen-



sions at 80 °C were also investigated. Measurements were performed using a Field Emission (FE-SEM) Zeiss SUPRA 40 VP and an Oxford Inca detector for EDX analysis.

DRIFT Analysis: The IR spectra were collected with a Bruker Tensor 27 spectrometer (accumulating 32 scans at a resolution of 4 cm<sup>-1</sup>) and displayed in Kubelka-Munk units.<sup>[24]</sup> Prior to each experiment, the sample was loaded in the high-temperature highpressure (HTHP) cell installed in the COLLECTOR apparatus for diffuse reflectance infrared Fourier transform (DRIFT) spectroscopy from Spectra-Tech, Inc. fitted with ZnSe windows. Before measurement, the powder was kept in argon flow to eliminate water traces until a stable IR spectrum was obtained (about 1 h). Then the sample was exposed to the reactive species for 7 min at a flow rate of about 75 cm<sup>3</sup>/min. The background spectrum of the clean surface was measured for spectra correction. The HTHP chamber was filled with the pyridine vapours, flowing argon through a bubbler containing the liquid. Pyridine used for the chemisorption was taken from a commercial source (Sigma-Aldrich, spectroscopic grade) and used without further purification.

XAS Measurements and Analysis: The XAS measurements were performed at beamlines XAS at the synchrotron radiation source Angströmquelle Karlsruhe (ANKA) and E4 at Hamburger Synchrotronstrahlungslabor (HASYLAB) at Deutsches Elektronen-Synchrotron (DESY) Hamburg. A Si(111) double crystal monochromator was used for energy scan and the XAS spectrum of a zinc metal foil was recorded simultaneously to the XAS measurement of the samples. EXAFS spectra of the liquid samples were measured using a Teflon sample cell and Kapton foil windows, with adjustable spacers for X-ray beam path length adjustment. The XAS data processing was performed using Athena and the EXAFS data analysis using theoretical standards from FEFF6.0 was carried out using the program Artemis.<sup>[37a]</sup> For all the samples, the curve fitting of the experimental data was performed in R-space from 1.5–3.7 Å. In the iterative fitting procedure of the theoretical standard to the experimental spectra the amplitude factor was kept fixed at 1.0. In anhydrous Zn(acac)<sub>2</sub>, zinc occupies two different sites labelled as Zn1 and Zn2. The unit cell has one Zn2 site and two Zn1 sites. Obtaining a theoretical standard for this sample therefore involved the combination of EXAFS contributions from the two different zinc sites in the ratio Zn2/Zn1 = 1:2. All the EX-AFS spectra were simultaneously fitted at three k-weightings,  $k^1$ ,  $k^2$  and  $k^3$ , and the parameters reduced  $\chi^2$  and R-factor were used to compare the different fitting models. The EXAFS studies were aimed at gaining mechanistic insight into the formation of ZnO nanoparticles starting from Zn(acac)<sub>2</sub> using ethanol medium and NaOH. The starting material was anhydrous  $Zn(acac)_2$  (1a), which has, as a solid, a trimeric structure in zinc.<sup>[37b]</sup> In the trimeric structure, the central Zn atom is hexacoordinate in a distorted octahedral arrangement whereas the coordination geometry of the two terminal Zn atoms is distorted trigonal bipyramidal. Sample 1a was dissolved in boiling ethanol to form a clear solution of  $Zn(acac)_2$ , which was subsequently added with an ethanol solution of NaOH (1c). The EXAFS measurements were carried out under similar conditions, but employing zinc acetate instead of Zn(acac)<sub>2</sub>. The resulting solution intermediate (1b) has been reported earlier to be of the type Zn<sub>4</sub>O(CH<sub>3</sub>COO)<sub>6</sub>.<sup>[38]</sup> The experimental spectra of this intermediate are used as a reference in order to verify whether a similar cluster or a polymeric structure also exists in the case of  $Zn(acac)_2$  solution intermediate.

**Supporting Information** (see also the footnote on the first page of this article): Calculated average sizes of ZnO nanoparticles. Calculated EXAFS parameters. XRD patterns. EDXS for ZnO\_H<sub>2</sub>O. Three figures with deconvolutions of the O1s region for ZnO\_#.

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